

Application of the EBS NRS nanobeam in thin-film magnetism

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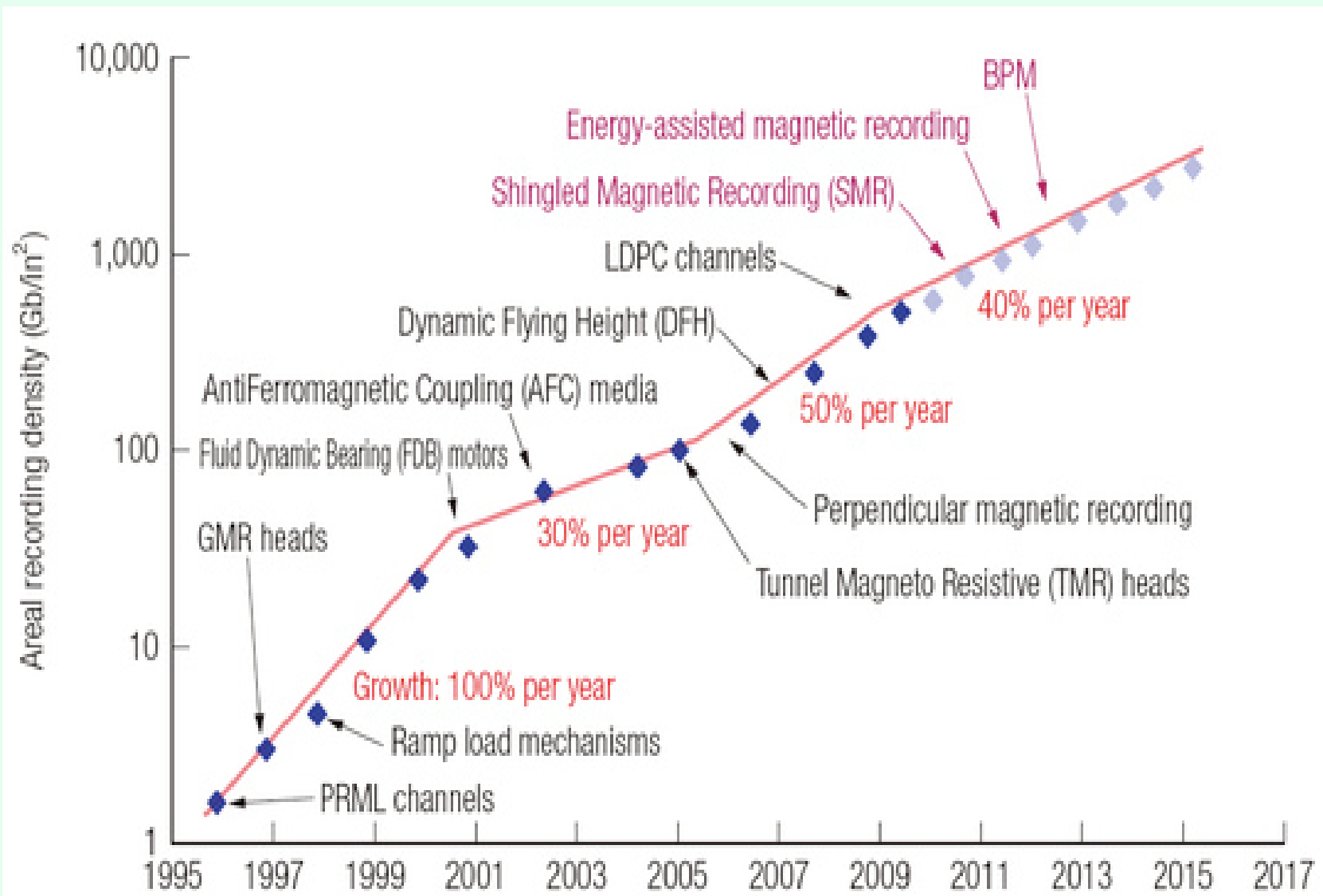
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***EBS-Workshop on Nuclear Resonance Scattering
ESRF, Grenoble, France, 11-12 March 2019***

Outline

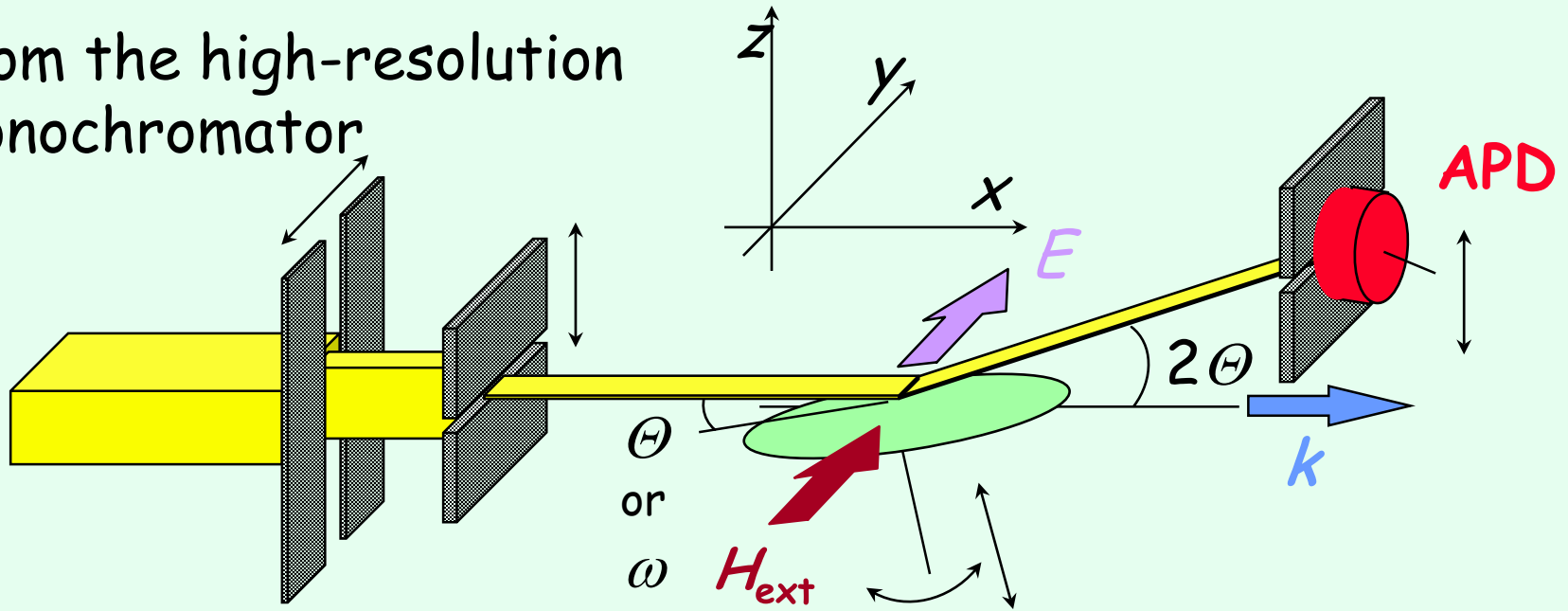
- Motivation
- Feasibility of NRS scanning magnetic microscopy with nanobeams
- Domain transformations in antiferromagnetically coupled multilayers
- Evolution of magnetism on a curved nano-surface
- Applications at grazing incidence
- Conclusions

Motivation



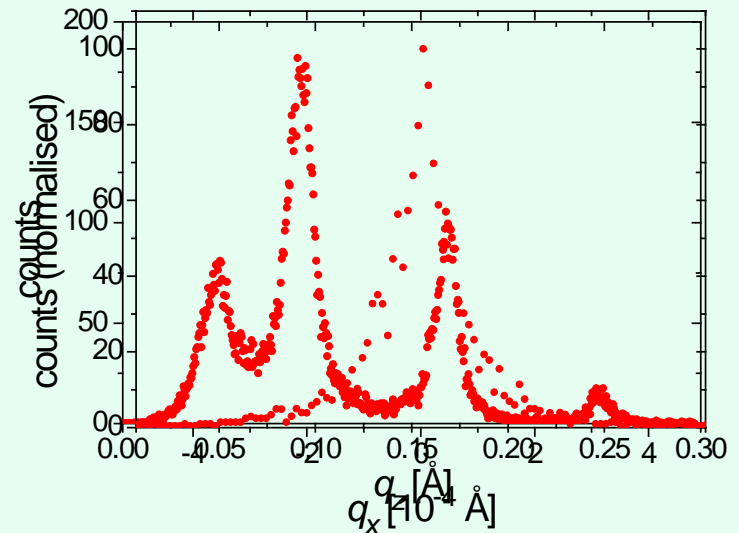
Arrangement of an SMR experiment

from the high-resolution monochromator



$$\Theta = 2\Theta_{SMR} \approx \frac{\pi}{2} \frac{\xi}{d}$$

$$\xi = \frac{1}{\Delta} \frac{\pi}{q_x}$$



Motivation

- Off-specular SMR yields information in the **reciprocal (momentum) space** only so that **no individual objects in the direct space** can be identified.
- In case of ^{57}Fe SMR, the method is practically limited to the correlation length range of about **200 nm to 2 μm** .
- Is there a chance to see thin-film magnetic phenomena in this range **in the direct space** using the EBS NRS nanobeam?

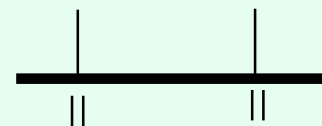
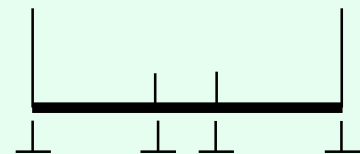
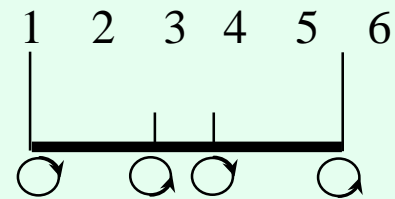
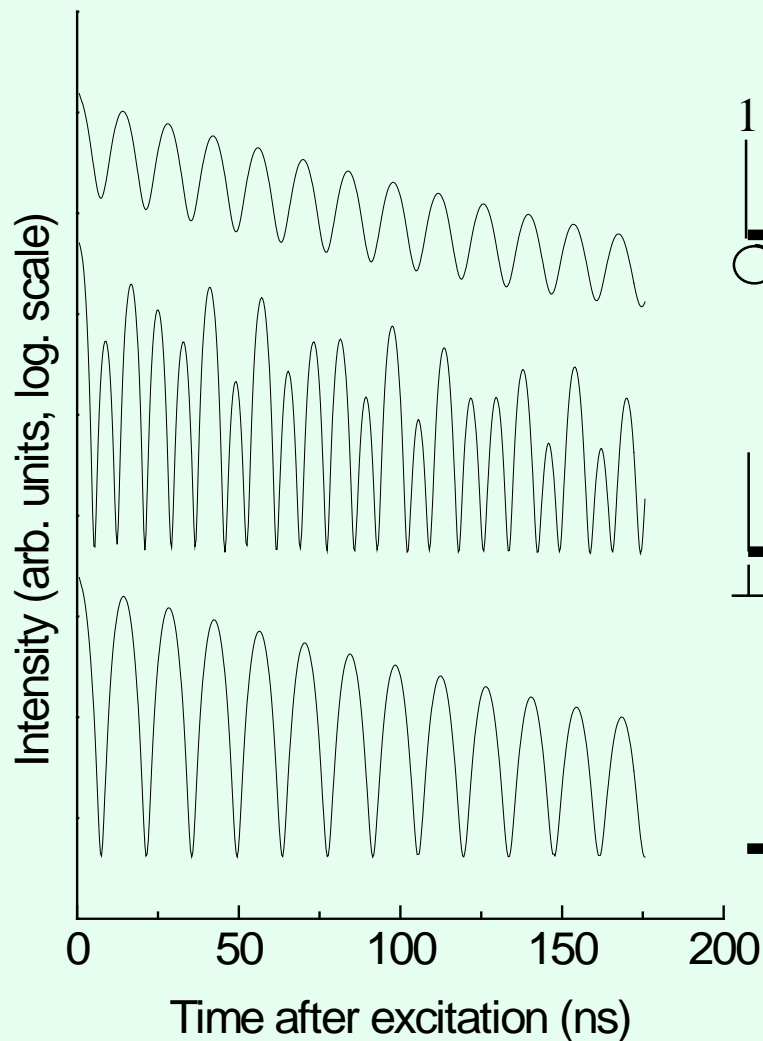
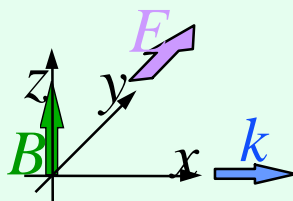
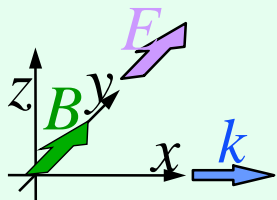
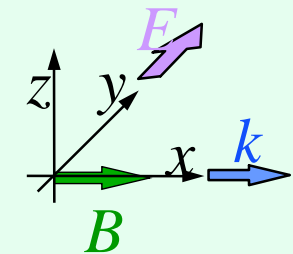
Feasibility of the approach

- **Forward scattering in thin films:** is it possible at all?
- **Electronic absorption:** for **low- Z substrates** may be low enough.

Substrate material	Transmission of 500 μm thickness for 14.41 keV
SiO ₂ (quartz)	0.4093
MgO	0.3873
α -Al ₂ O ₃ (sapphire)	0.3097
Si	0.2469
BaTiO ₃	2.8×10^{-7}

- **Effective thickness** for ⁵⁷Fe: $\lambda = 46 t [\text{nm}] f$

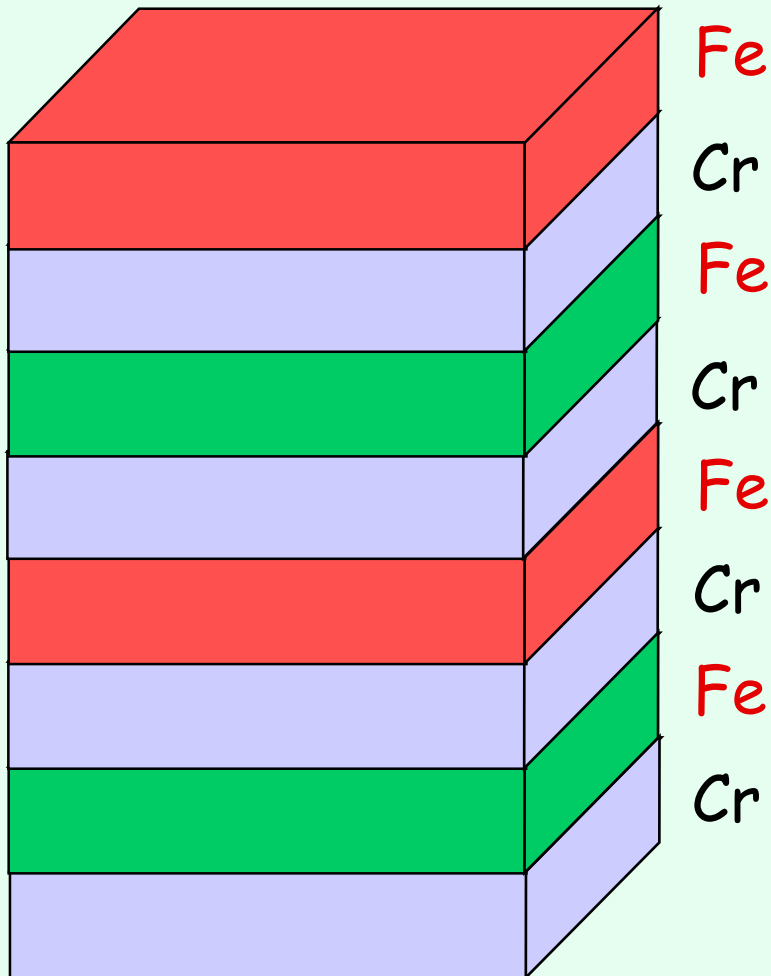
Feasibility: contrast of the hyperfine field direction (Smirnov figures)



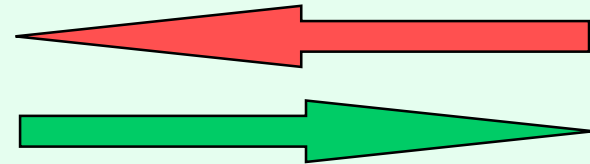
Feasibility: contrast of the hf field direction

- Hyperfine field in the y - z plane: contrast between $B_{\parallel y}$ and $B_{\parallel z}$ in linearly polarized light.
- Hyperfine field along the x axis: contrast between $B_{\uparrow x}$ and $B_{\downarrow x}$ in circularly polarized light.

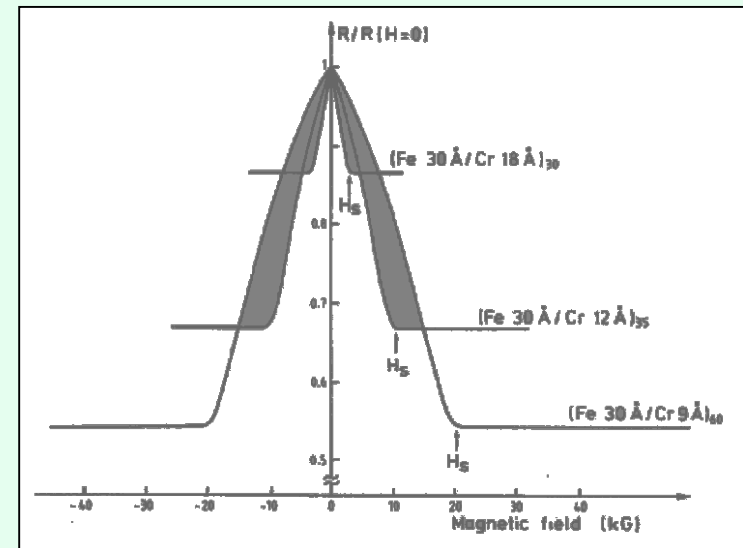
Antiferromagnetically coupled Fe/Cr multilayer



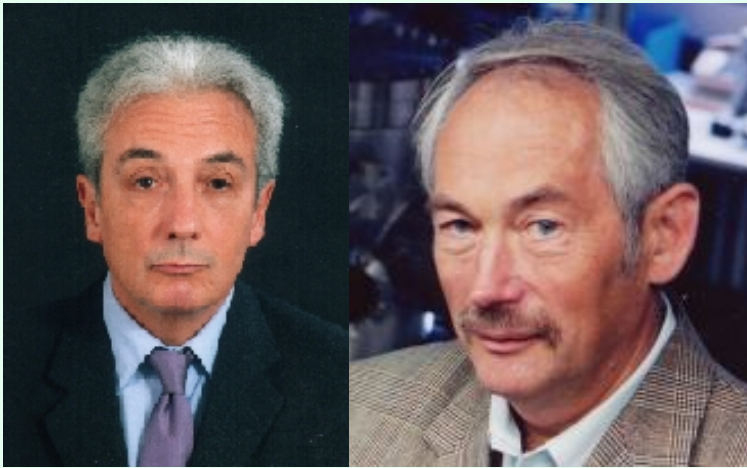
Layer magnetisations:



Giant magnetoresistance

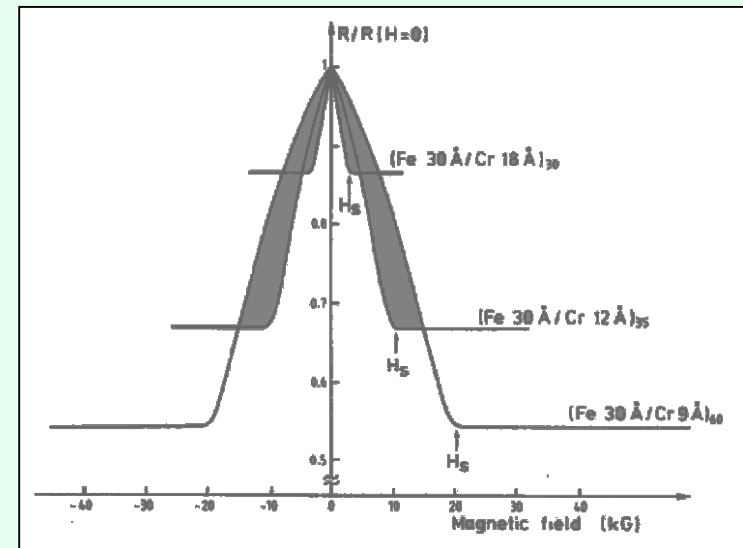


Antiferromagnetically coupled Fe/Cr multilayer

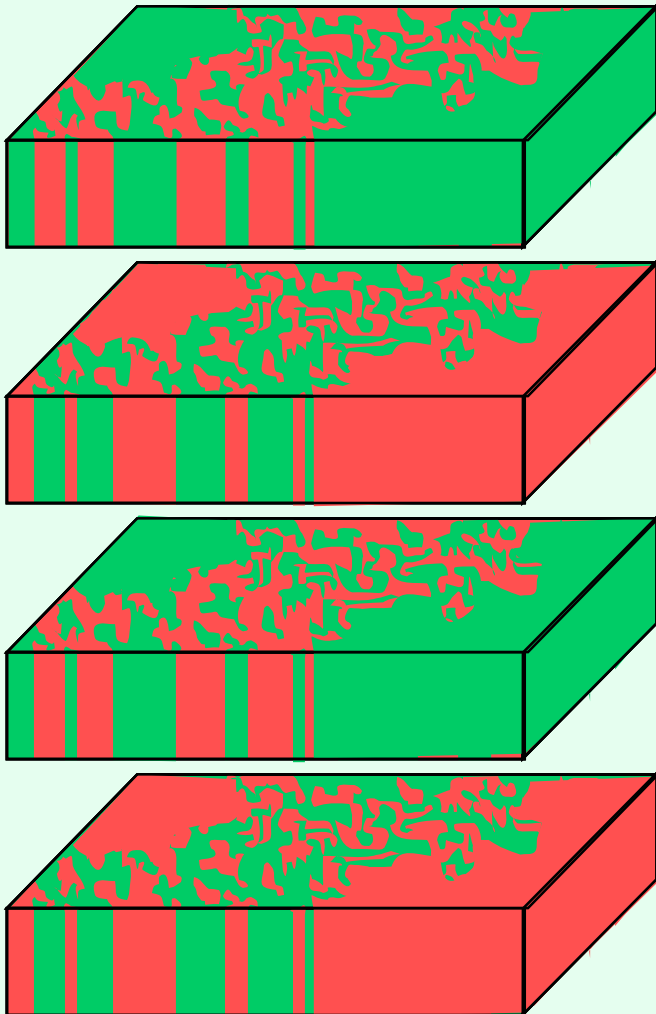


A. Fert and P. Grünberg
Nobel Prize in physics, 2007

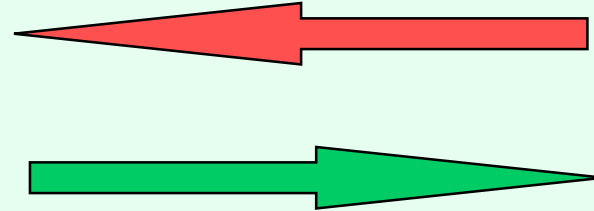
Giant magnetoresistance



Patch domains in AF-coupled multilayers



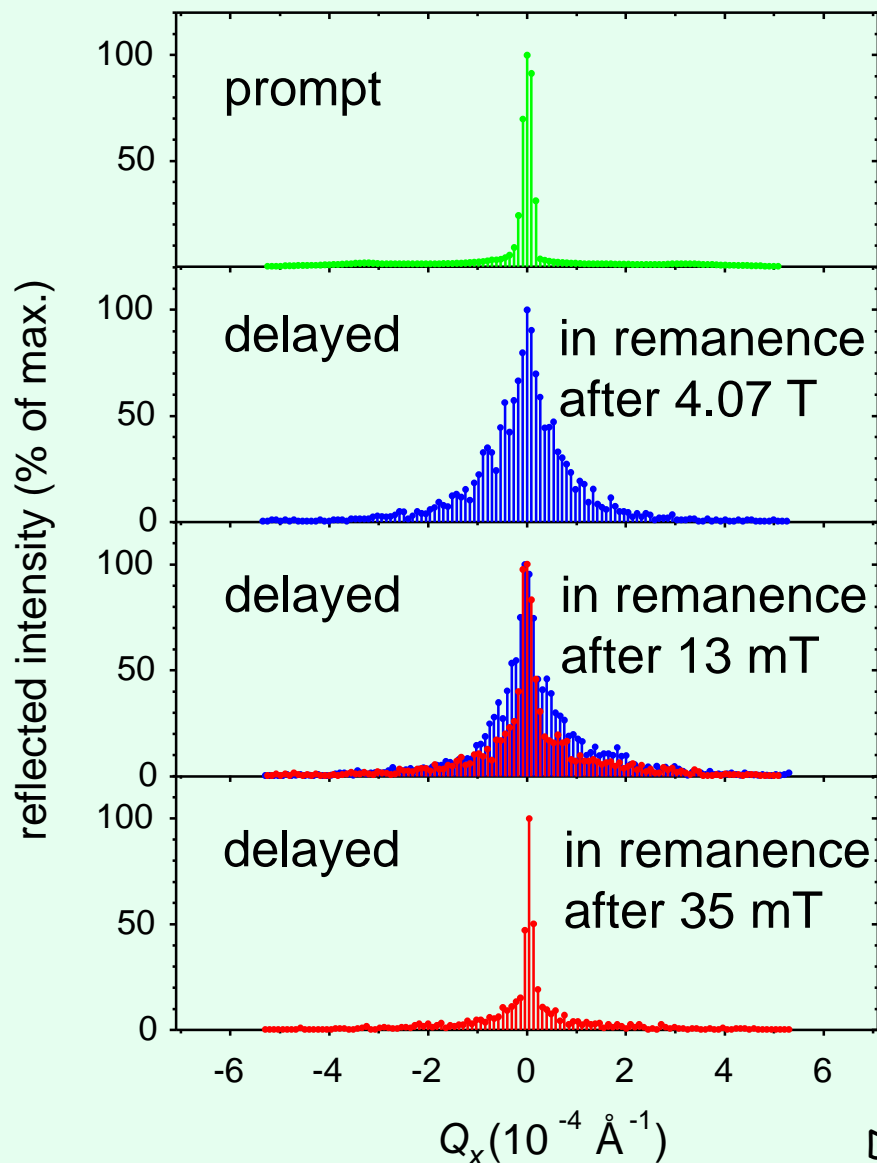
Layer magnetisations:



The 'magnetic field lines' are shortcut by the AF structure → the stray field is reduced → no 'ripple' but 'patch' domains are formed.

Spin-flop-induced domain coarsening (SMR)

MgO(001)[⁵⁷Fe(26Å)/Cr(13Å)]₂₀
 2Θ @ AF reflection, easy axis



Correlation length:
 $\xi = 1/\Delta Q_x$

← 90° rot.

Delayed photons
 before the spin flop

$$\xi = 1000 \text{ nm}$$

Delayed photons
 after the spin flop

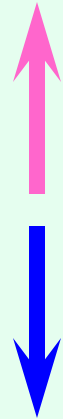
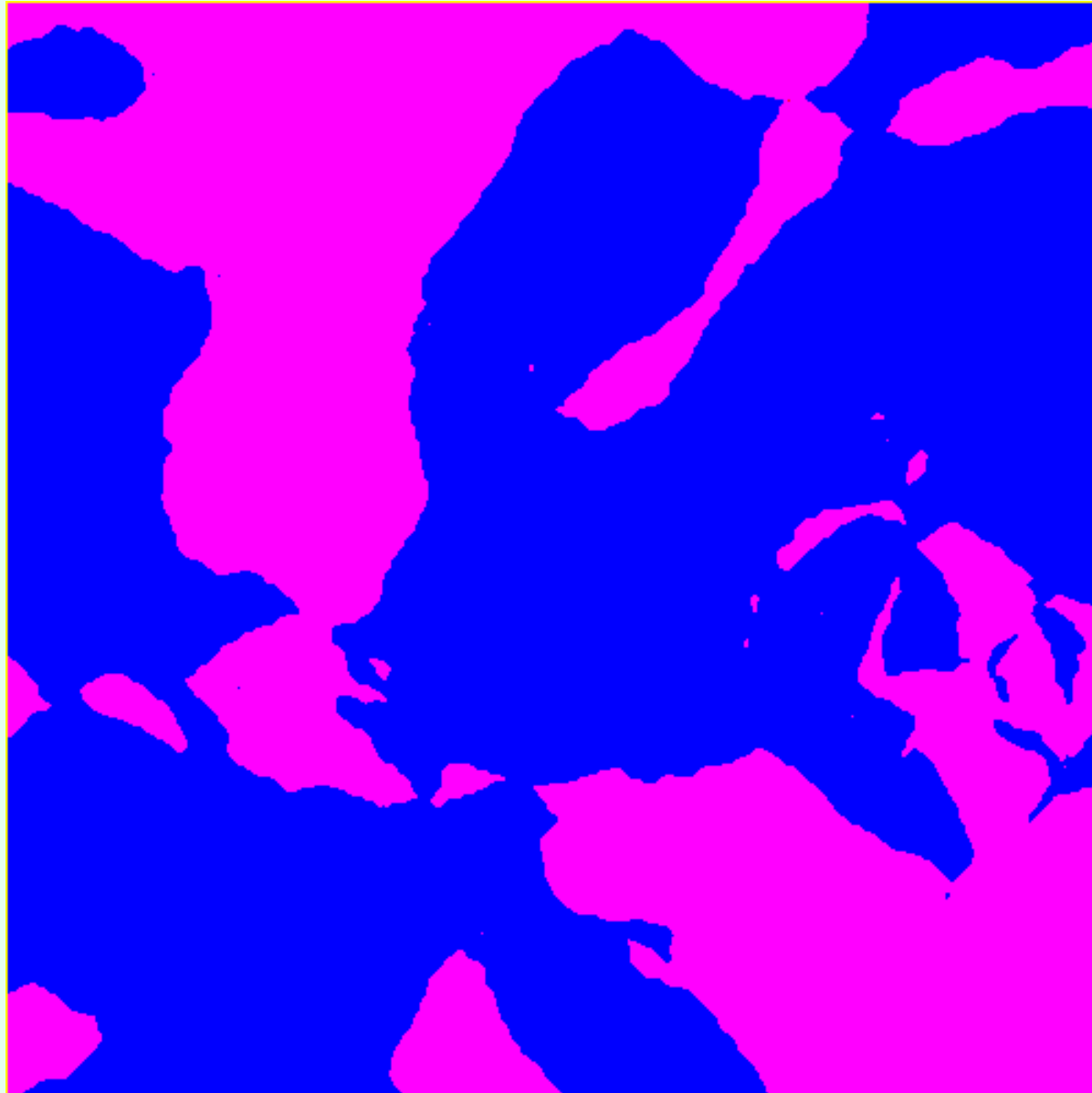
$$\xi_1 > 5000 \text{ nm}$$

$$\xi_2 = 1000 \text{ nm}$$





D.L. Nagy et al., PRL **88**, 157202 (2002).

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Domain coarsening during spin flop



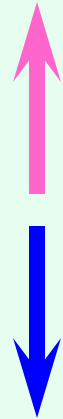
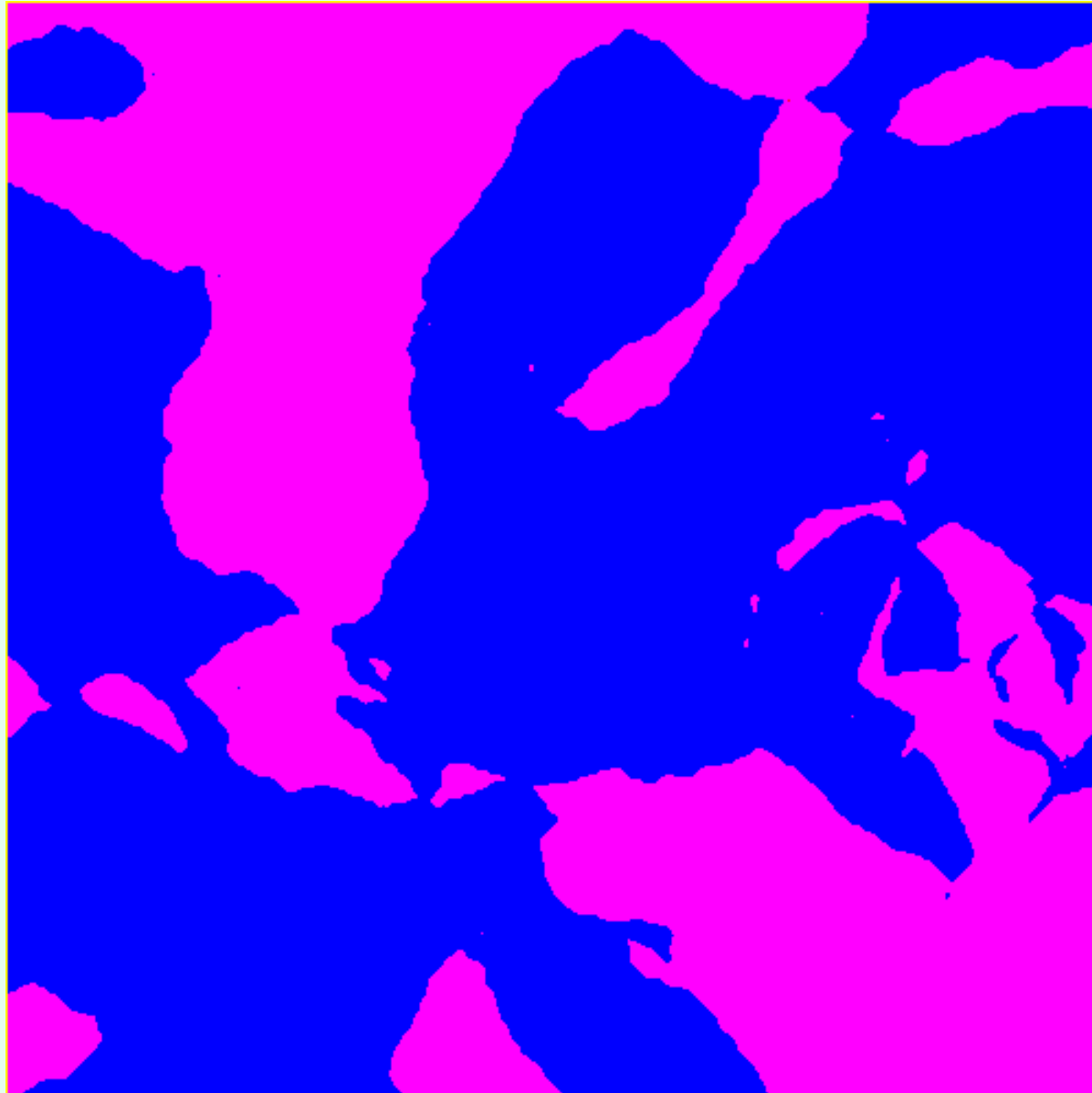
What can we see from domain coarsening?

- No contrast between  and 
The initial picture is homogeneous. ☹️
- No contrast between  and 
The final picture is homogeneous. ☹️
- The **intermediate pictures** show the step-by-step progress of the spin flop and the creation of the **large domains**. 😊
- Can we still have a contrast between domains of the **same alignment** but **opposite direction** of the layer magnetization?

What can we see from domain coarsening?

- To have a contrast between hyperfine fields of **opposite direction**, photons of **circular polarization** are needed → insert a **$\lambda/4$ phase-retarder** plate!
- AF-coupled domains still have **both directions** of the hyperfine field → the sample should be prepared by marking **every second layer** with the resonant isotope ^{57}Fe .
- To have a **k-parallel** (i.e. x -parallel) **component** of layer magnetization, the sample should be **tilted** (e.g. by 45°) around the z -axis.

Domain coarsening during spin flop

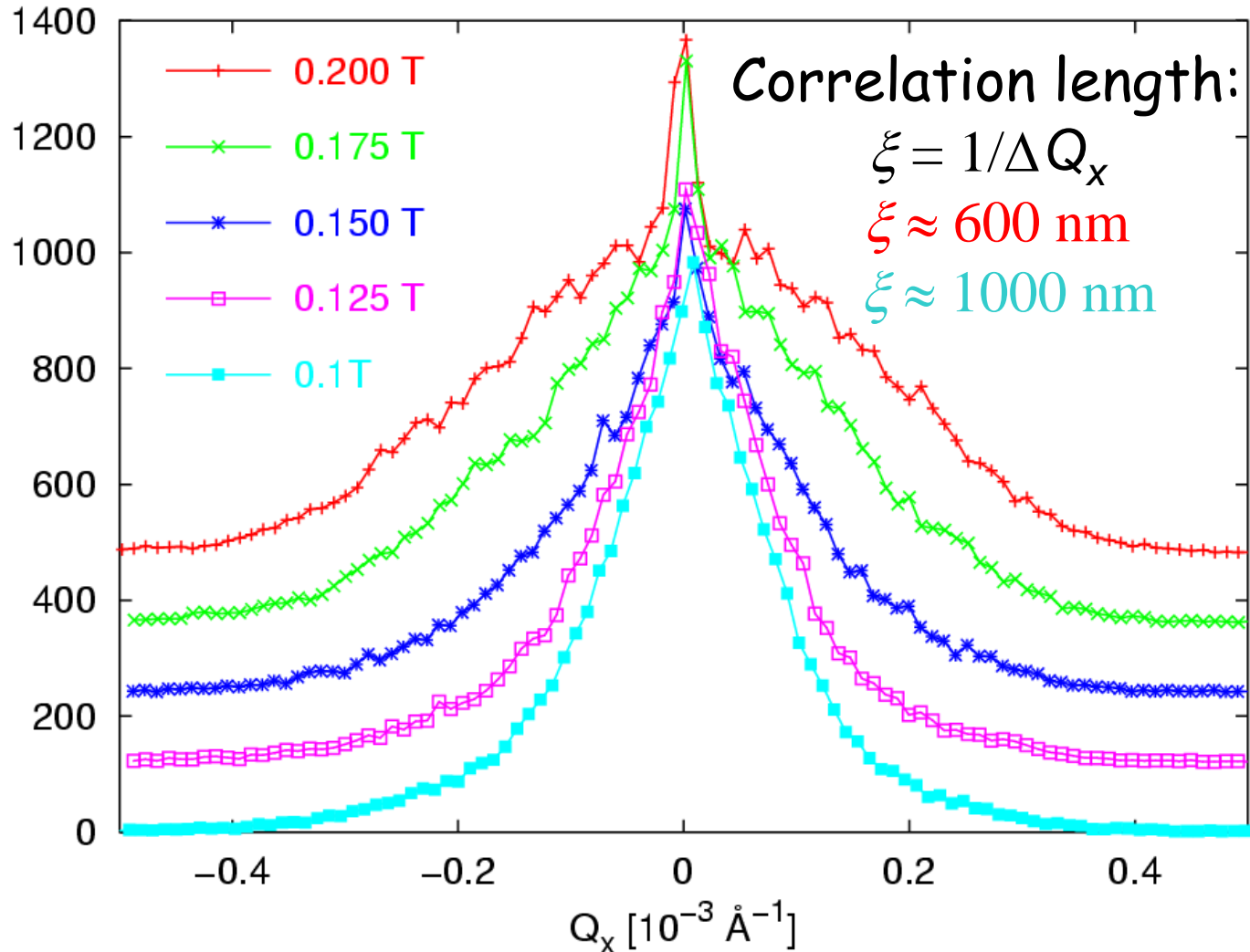


From saturation to remanence: the domain ripening

- In decreasing field the **domain-wall angle** and, therefore, the **domain-wall energy** as well as its **surface density** is increasing.
- In order to decrease the **surface density of the domain-wall energy**, the multilayer spontaneously **increases the average size** of the patch domains ('ripening').
- The spontaneous domain growth is **limited** by domain-wall pinning (**coercivity**).

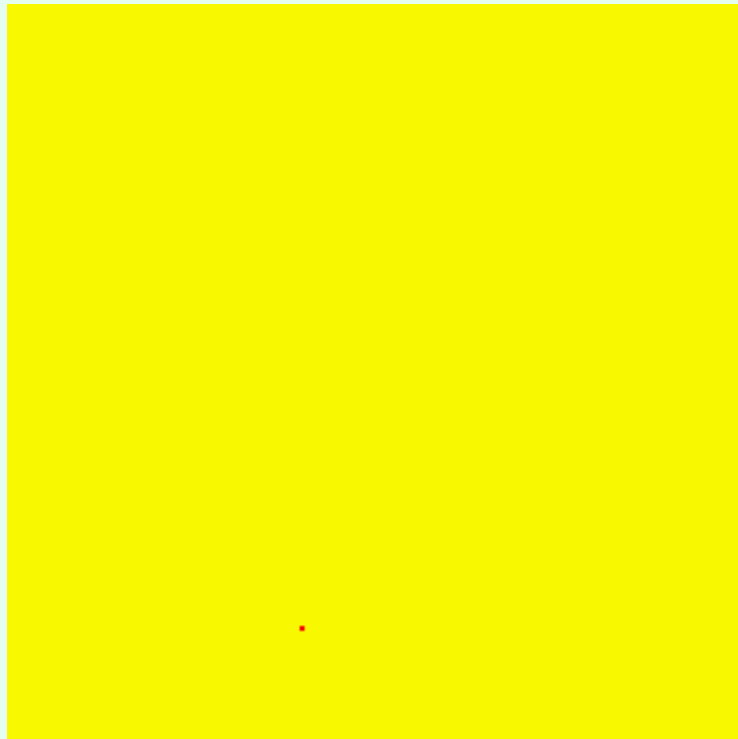
Domain ripening: off-specular SMR

MgO(001)[⁵⁷Fe(26Å)/Cr(13Å)]₂₀
2 θ @ AF reflection, hard axis



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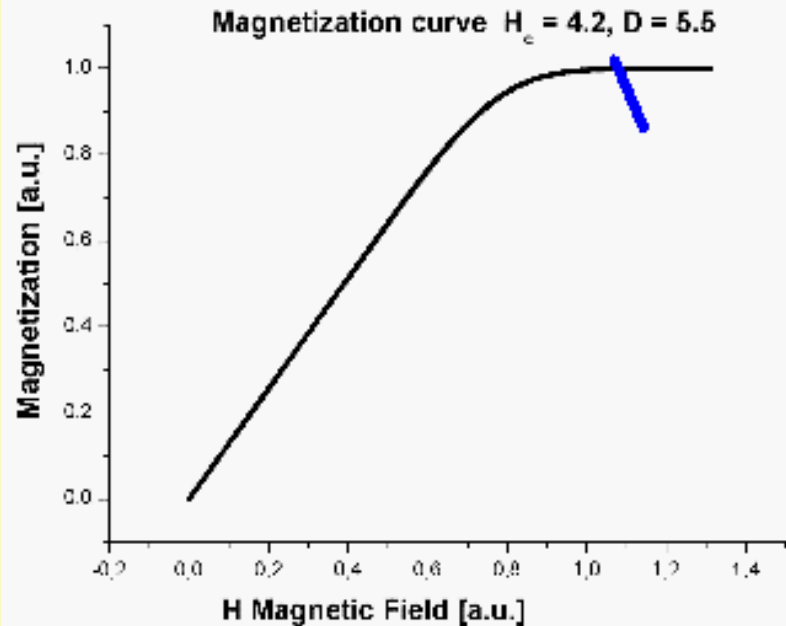
Domain formation and ripening



Domain pattern

Autocorrelation

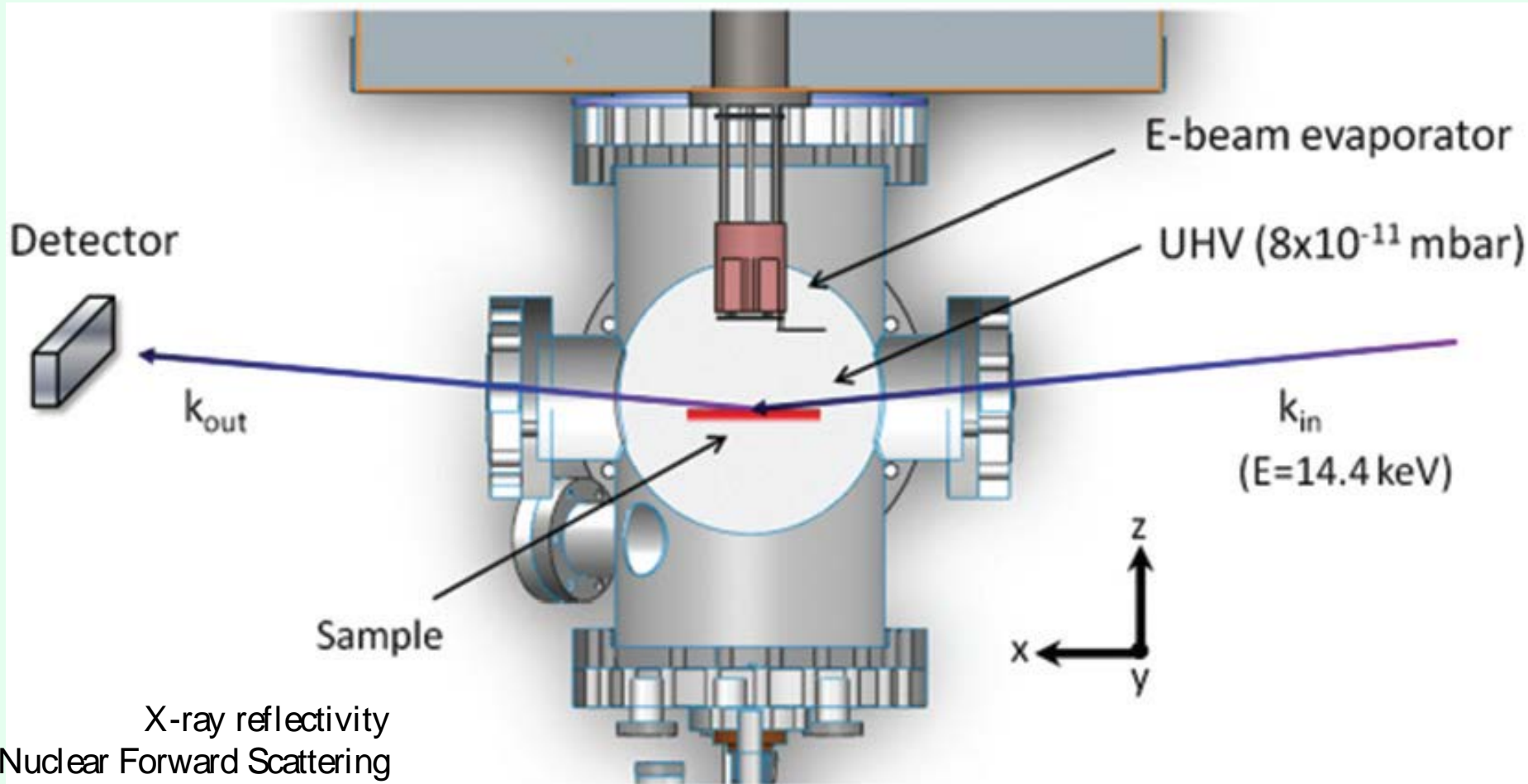
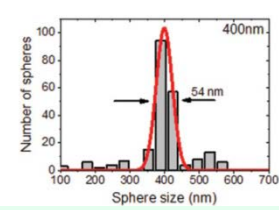
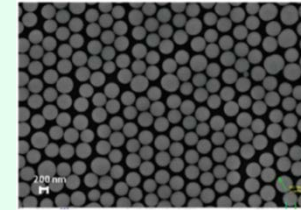
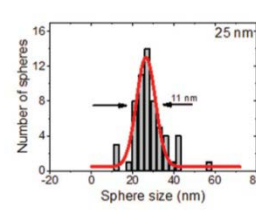
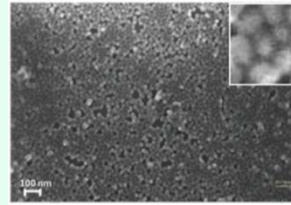
Domain ripening at $H_c = 4.2$, $D = 5.5$



What can we see from domain ripening?

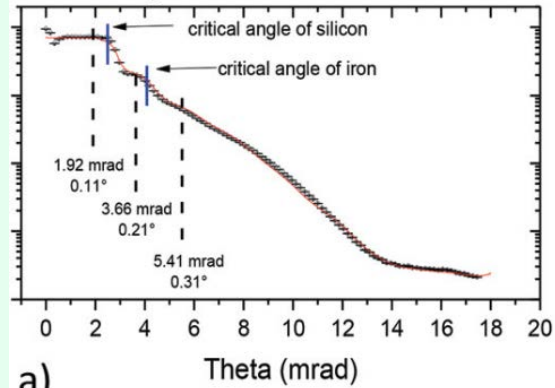
- To have a contrast between hyperfine fields of **opposite direction**, photons of **circular polarization** are needed → insert a **$\lambda/4$ phase-retarder** plate!
- AF-coupled domains still have **both directions** of the hyperfine field → the sample should be prepared by marking **every second layer** with the resonant isotope ^{57}Fe .
- To have a **k-parallel** (i.e. *x*-parallel) **component** of layer magnetization, the sample should be **tilted** (e.g. by 45°) around the *z*-axis.

Evolution of magnetism on a curved nanosurface

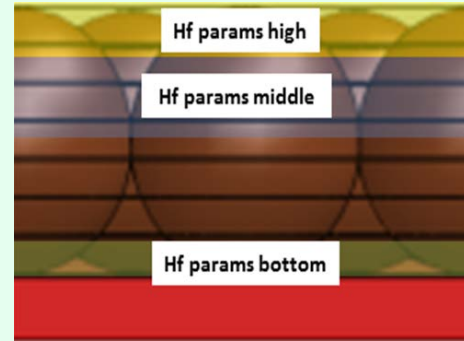


Evolution of magnetism on a curved nanosurface

Reflectivity of 35 Å Fe on flat substrate



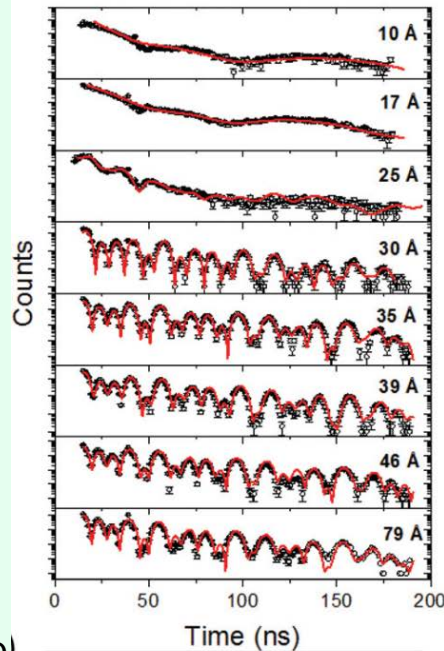
a)



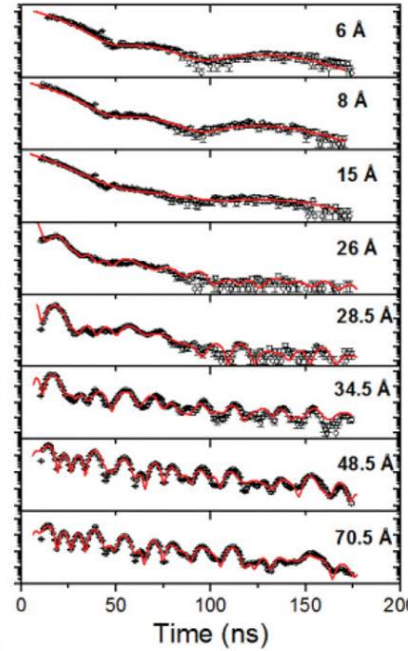
Flat

25 nm

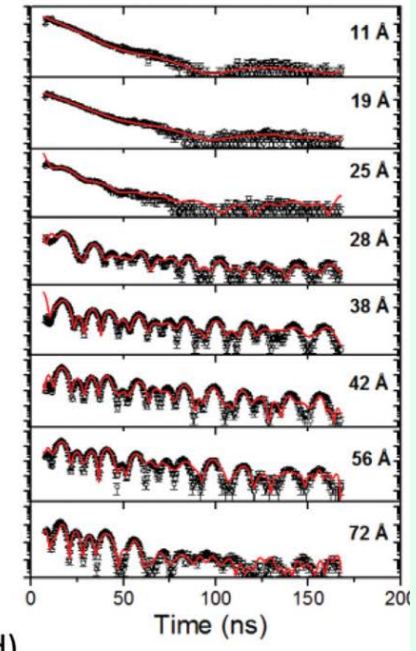
400 nm



b)

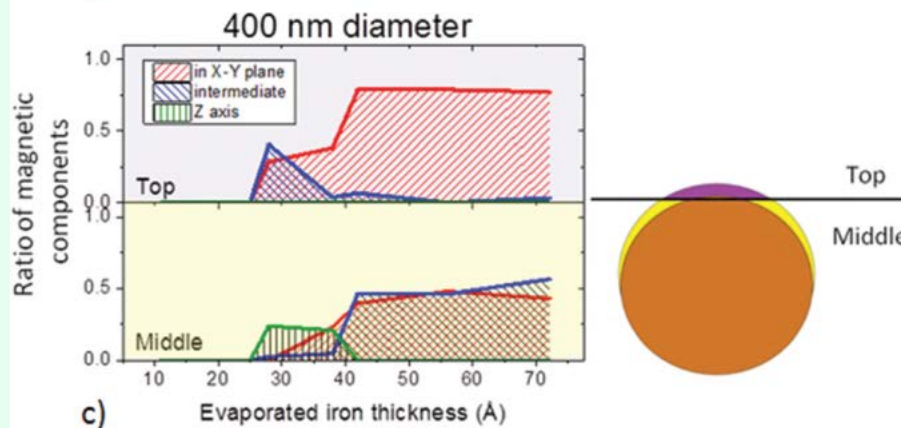
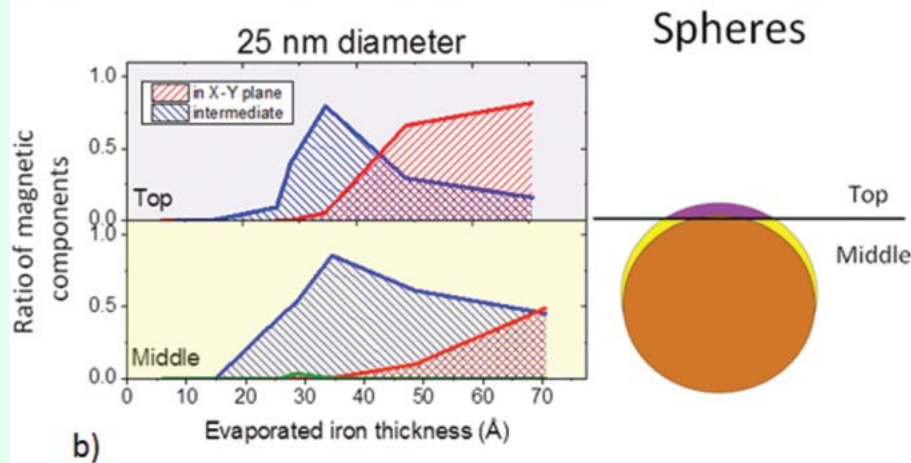
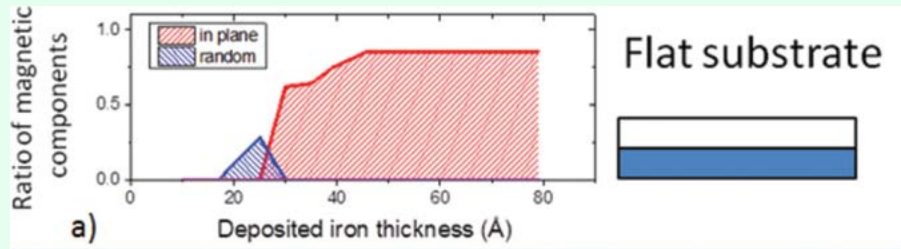


c)



d)

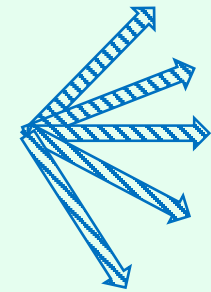
Evolution of magnetism on a curved nanosurface



in-plane



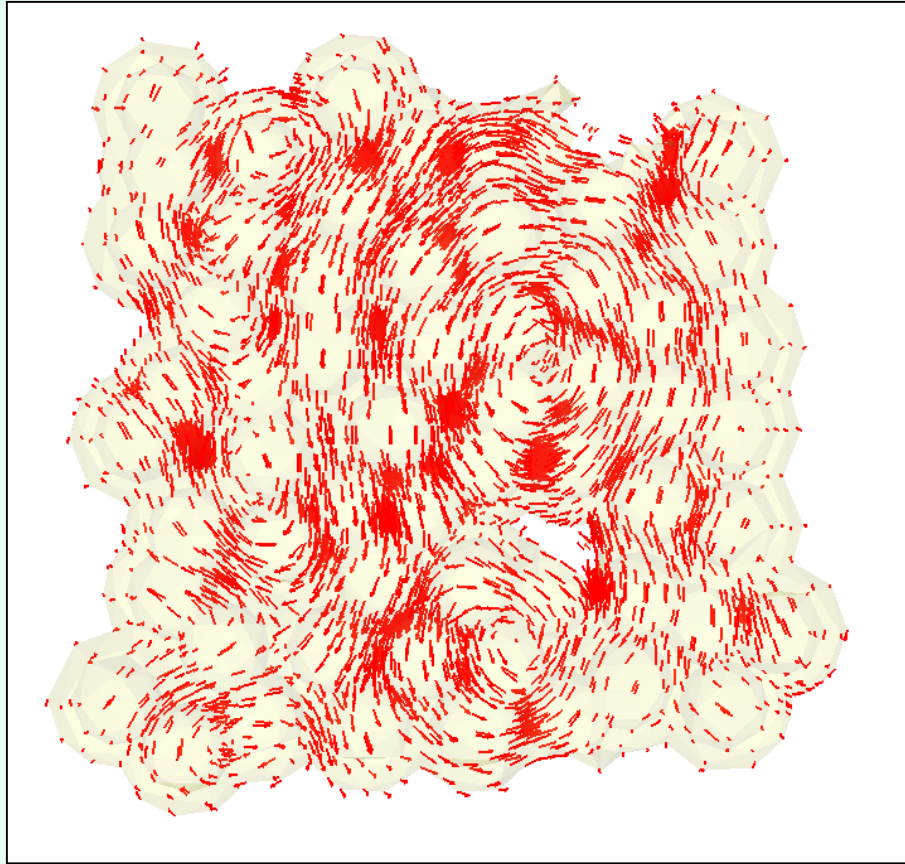
random



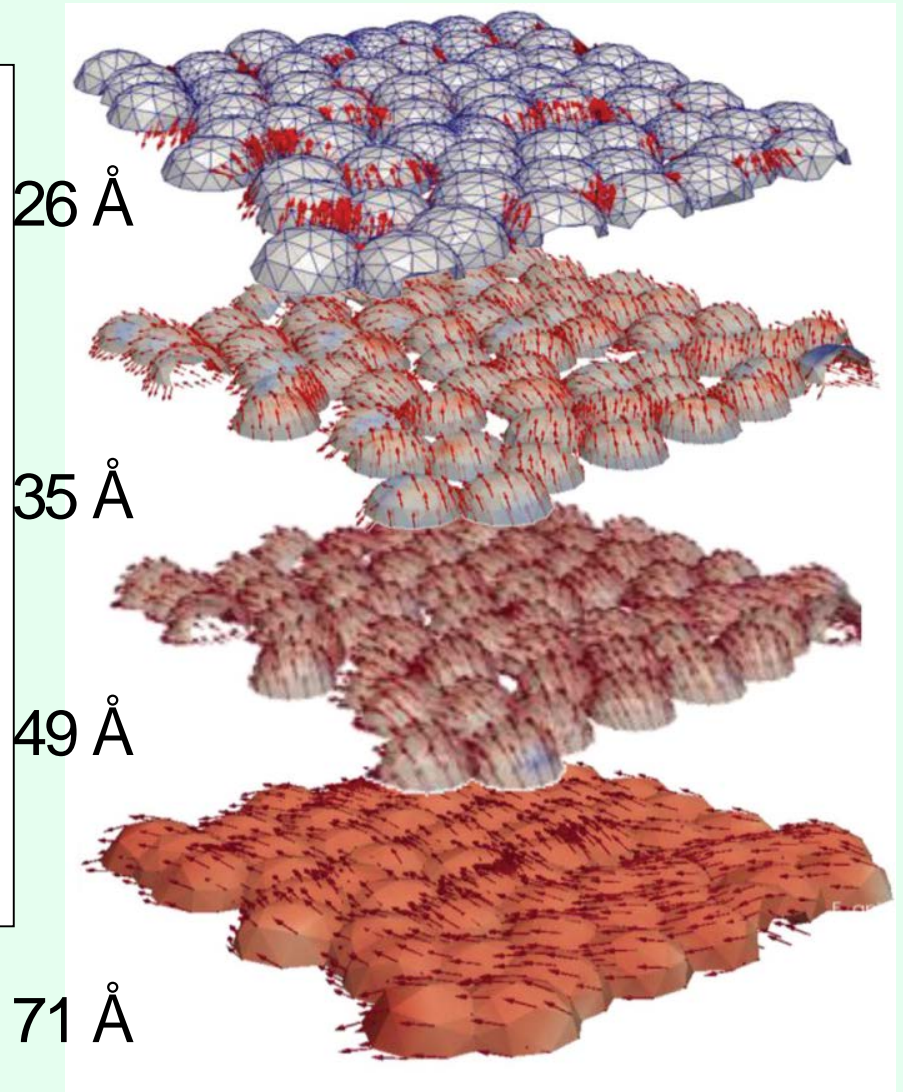
perpendicular



Evolution of magnetism on a curved nanosurface



Micromagnetic simulation



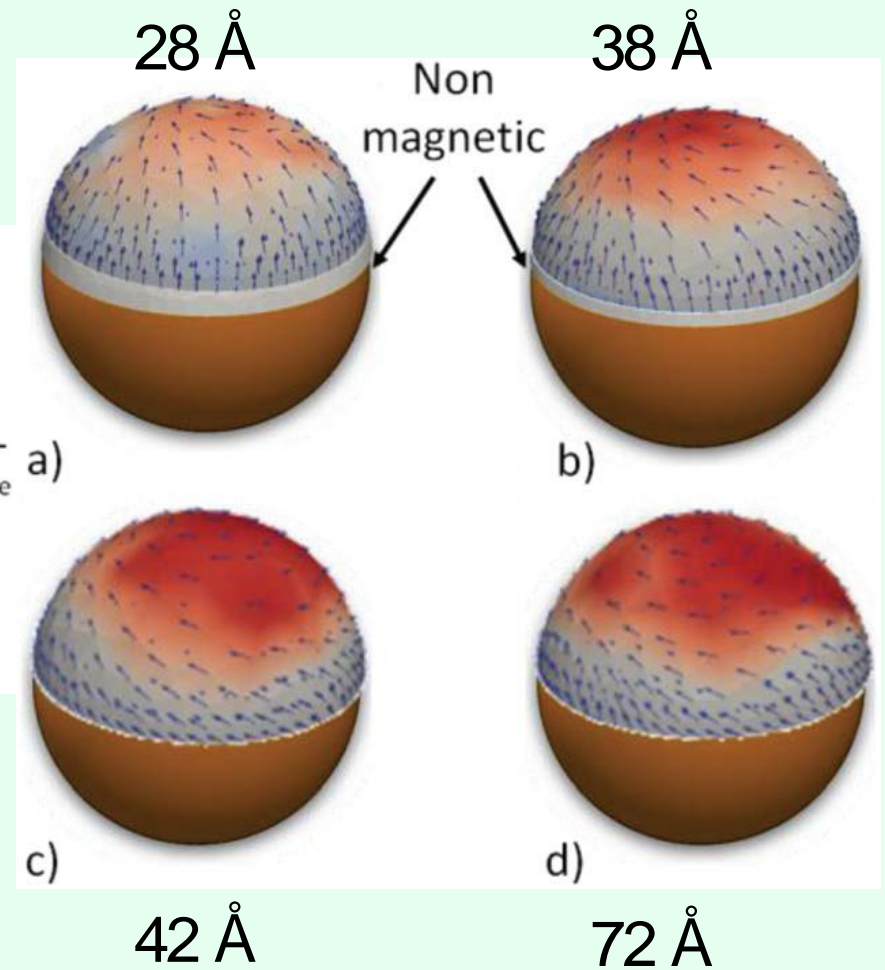
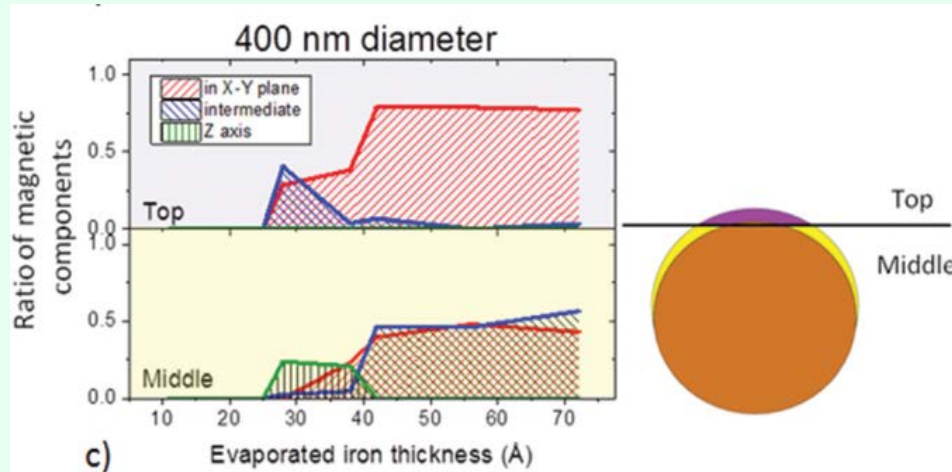
26 Å

35 Å

49 Å

71 Å

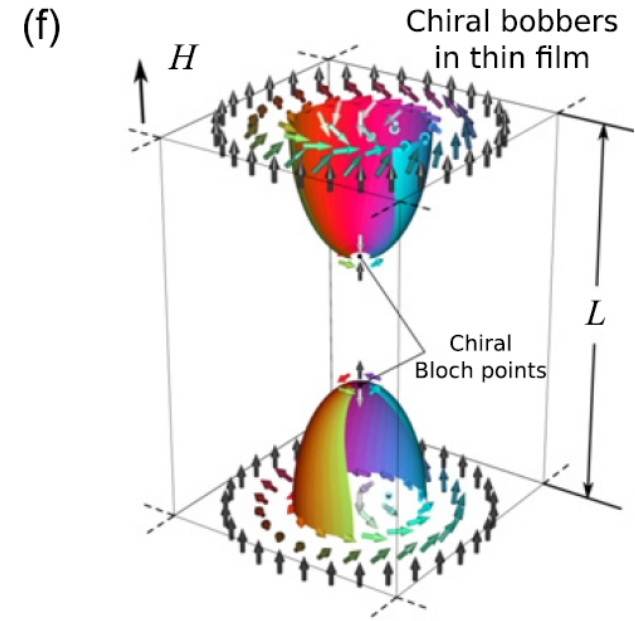
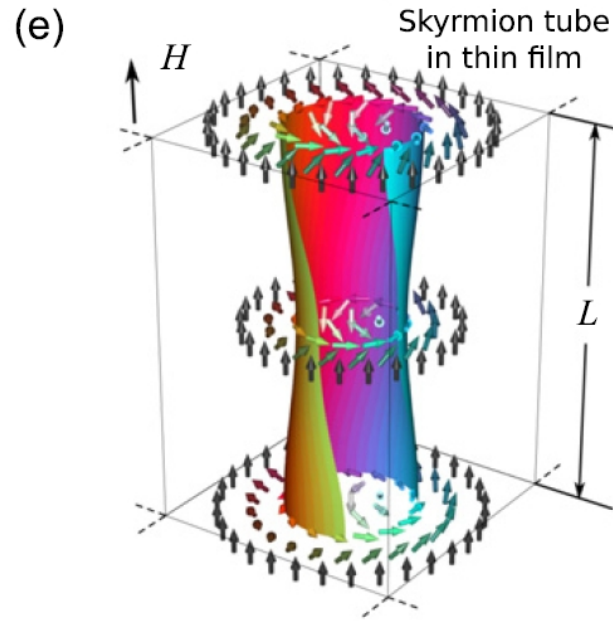
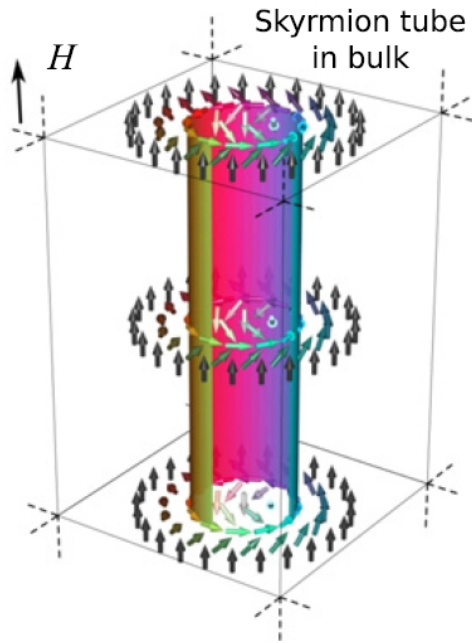
Evolution of magnetism on a curved nanosurface



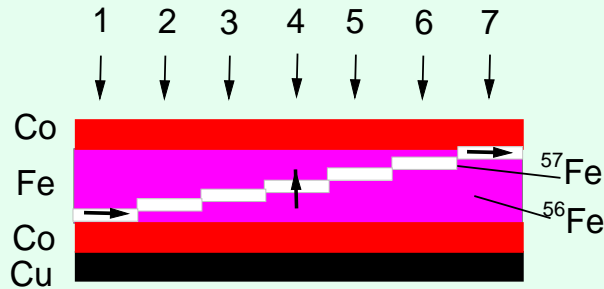
Si substrate/nanosphere: OK

5–50 nm ^{57}Fe : may be OK

Skyrmions in thin films?

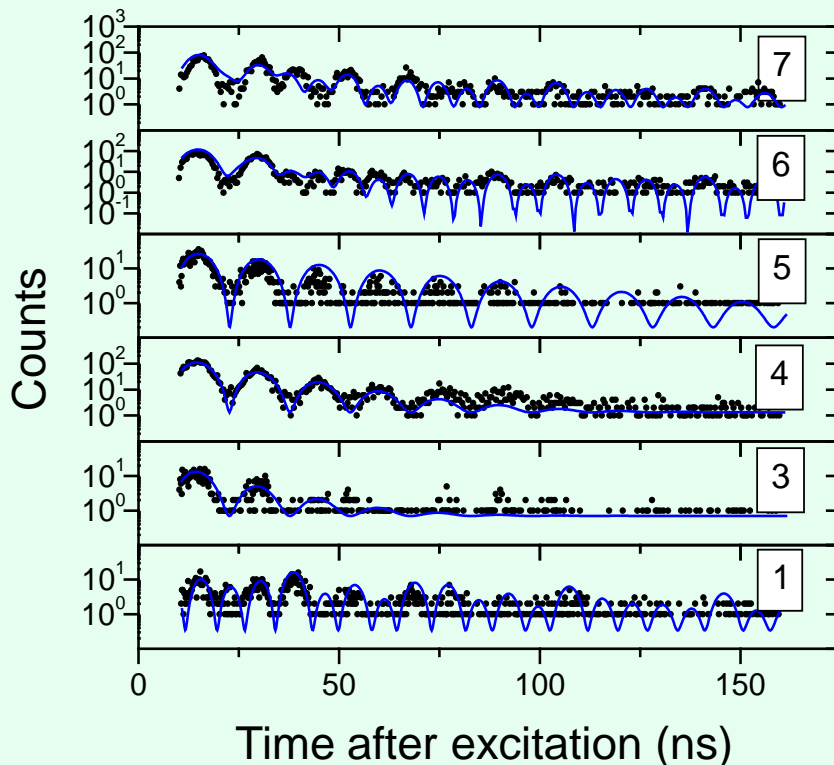


Improved lateral resolution using the non-focussed 50 μm beam at grazing incidence



Monolayer resolution can be achieved by using the resonant isotope marker technique.

In a Co/Fe(7ML)/Co trilayer, the magnetisation of the Fe layers at the Co/Fe interface is parallel while that of the internal Fe layers is perpendicular to the plane.



Acknowledgements...



Thank you for your attention!

