



Visualizing Dynamic Magnetism in Nanostructures using Electron Microscopy

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Outline

- Motivation
- Transmission Electron Microscopy
- Magnetic Minerals
 - Electron holography
- Phase transition in FeRh thin films
 - Differential phase contrast imaging (DPC)







NRS Workshop - Grenoble

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Motivation

The demand for improved functionality and reduction in size of a range modern devices has led to the rapid development of new magnetic materials

\rightarrow driving the need to visualise localised magnetism on the nanoscale

Magnetic data storage

Nanomedicine

Earth science









Transmission electron microscopy



Sample thickness < 200 nm

Resolution ~ 80 pm

Information:
➢ Morphology
➢ Chemistry
➢ Structure
➢ Magnetism
➢ Electronic
➢ Etc.

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Magnetism in Earth science



between reversals

during a reversal

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To interpret palaeomagnetic data we need to understand mechanisms that:

- 1. induce magnetic remanence
- 2. alter magnetic remanence
- 1. Chemical alteration change in oxidation state



after Bulter (1982)





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Electron holography



- FEI Titan Analytical
- C_s correction on condenser lens, *i.e.* probe corrected
- Operated at 300kV
- HR-STEM and chemical mapping at atomic scale
- Biprism and Lorentz lens for electron holography of magnetic fields

DTU Cen Center for Electron Nanoscopy



Electron holography



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Electron holography

Phase shift:

$$\varphi(X) = C_E \int V(x,z) \, dz -$$

Mean inner potential

$$-\left(\frac{e}{h}\right)\iint B_{\perp}(x,z)\,dx\,dz$$

Magnetic induction

Total phase shift

Mean inner potential





Magnetic contribution



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Electron holography





Imperial College London

Oxidation of magnetic minerals



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Oxidation of magnetic minerals

Imperial College London

Oxidation of Fe_3O_4 : $2Fe_3O_4 + \frac{1}{2}O_2 \rightarrow \Upsilon - 3Fe_2O_3$



Slightly weaker ferrimagnetic



Oxidation of magnetic minerals

Oxidation of Fe_3O_4 : $2Fe_3O_4 + \frac{1}{2}O_2 \rightarrow \Upsilon - 3Fe_2O_3 \rightarrow \alpha - 3Fe_2O_3$

Canted antiferromagnetic

M



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Environmental TEM



- FEI Titan E-Cell
- C_s correction on objective lens, *i.e.* image corrected
- Operated at 300kV
- Various gases, *i.e.* H₂, He and H₂0 up to 1000 Pa
- Heating specimen holder
 < 1000°C



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Energy dispersion of 0.02eV & resolution of 0.5eV



EEL spectrum from Fe₃O₄ particles

EEL spectrum from Υ -Fe₂O₃ sample

EEL spectrum from reference α -Fe₂O₃ sample

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Heated at 700°C, under 9mbar O₂ for 8 hours



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Imperial College London

Almeida, T. P et al., Nature Communications, 2014

Heated at 700°C, under 9mbar O₂ for 8 hours



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Oxidised

Before oxidation



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Almeida, T. P et al., Nature Communications, 2014

Environmental TEM



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Almeida, T. P et al., Nature Communications, 2014

Thermomagnetic behaviour



- FEI Titan HOLO
- C_s correction on objective lens, *i.e.* image corrected
- Operated at 60 300kV
- 3 biprisms and Lorentz lens for electron holography of magnetic fields
- 11 mm pole piece gap to allow tilting to ± 75°
- Specifically designed to allow for *in situ* experiments

Ernst Ruska-Centrum für Mikroskopie und Spektroskopie mit Elektronen



Thermomagnetic behaviour





Ultra stable holder during heating up to < 1000°C

TRM = total phase shift – MIP (measured at each temperature in separate experiment)

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Thermomagnetic behaviour

Fe₃O₄ grain at remanence upon heating to 550°C and cooling to room temperature



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Almeida, T. P et al., Science Advances, 2016

Thermomagnetic behaviour

Fe₃O₄ grain at remanence upon heating to 600°C and cooling to room temperature



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Almeida, T. P et al. Geophysical Research Letters, 2016 NRS Workshop - Grenoble

Liquid cell electron holography

Combining electron holography with liquid-cell TEM holder



Trevor.almeida@glasgow.ac.uk Prozorov T. et al. Journal of Royal Society: Interface, 2017 NRS Workshop - Grenoble

Combining electron holography with liquid-cell TEM holder





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Background / motivation

Equiatomic intermetallic iron-rhodium compound (Fe₄₈Rh₅₂ to Fe₅₆Rh₄₄)



Rh: ~1 μ_Β

Zakharov A I *et al.*, *J. Exp. Theor. Phys.*, 1964 Kouvel J S *et al.*, *J. Appl. Phys.* 1962



Sample preparation

DC magnetron sputter co-deposition

- FeRh targets
- MgO and GaAs substrates
- Samples:
 - 1) FeRh on MgO
 - 2) Planar FeRh TEM samples via HF-etching

Focused Ion Beam / Scanning Electron Microscope (FIB-SEM)

• Cross-sectional and planar FeRh samples







Lightning holder with heater / biasing MEMS chip



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Sample characterisation



- JEOL ARM200cF "MagTEM"
- C_s correction on condenser lens, *i.e.* probe corrected
- Operated at 60kV 200kV
- HR-STEM and chemical mapping at atomic scale, with EDX and EELS
- Lorentz lens and segmented / pixelated detectors for imaging of magnetisation



Kelvin Nanocharacterisation Centre



Phase transition in FeRh

HF-etched FeRh planar sample as a function of temperature **20°C** ▶ 140°C ▶ 20°C **Bright field TEM imaging** Low-angle electron diffraction 2 um 20 µrad

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Differential phase contrast imaging



•STEM mode focused probe on sample, probe semi angle a.

•Beam deflected by Lorentz force.

•Segmented detector can then be used to map deflection by taking difference signals from opposite segments (quadrants or halves)



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In situ imaging of FeRh transition

HF-etched FeRh planar sample as a function of temperature



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In situ imaging of FeRh transition

Medipix pixelated detector can remove effects of diffraction contrast











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Krajnak, M. et al. Ultramicroscopy, 2016



In situ imaging of FeRh transition

HF-etched FeRh examined using the **Medipix** detector at 80°C





Cross-section of 55 nm FeRh on MgO substrate

20°C → 200°C



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Planar TEM lamella of FeRh on NiAl on GaAs substrate at 150°C











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800 mV







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900 mV







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1 V





1 µm

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1.1 V





Bottom contact

n films

Summary

- Specialised TEM techniques can be used to visualise nanoscale magnetism and provide fundamental insight for a range of applications
- Combining electron holography with in situ TEM and ETEM provided direct access to:
 - > The effect of oxidation on vortex-state Fe_3O_4 grains
 - Thermomagnetic behaviour of vortex states
- Lorentz techniques like Fresnel, SAES, conventional and pixelated DPC imaging allow for:
 - Visualising the dynamic nucleation and agglomeration ferromagnetic domains during transition in FeRh
 - DW and phase boundary motion can be controlled by current pulsing and Ir / Pd doping of the FeRh films











Acknowledgements

Special Thanks to:

Adrian Muxworthy

Imperial College London

Wyn Williams Lesleis Nagy

Stephen McVitie Damien McGrouther Sam McFadzean Gary Paterson







Acknowledgements

Special Thanks to:

Takeshi Kasama Thomas Hansen Jakob Wagner Jens Kling Christian Damsgaard



Technical University of Denmark

Rafal Dunin-Borkowski András Kovács Chris Boothroyd Vadim Migunov

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Thank you

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