



ID11 Materials Science: Diffraction and Imaging

Jonathan Wright

Haixing Fang

Wolfgang Ludwig

Pierre-Olivier Aufran

James Ball

Pedro Damas-Resende

Zheheng Liu

Henri Gleyzolle

Eric Gagliardini

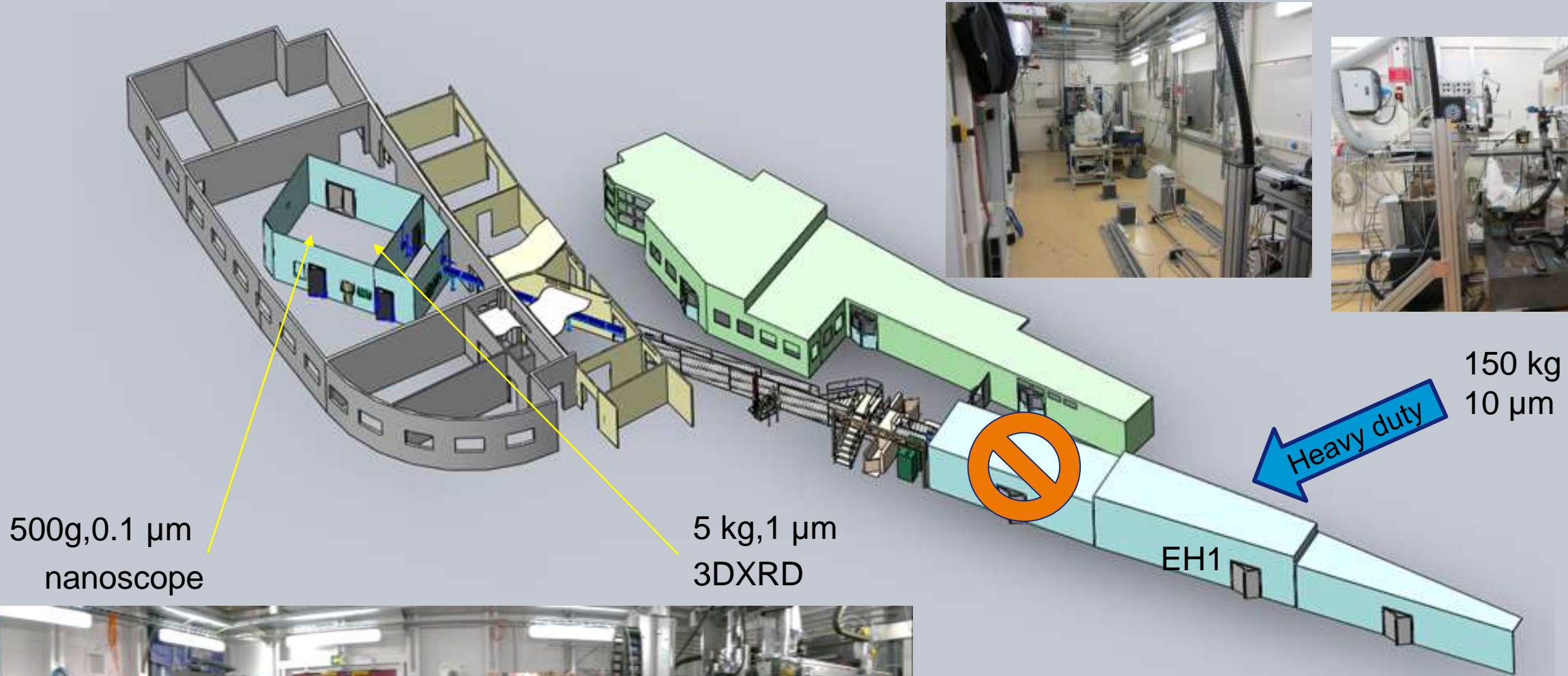
Emmanuel Papillon

Jose Maria Clement



Photo credit: S. Cande/ESRF

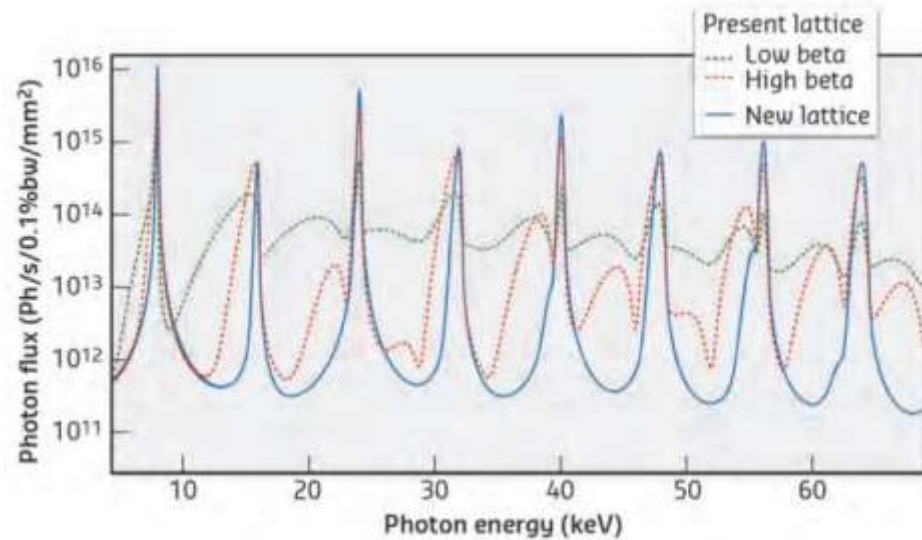
ID11 Materials Science beamline : upgraded 2006, nanofocus station since 2016



ESRF Upgrade Phase II : new source in 2019-2020 + new detectors

~40X more photons
for ID11 optics

2 in-vacuum undulators
High power density at front end



Eiger2 4M CdTe, 500 fps



2x Marana sCMOS (24 fps @16bit)



New Optique Peter Imaging Microscope



Beam heating from a fourth-generation synchrotron source

J. Synchrotron Rad. (2021). **28**, 1377–1385

Eleanor Lawrence Bright,* Carlotta Giacobbe and Jonathan P. Wright

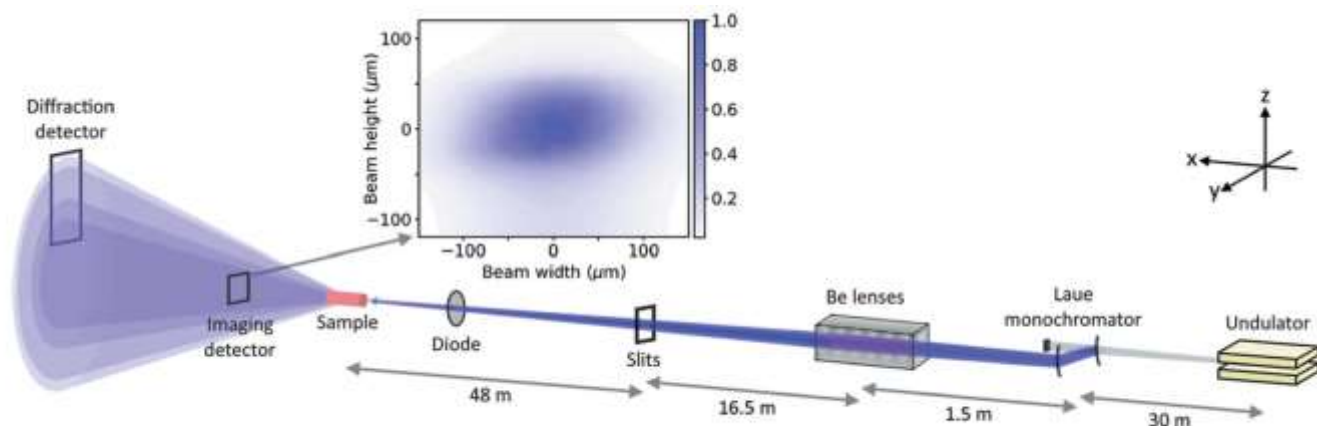


Figure 1
Schematic of the experimental set-up at the ID11 beamline, with a plot showing an image of the focused beam.

<https://doi.org/10.1107/S160057752100669X>

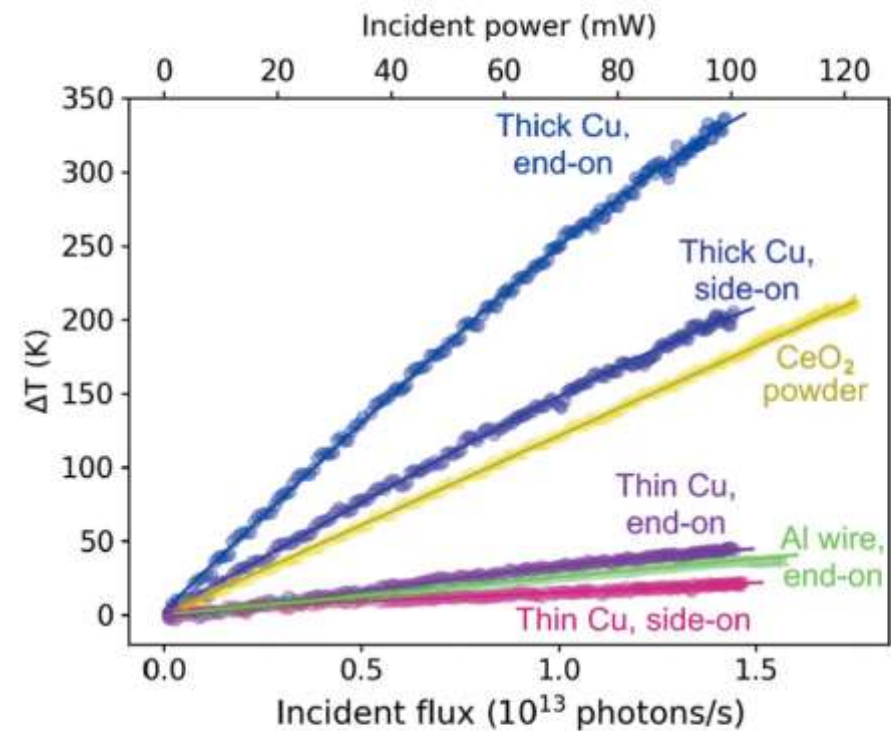


Figure 2
Change in temperature of samples as a function of incident flux, with 0.19 mm-thick and 0.025 mm-thin Cu wire aligned end-on or side-on to the beam, a 0.5 mm capillary of CeO₂, and 0.5 mm Al wire.

Recrystallisation of copper due to beam heating. Lawrence-Bright *et-al*

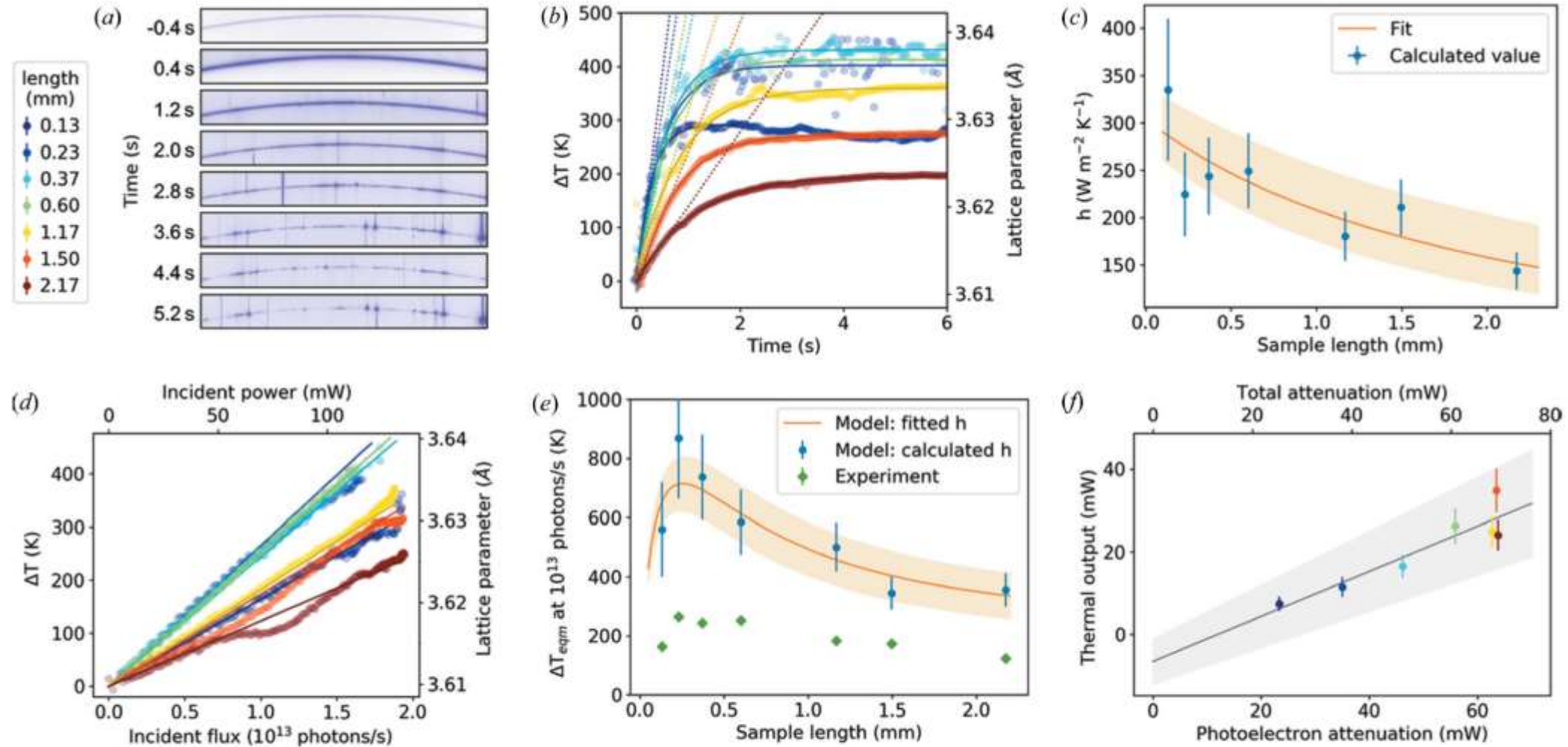


Figure 5

Results for 0.19 mm-diameter Cu wire samples of varying lengths (see legend), with (a) images of the (222) reflection as a function of time after insertion of focusing lenses into the beam from a 0.86 mm-length sample, (b) the change in temperature as a function of time after insertion of focusing lenses, (c) calculated h values as a function of length, (d) change in temperature as a function of incident flux, (e) measured and modelled change in temperature at 10^{13} photons s^{-1} , and (f) calculated thermal output at 10^{13} photons s^{-1} against X-ray beam attenuation.

Diffraction Experiments : *adapted to the sample*

40-70 keV
>0.1 μm point

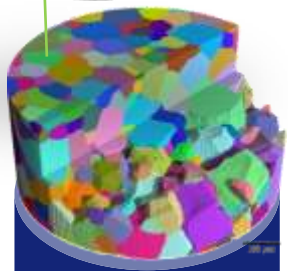
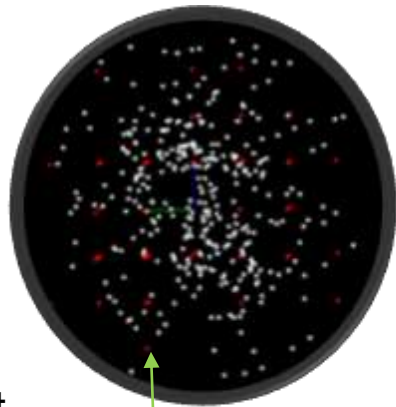
2D slice

>0.1 x 50 μm slice

3D beam
~1.8 mm

>1x1 μm 3DXRD
>1x<1800 μm
<1.4x<1.8 mm

>10x>30 μm EH1
<0.7x<0.7 mm



Sample Rotates

Single Crystal Methods

Atomic structures
Imaging
Microstructures

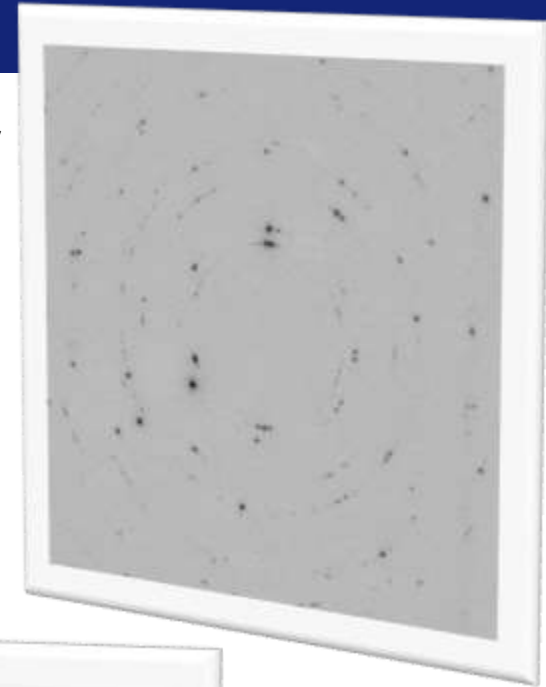
"spotty"

Multi-phase +
Composites

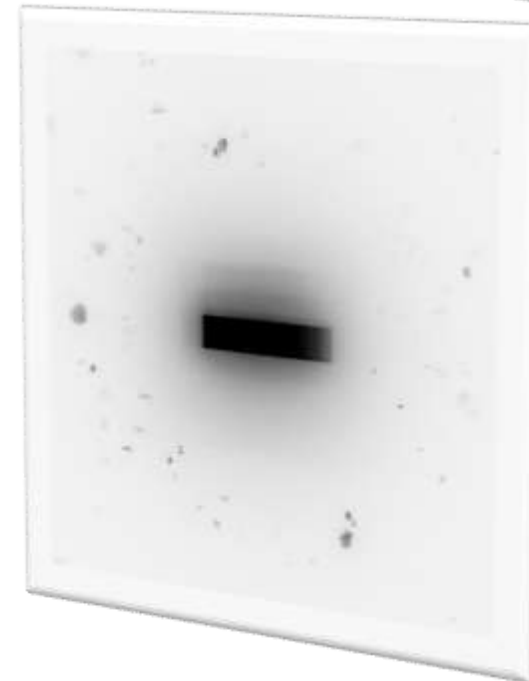
Phase transitions

Powder Methods

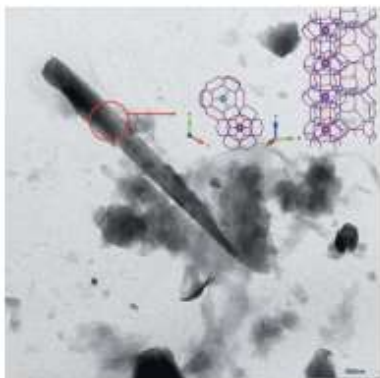
2D detector
Far field
Eiger4M
(500 fps)



2D detector
Near field
PCT + DCT
(24 fps)



Single crystal diffraction, nanoscope station (2023 pubs)



The crystal structure of the killer fibre erionite from Tuzköy (Cappadocia, Turkey)

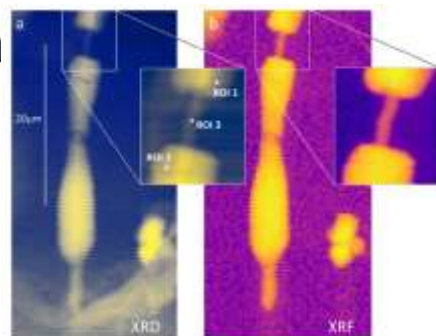
Carlotta Giacobbe,^{a*} Anna Moliterni,^{b*} Dario Di Giuseppe,^c Daniele Malferrari,^c Jonathan P. Wright,^a Michele Mattioli,^d Simona Raneri,^e Cinzia Giannini,^b Laura Fornasini,^e Enrico Mugnaioli,^f Paolo Ballirano^g and Alessandro F. Gualtieri^c

IUCrJ (2023). 10, 397–410

Closing the knowledge gap on the composition of the asbestos bodies

F. Bardelli · C. Giacobbe · P. Ballirano · V. Borelli ·
F. Di Benedetto · G. Montegrossi · D. Bellis ·
A. Pacella

Environ Geochem Health (2023) 45:5039–5051
<https://doi.org/10.1007/s10653-023-01557-0>



Chemistry
A European Journal

Chemistry
Europe
European Chemical
Societies Publishing

Research Article | Open Access | CC BY NC ND

Modular Principle for Complex Disordered Tetrahedral Frameworks in Quenched High-Pressure Phases of Phosphorus Oxide Nitrides

Daniel Günther, Dr. Dominik Baumann, Prof. Dr. Wolfgang Schnick ✉ Prof. Dr. Oliver Oeckler ✉

What happens when you're under pressure: The tetrahedral frameworks of $P_{12}O_{21}N_{14}$ and $P_{12}O_{21}N_{16}$ correspond to quenched high-pressure phases and can be described as different arrangements of the same modular building units. The crystal structures were determined from fragments intergrown with the structurally related compound $H_2P_2O_7$.

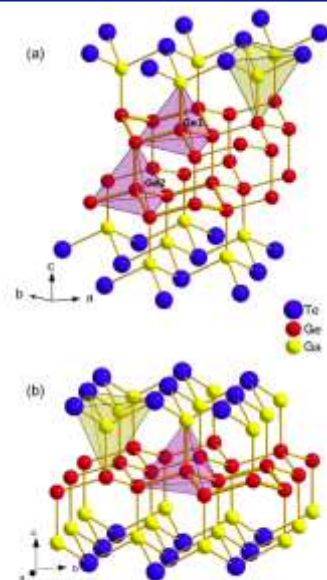


Figure 1. One layer of the crystal structures of $GaGe_2Te$ (a) and $GaGeTe$ (b); the coordination polyhedra around the Ge and Ga atoms as well as some chair-like six-membered rings are highlighted.

DOI: 10.1002/zaac.202300107

Layered $GaGe_2Te$: structure and chemical bonding

Tobias Juhlke,^[a] Simon Steinberg,^[b] Lennart Staab,^[a] Eleanor Lawrence Bright,^[c] and Oliver Oeckler^{*[a]}

Chemistry
A European Journal

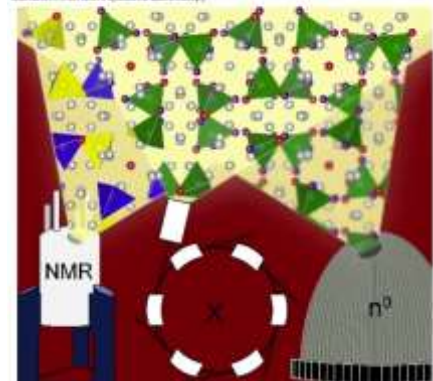
Chemistry
Europe
European Chemical
Societies Publishing

Research Article | Open Access | CC BY NC ND

Comprehensive Investigation of Anion Species in Crystalline Li^+ Ion Conductor $Li_{27-x}[P_4O_{7+x}N_{9-x}]O_3$ ($x \approx 1.9(3)$)

Stefanie Schneider, Dr. Eva-Maria Wendinger, Dr. Volodymyr Baran, Dr. Anna-Katharina Hätz, Prof. Dr. Bettina V. Lotsch, Dr. Markus Nentwig, Prof. Dr. Oliver Oeckler, Dr. Thomas Bräuniger, Prof. Dr. Wolfgang Schnick ✉ ... See fewer authors ↗

The structure of the crystalline lithium neopentylphosphate $Li_{27-x}(P_4O_{7+x}N_{9-x})O_3$ contains three different anions, namely PO_4^{3-} , PCO_3^{2-} , and PO_3^- . It was comprehensively investigated by single-crystal and powder X-ray diffraction, neutron powder diffraction, NMR spectroscopy and elemental analysis. A total ionic conductivity of $6.6 \times 10^{-4} \text{ S cm}^{-1}$, which is similar to that of amorphous a-LIPOH was determined with impedance spectroscopy.



VIP Polynitrides Very Important Paper

How to cite: *Angew. Chem. Int. Ed.* **2022**, *61*, e202207469
 International Edition: doi.org/10.1002/anie.202207469
 German Edition: doi.org/10.1002/ange.202207469

Anionic N₁₈ Macrocycles and a Polynitrogen Double Helix in Novel Yttrium Polynitrides YN₆ and Y₂N₁₁ at 100 GPa

Andrey Aslandukov,* Florian Trybel, Alena Aslandukova, Dominique Laniel, Timofey Fedotenko, Saiana Khandarkhaeva, Georgios Aprilis, Carlotta Giacobbe, Eleanor Lawrence Bright, Igor A. Abrikosov, Leonid Dubrovinsky, and Natalia Dubrovinskaia

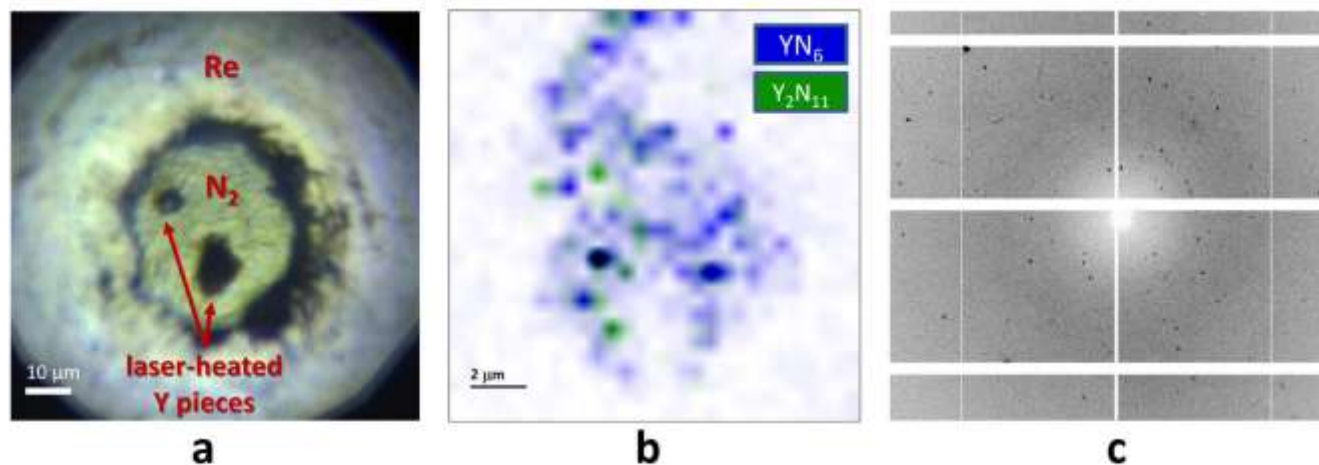
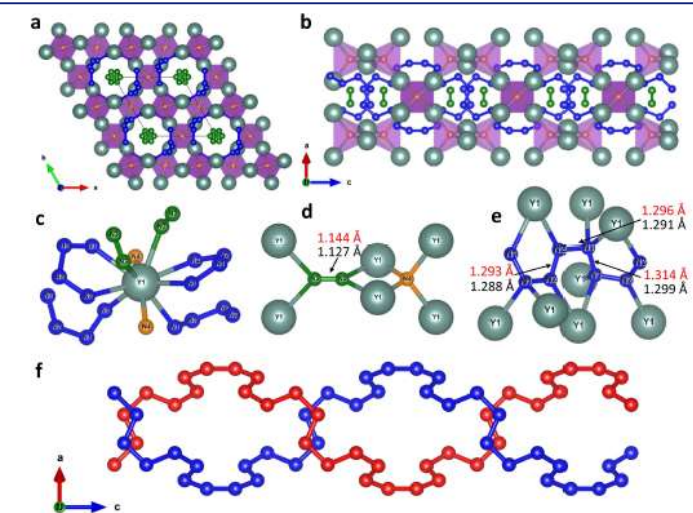
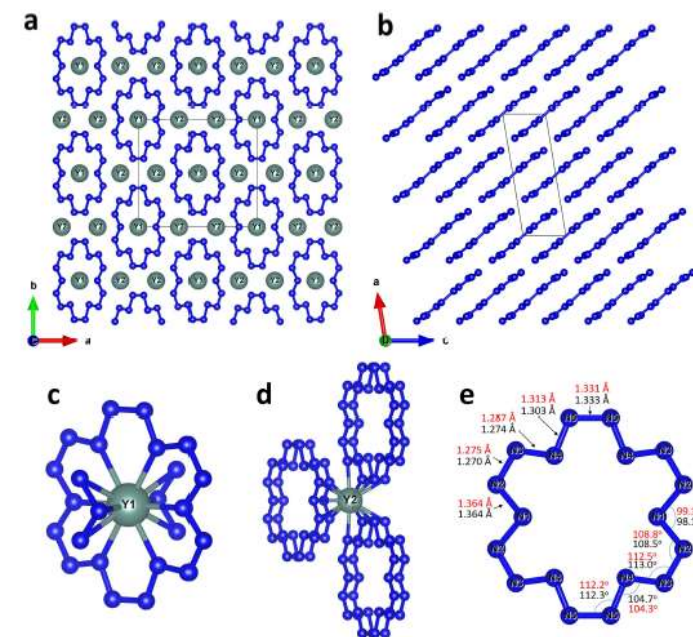


Figure 1. Experimental details. a) Microphotograph of the sample chamber. b) 2D X-ray diffraction map showing the distribution of the two yttrium nitrides phases within the heated sample. The color intensity is proportional to the intensity of the following reflections: the (2 0 0), (0 2 0), and (1 1 -2) of YN₆ for the blue regions; the (1 0 1), (2 -1 0), and (2 -1 4) of Y₂N₁₁ for the green regions. c) Example of an X-ray diffraction pattern collected from the laser-heated sample at 100 GPa.



High pressure diamond anvil crystallography (pubs 2023)

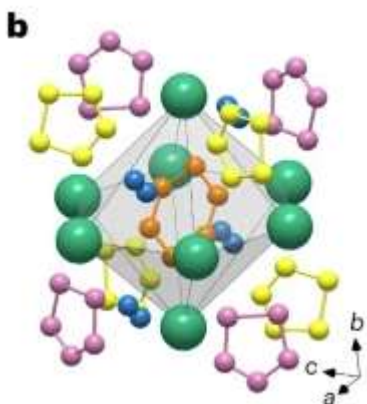
2D XRD map + select xtal + rotation scan

nature chemistry

Article

<https://doi.org/10.1038/s41557-023-01148-7>

Aromatic hexazine $[N_6]^{4-}$ anion featured in the complex structure of the high-pressure potassium nitrogen compound K_9N_{56}



Dominique Laniel^{1,2}, Florian Trybel³, Yuqing Yin^{1,4}, Timofey Fedotenko¹, Saiana Khandarkhaeva⁵, Andrey Aslandukov¹, Georgios Aprilis⁶, Alexei I. Abrikosov⁷, Talha Bin Masood⁷, Carlotta Giacobbe⁶, Eleanor Lawrence Bright⁶, Konstantin Glazyrin⁹, Michael Hanfland⁶, Jonathan Wright⁵, Ingrid Hotz⁷, Igor A. Abrikosov³, Leonid Dubrovinsky⁴ & Natalia Dubrovinskaia^{1,3}

JACS Au

pubs.acs.org/jacsau

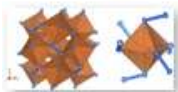


Article

Unraveling the Bonding Complexity of Polyhalogen Anions: High-Pressure Synthesis of Unpredicted Sodium Chlorides Na_2Cl_3 and Na_4Cl_5 and Bromide Na_4Br_5

Yuqing Yin,⁸ Alena Aslandukova, Nityasagar Jena, Florian Trybel, Igor A. Abrikosov, Bjoern Winkler, Saiana Khandarkhaeva, Timofey Fedotenko, Elena Bykova, Dominique Laniel, Maxim Bykov, Andrey Aslandukov, Fariia I. Akbar, Konstantin Glazyrin, Gaston Garbarino, Carlotta Giacobbe, Eleanor L. Bright, Zhitai Jia, Leonid Dubrovinsky, and Natalia Dubrovinskaia

Acta Cryst. (2023). E79, 923-925
<https://doi.org/10.1107/S2056989023008058>



Synthesis and crystal structure of silicon pernitride SiN_2 at 140 GPa

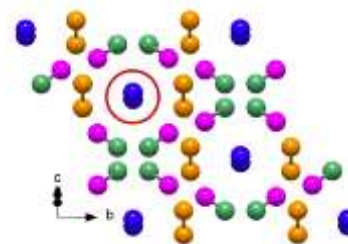
P. L. Jurzick¹, G. Krach, L. Brüning², W. Schnick³ and M. Bykov⁴

nature communications

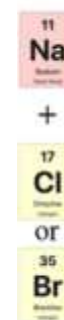
Article

<https://doi.org/10.1038/s41467-023-41968-2>

Structure determination of ζ - N_2 from single-crystal X-ray diffraction and theoretical suggestion for the formation of amorphous nitrogen



Dominique Laniel¹, Florian Trybel², Andrey Aslandukov^{3,4}, James Spender¹, Umbertoluca Ranieri¹, Timofey Fedotenko⁵, Konstantin Glazyrin⁵, Eleanor Lawrence Bright⁶, Stella Chariton⁷, Vitali B. Prakapenka⁷, Igor A. Abrikosov², Leonid Dubrovinsky⁴ & Natalia Dubrovinskaia^{2,3}



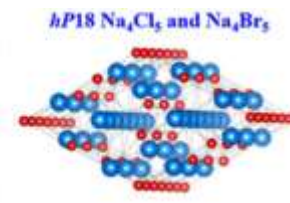
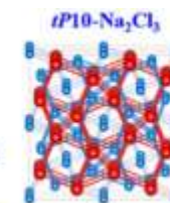
41-80 GPa
~2000 K

Unpredicted

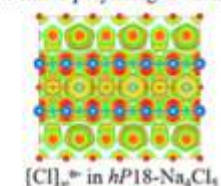
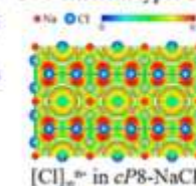
$tP10-Na_2Cl_3$

$hP18-Na_4Cl_5$

$hP18-Na_4Br_5$

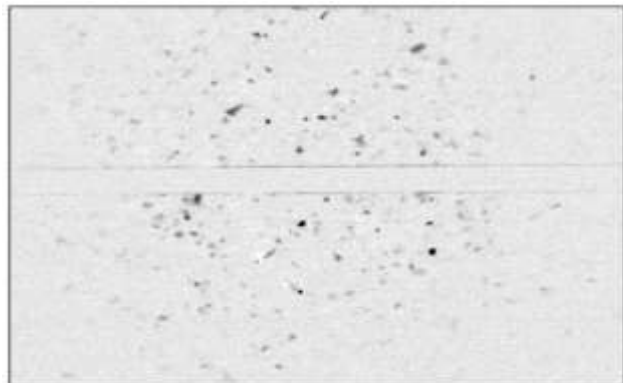


Two different types of infinite linear polyhalogen chains



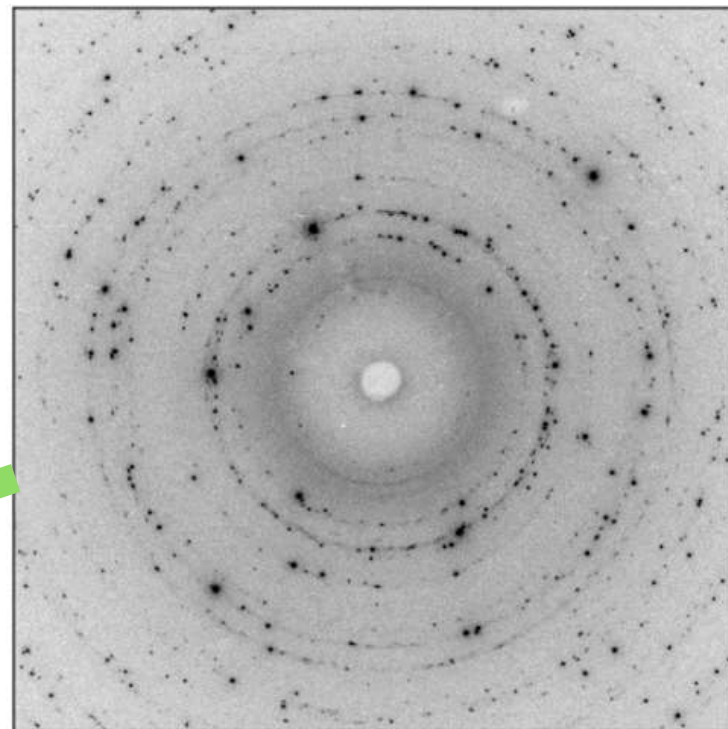
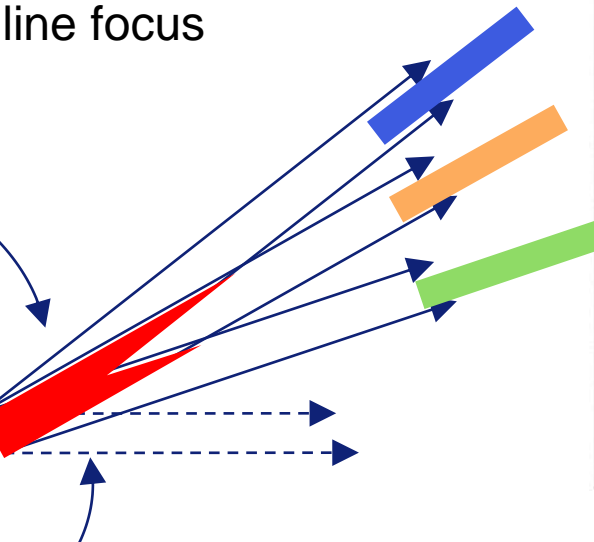
Near-field and far-field detectors : suite of 3DXRD methods

Pixels smaller than crystals



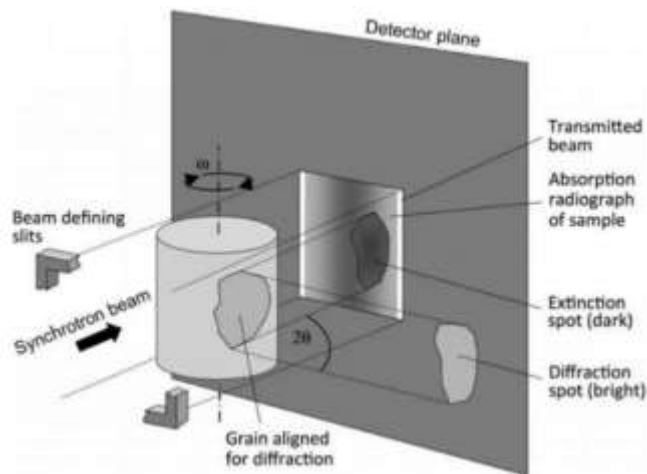
Near Field HEDM
(usually 2D, line focus
slices)

1D Point focus
2D Line focus
3D Beam



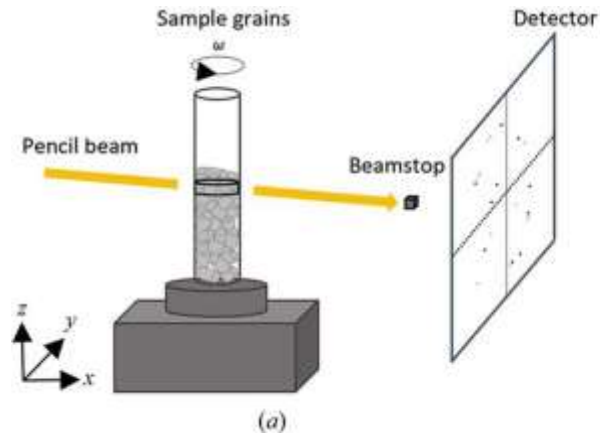
Pixels larger than crystals

- Single crystal XRD
- Far-Field HEDM
- "Box Beam"
- "point scanning"

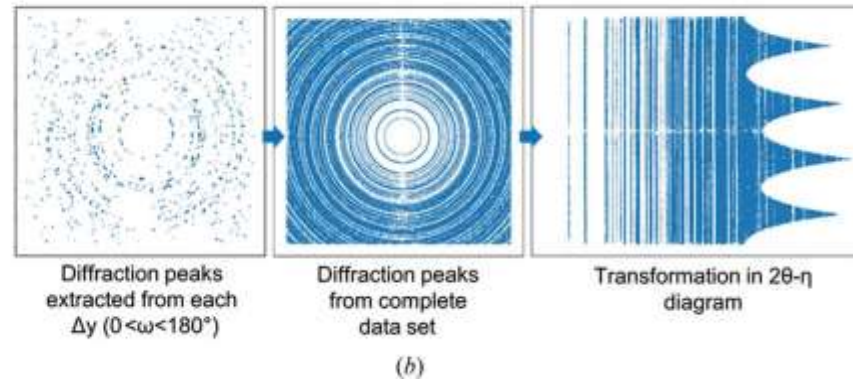


DCT =
Diffraction
Contrast
Tomography
(3D)

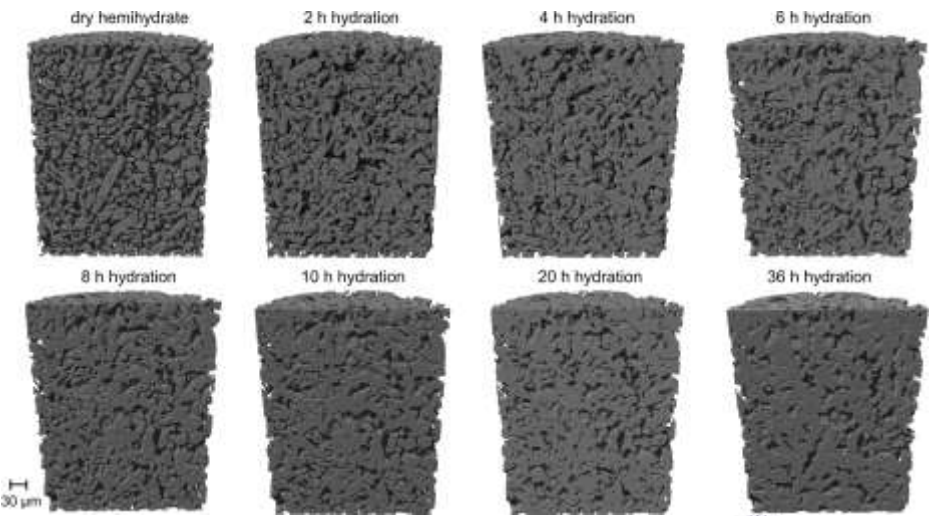
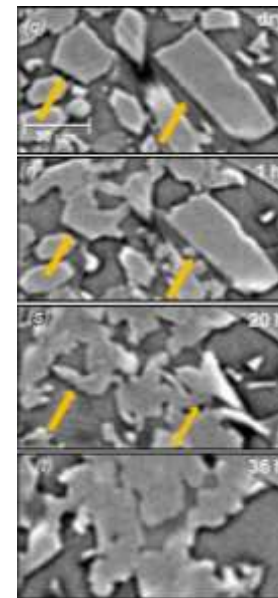
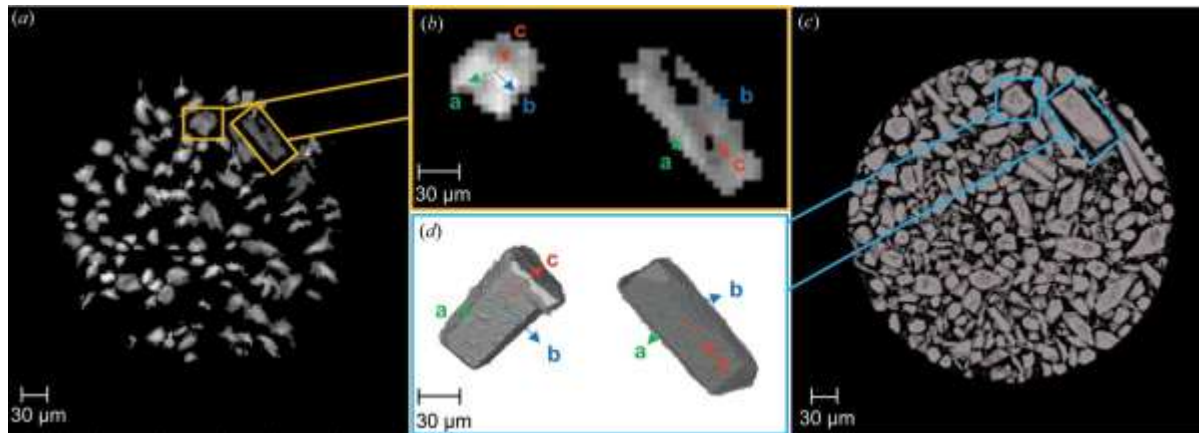
Calcium Sulfate hemihydrate–gypsum transformation



s3DXRD



PCT



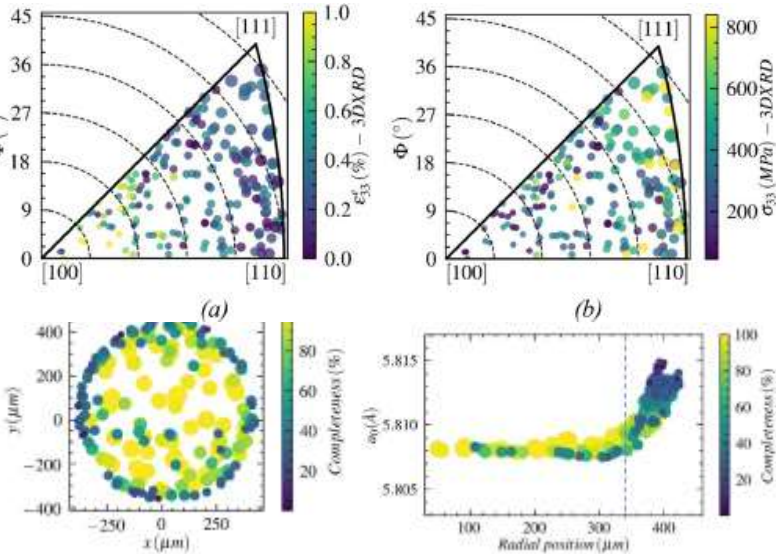
J. Appl. Cryst. (2023) **56**, 660.

Michela La Bella,^{a,b,*} Rogier Besselink,^b Jonathan P. Wright,^a
Alexander E. S. Van Driessche,^{b,c} Alejandro Fernandez-Martinez^b and
Carlotta Giacobbe^a

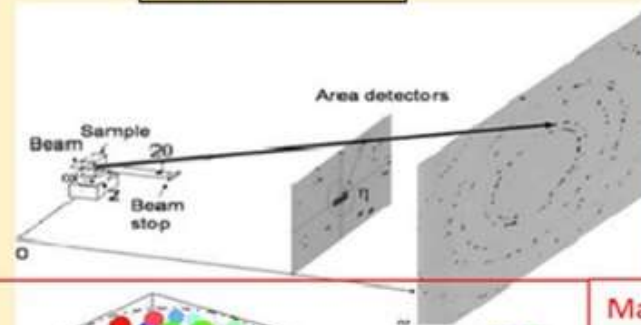
Shape Memory Alloy

Super-elastic deformation in Cu-Al-Be alloy:

- Microstructure from DCT
- Full strain tensor via 3DXRD
- FEM simulation model



Experimental

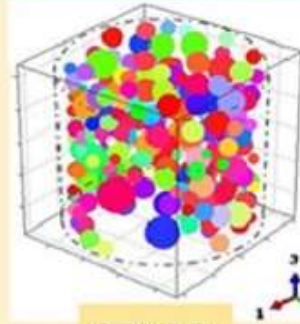


Simulation

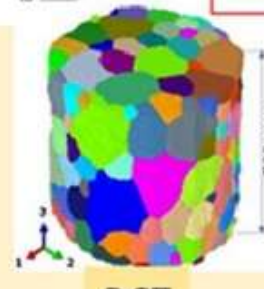
$$\Psi = \frac{1}{2} \sum_{ijkl} S_{ijkl} \Sigma_{kl} + \sum_{ij} \varepsilon_{ij}^n f^n - B(T - T_0) \sum_n f^n - \frac{1}{2} \sum_{n,m} H^{nm} f^n f^m$$

NEPER software : Laguerre tessellations
Grain boundaries idealization

Macroscopic scale



3DXRD

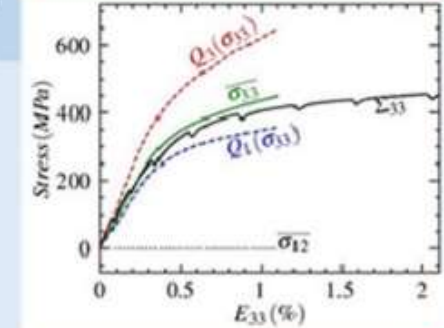


DCT

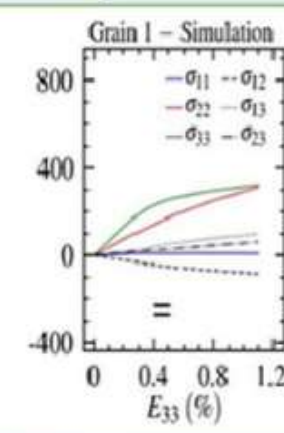
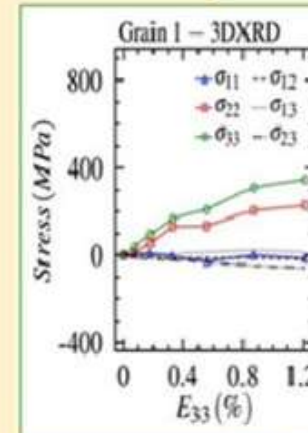


Meshing

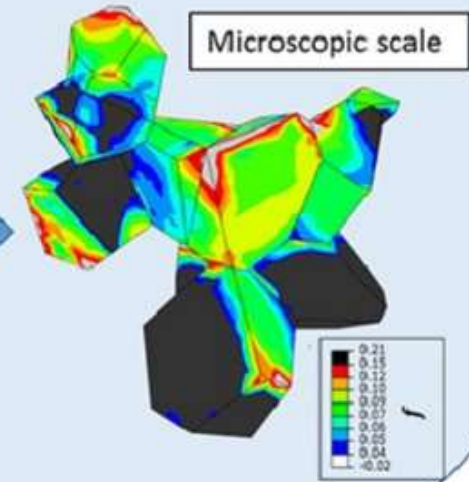
Abaqus
C3D10



Mesoscopic scale



Microscopic scale



Acta Materialia 235 (2022) 118107

Multi-scale *in situ* mechanical investigation of the superelastic behavior of a Cu-Al-Be polycrystalline shape memory alloy

Y. El Hachi^a, S. Berveiller^{a,*}, B. Piotrowski^a, J. Wright^b, W. Ludwig^c, B. Malard^d

Topo Tomography

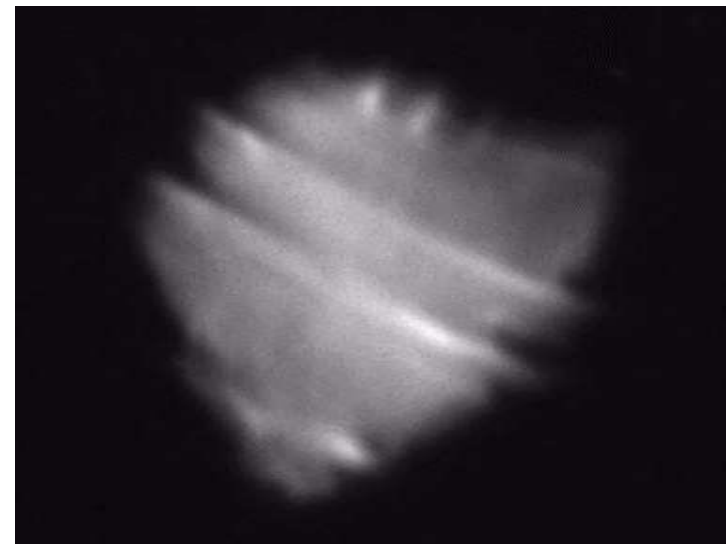
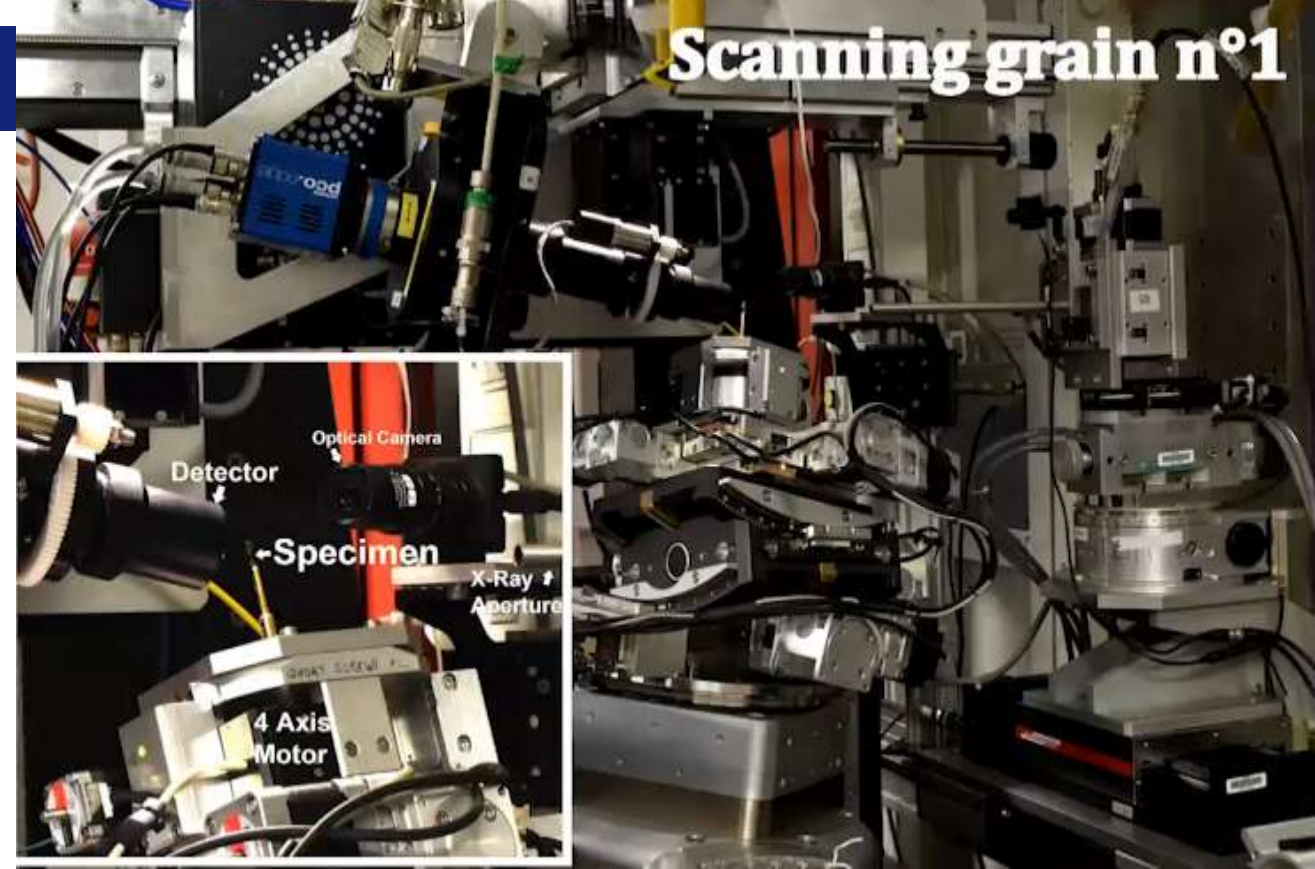
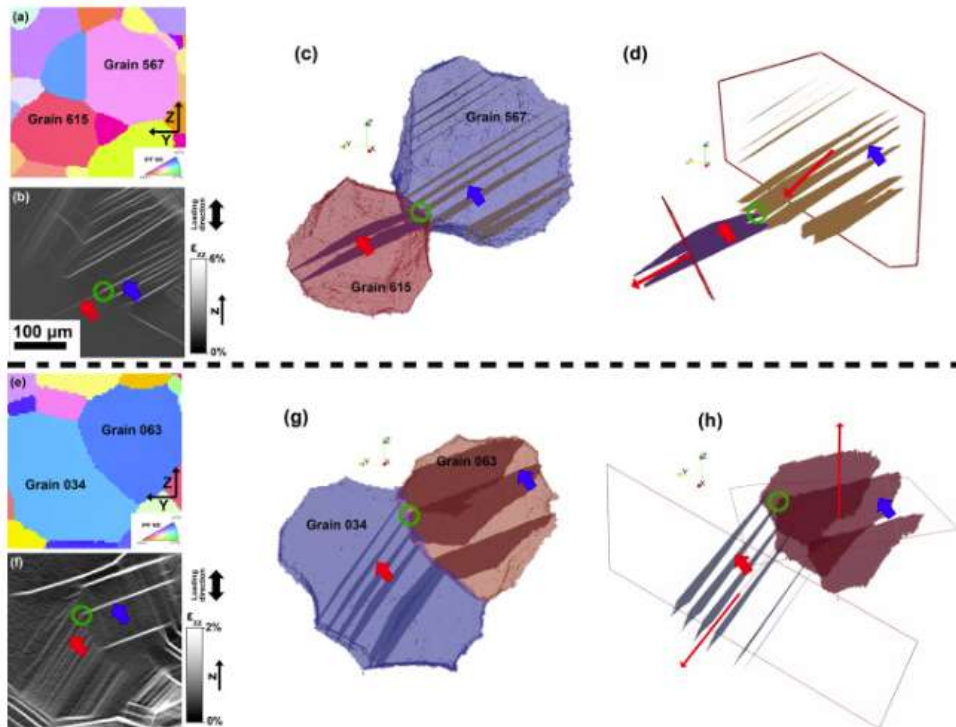
Scanning grain n°1

Observation of bulk plasticity in a polycrystalline titanium alloy by diffraction contrast tomography and topotomography

J.C. Stinville^a, W. Ludwig^{b,c}, P.G. Callahan^d, M.P. Echlin^e, V. Valle^f, T.M. Pollock^e,
H. Proudhon^g  

Materials Characterization

Volume 188, June 2022, 111891

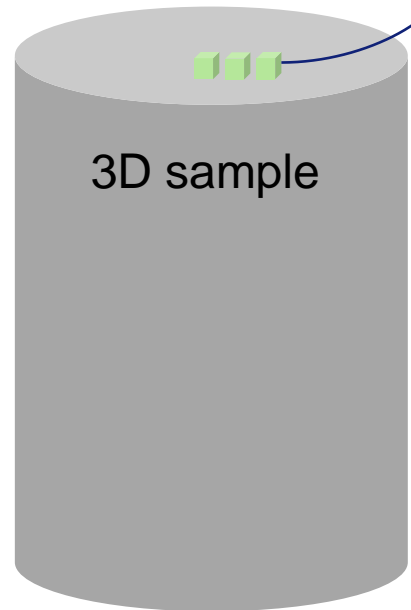


New control system.
Macros developed to
drive these scans
(Wolfgang Ludwig,
Henry Proudhon,
LTP)

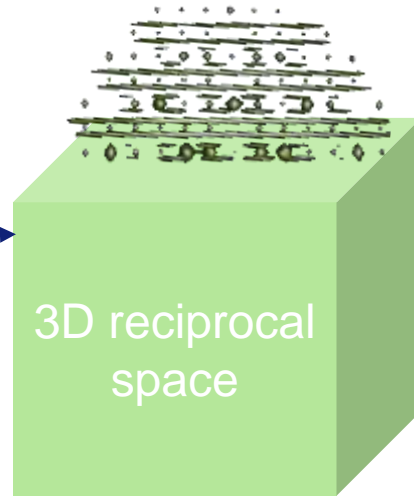
Inverse problems & reconstruction methods

Nominally a 6D problem

3D diffraction data for each voxel

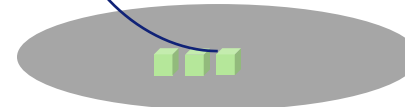


3D sample



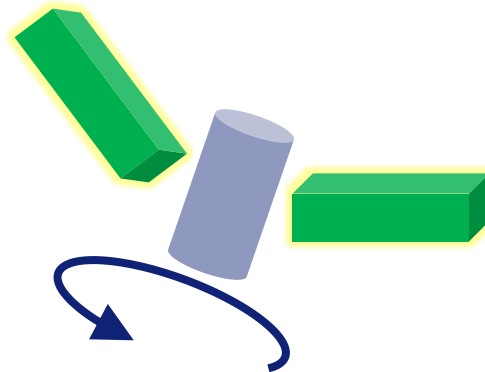
3D reciprocal space

Slices
Stack of **5D**
problems

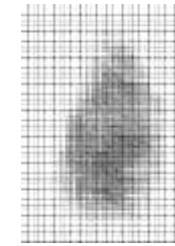


2D slices

Topo-Tomo:
Full 3D image of sample
... at one Bragg peak



DCT : **3D** data
6D problem



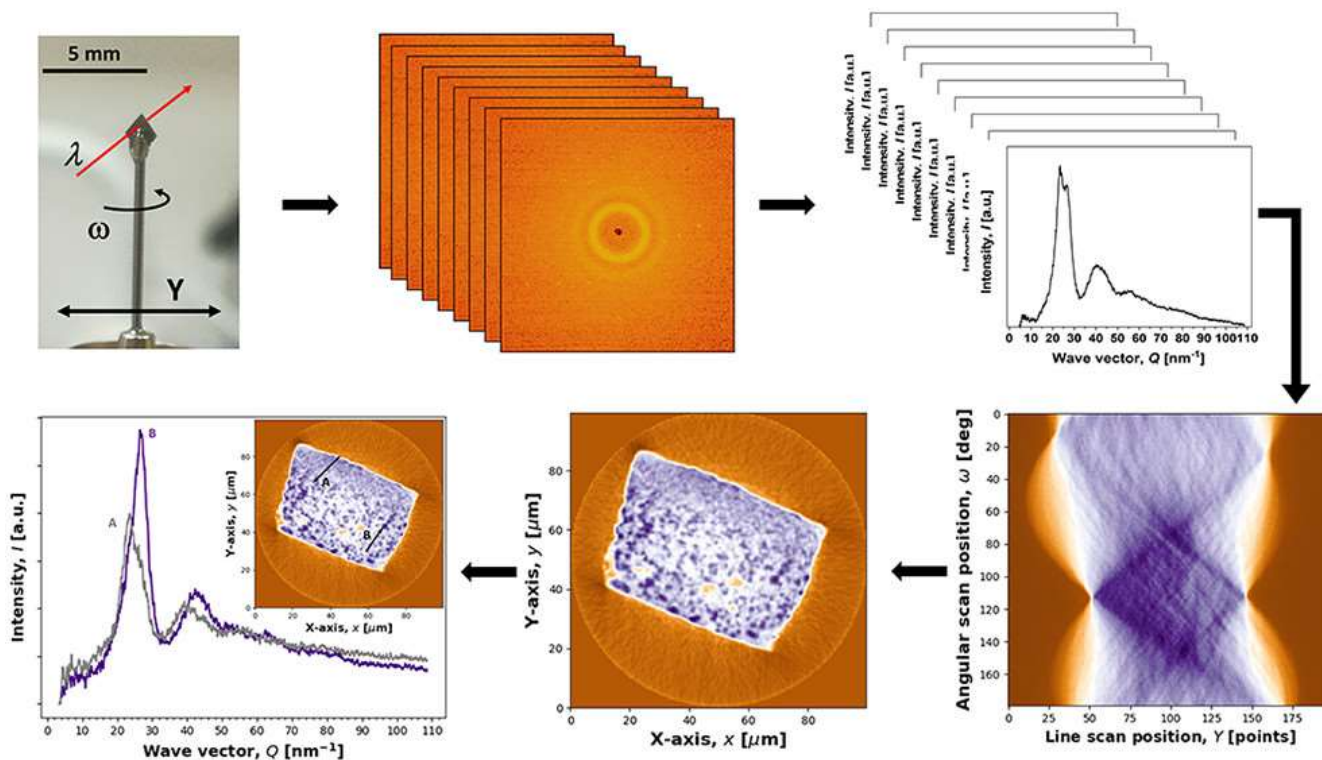
XRDCT / s-3DXRD
4D data per slice



HEDM: **3D** data per slice
(+ detector distances)

Sample size = Number of Orientations
Deformation adds more orientations

Imaging Via Diffraction : scanning method for glasses



Amorphous, nanocrystalline

Smooth $S(Q)$, 1D function

Radially symmetric

Average 2D image \rightarrow 1D $S(Q)$

iradon for each $S(Q)$ pixel value

ACS NANO

www.acsnano.org

X-ray Diffraction Computed Nanotomography Applied to Solve the Structure of Hierarchically Phase-Separated Metallic Glass

Mihai Stoica,* Baran Sarac, Florian Spieckermann, Jonathan Wright, Christoph Gammer, Junhee Han, Petre F. Gostin, Jürgen Eckert, and Jörg F. Löffler

Cite This: ACS Nano 2021, 15, 2386–2398

Read Online

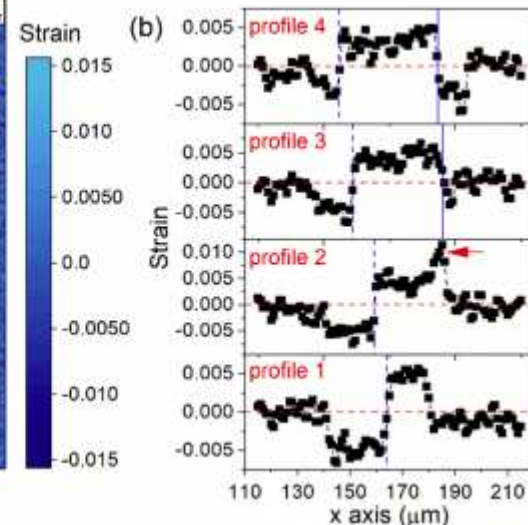
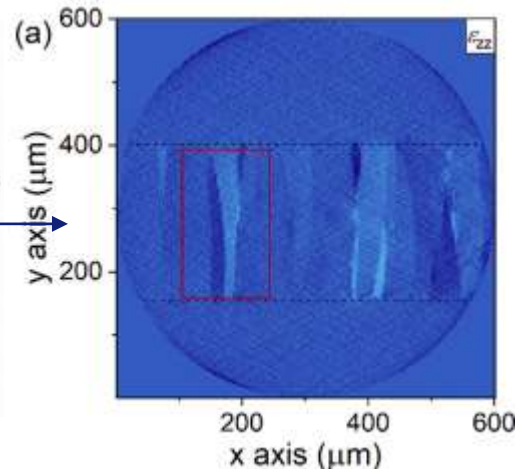
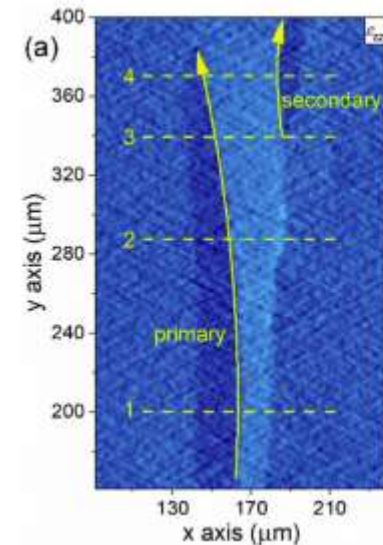
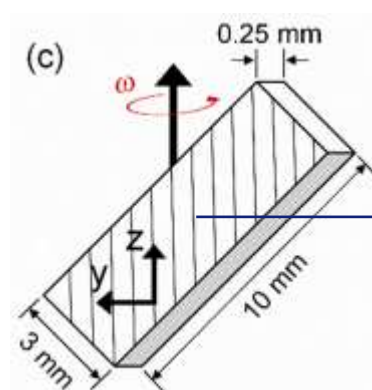
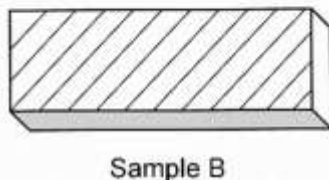
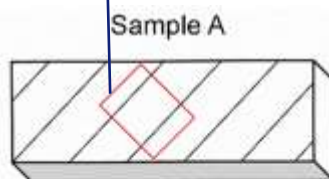
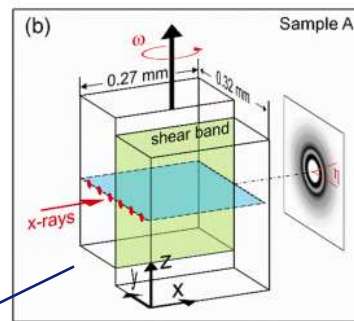
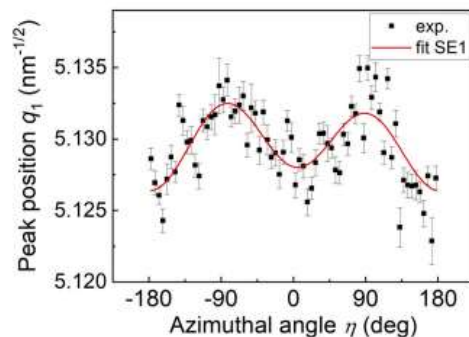
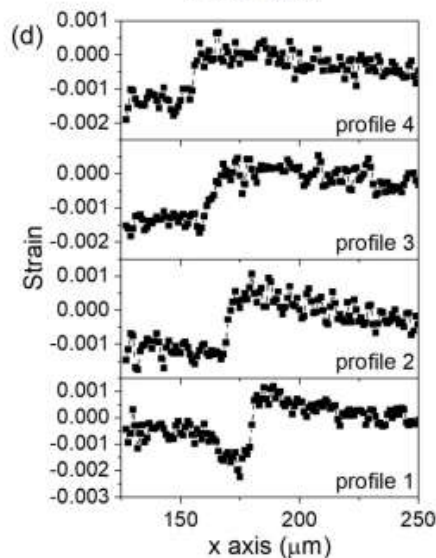
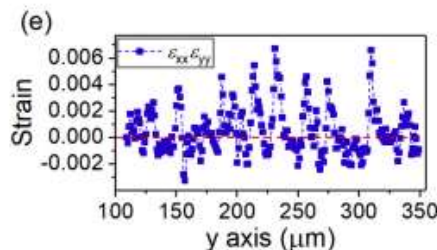
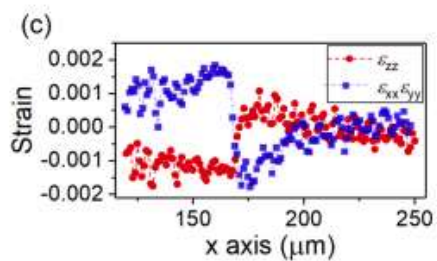
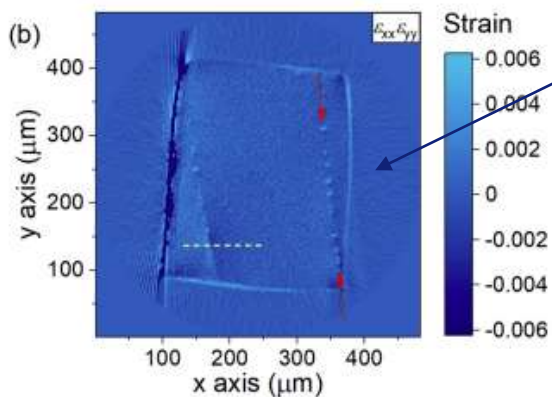
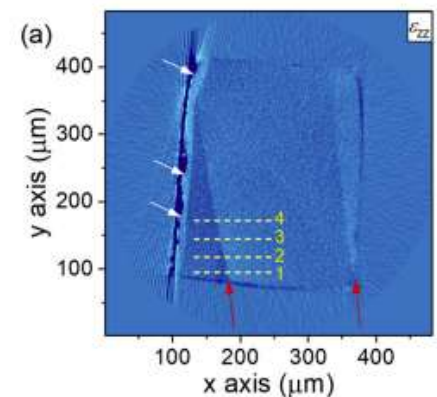
XRDCT in bulk metallic glasses

Strain fields as local probe for X-ray diffraction tomography: Non-destructive reconstruction of shear band paths in metallic glasses

Sergio Scudino ^{a,*}, Junhee Han ^b, Rub Nawaz Shahid ^{a,1}, Dina Bieberstein ^a, Thomas Gemming ^a, Jon Wright ^c

S. Scudino, J. Han, R.N. Shahid et al.

Journal of Alloys and Compounds 958 (2023) 17048



Strain tensor reconstruction

Scanning 3DXRD Measurement of Grain Growth, Stress, and Formation of Cu_6Sn_5 around a Tin Whisker during Heat Treatment

Johan Hektor ^{1,*}, Stephen A. Hall ^{1,*}, N. Axel Henningsson ¹, Jonas Engqvist ¹, Matti Ristinmaa ¹, Filip Lenrick ² and Jonathan P. Wright ³

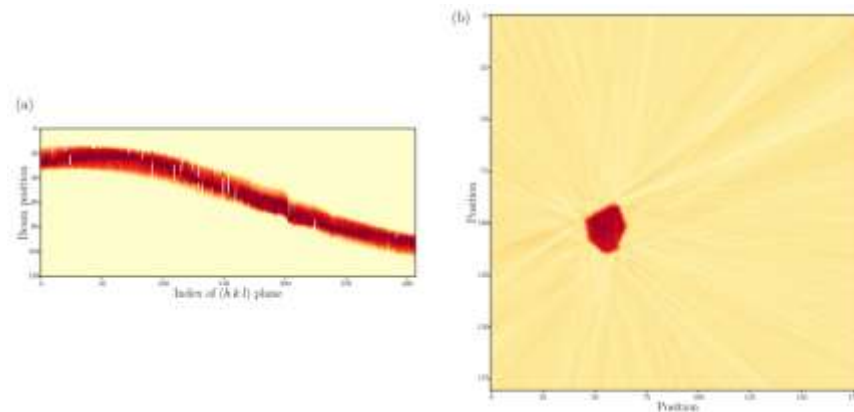
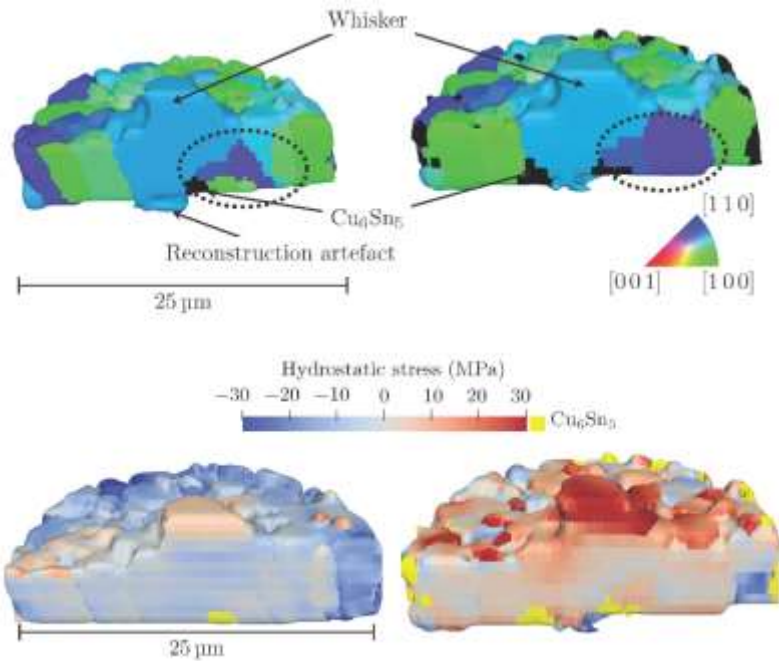


Figure 3. (a) Sinogram of one tin grain: The sinogram shows the sum of the intensities of all diffraction peaks belonging to the specific grain as a function of the diffracting lattice planes and the beam coordinate, y . The rows of the sinogram are normalised by the maximum intensity at each beam position. (b) Grain shape and position for one grain reconstructed by the inverse Radon transform of the sinogram.

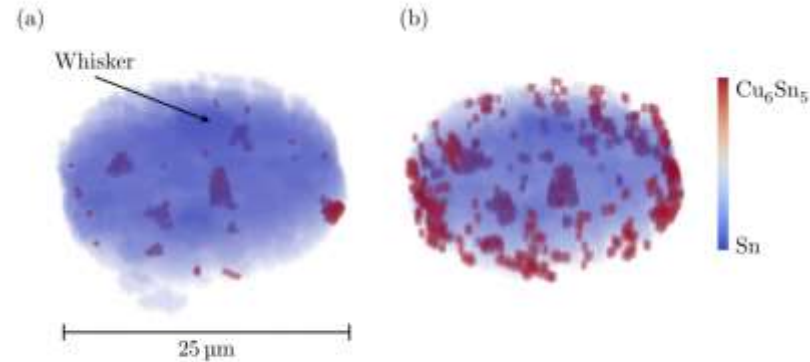


Figure 10. 3D map showing the location of Cu_6Sn_5 grains (red) in the Sn coating (blue): (a) before heat treatment; and (b) after heat treatment of 150°C for three hours.

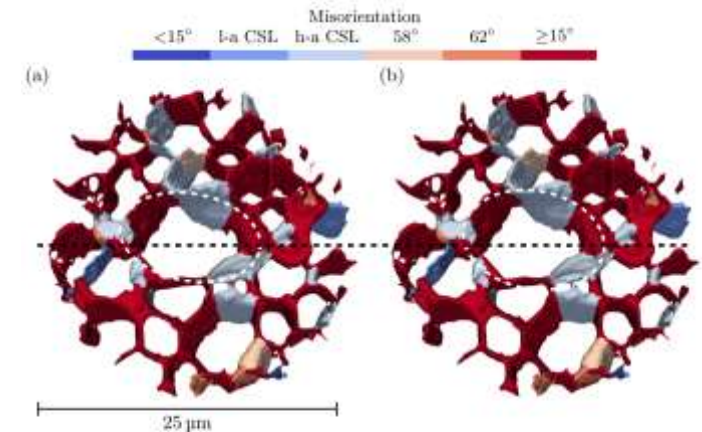
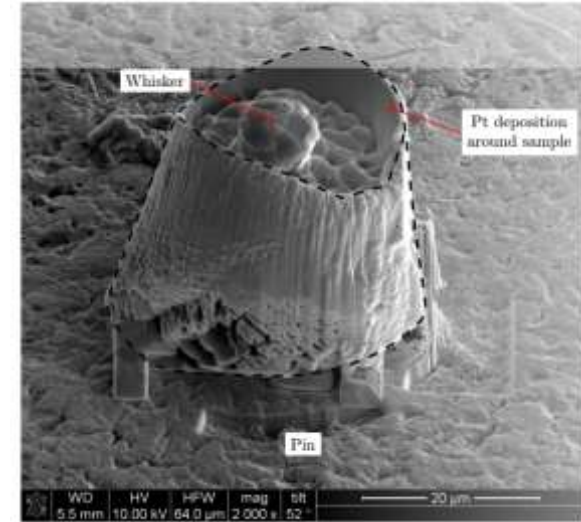
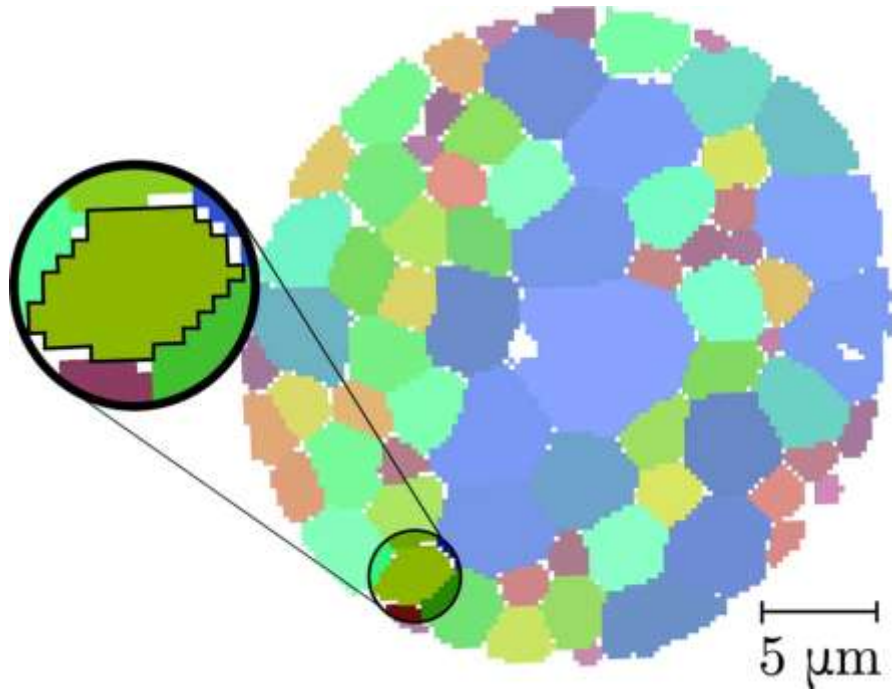


Figure 8. 3D renderings of the grain boundary network before and after heat treatment. The grain boundaries are coloured based on their misorientation angle. Blue and light blue represent low-angle CSL ($\le 15^\circ$ CSL) and high-angle CSL ($> 15^\circ$ CSL) boundaries, respectively. The dashed ellipses indicate the location of the whisker. The dashed black line indicates where the volume in Figure 4 is sliced: (a) before heat treatment; and (b) after heat treatment.

Strain reconstruction : Type II + III strain fields



Orientation – does not shift peaks radially

Strain – shifts peaks in all directions



Reconstructing intragranular strain fields in polycrystalline materials from scanning 3DXRD data

N. Axel Henningsson,^{a*} Stephen A. Hall,^a Jonathan P. Wright^b and Johan Hektor^{c,d}

Volume 53 | Part 2 | April 2020 | Pages 314–325 | 10.1107/S1600576720001016

SCR = Single Crystal Refinement (+ Hayashi *et al*)

PCR = PolyCrystal Refinement

ASR = Algebraic Strain Refinement (linear formulation)

Acta Cryst. (2023). A79, 542–549

An efficient system matrix factorization method for scanning diffraction based strain tensor tomography



Axel Henningsson* and Stephen A. Hall

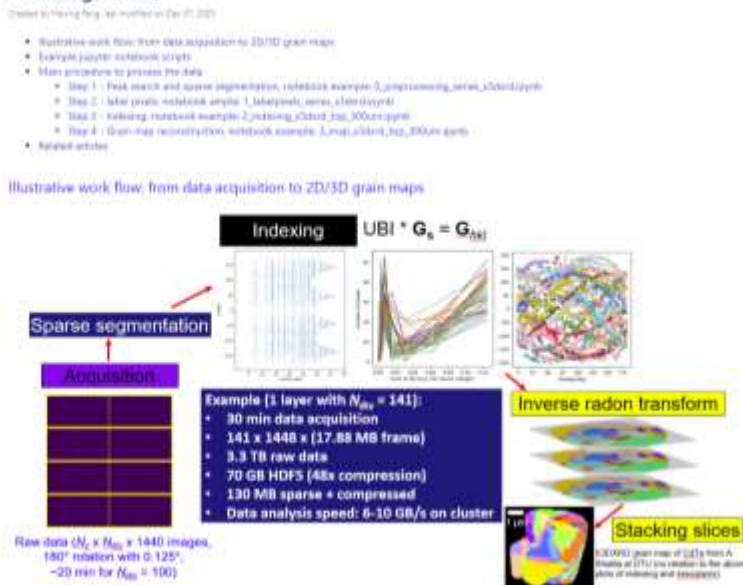
Strain + Orientation with small deformations

Methods are advancing rapidly...

... but experiments can still keep up with processing

3DXRD methods : how fast ? Time per voxel or time per orientation?

Scanning 3DXRD



Scanning + Eiger

1 rotation ~ ≥12 s on nscope
 30 steps : 6 minutes
 30x30x30 volume = 3 hrs

100 steps = 20 minutes
 1200 steps = 4 hours

In one shift:

3D : [49, 49, 49]
 voxels=1e5

2D : [2400, 2400]
 voxels=5.7e6

5) gtMergeDCTSeries: stitching multiple vertical scans

The following function finds the relative shifts between each of the vertical scans and stitches multiple scans into one volume.

```

$ stitch_williams_2_scans_2_overlap is the overlapped length between two consecutive vertical scans
gtMergeDCTSeries(parameters, 2_overlap, overlap)

$ a more specific example by inputting arguments different from the default setting
gtMergeDCTSeries(parameters, 2_overlap, "subVolume", "Final_2_3D", "FinalCase_profile", "stitched_subvol")

$ If you have forward reconstructed grain maps, i.e. $ ForwardRecon/DCTSerialDataSub_volumes
gtMergeDCTSeries(parameters, 2_overlap, "recon_mode", "recon")
    
```

By default, it will create a folder above the parameters directory named DS_stitched_gt, which contains DS_stitched_gt.mat, stitch_for_stitching_toward_model_reconstructed_grain_maps, it will create a folder above the parameters directory named DS_stitched, and stitched_forward.datml.

You can open the .datml file with ParaView software for visualization.

Figure below shows the principle of the stitching:



Near Field Detectors

~6 minutes per slice
 (count longer for deformed)

Z-slices "HEDM":
 ~ [500, 500, 80]
 voxels=2e7

nf-HEDM ~ 500x500x1

Z-stack DCT volumes:
 ~ [500, 500, 400, 10] x 8
 voxels=8e9
 3e⁵ orientations for 30³

DCT ~ 500x500x400

Haixing Fang has written docs! see: confluence.esrf.fr

Grain growth via DCT

Three-dimensional grain growth in pure iron. Part I. statistics on the grain level

Jin Zhang^a, Yubin Zhang^{b,*}, Wolfgang Ludwig^c, David Rowenhorst^d, Peter W. Voorhees^e, Henning F. Poulsen^{a,**}

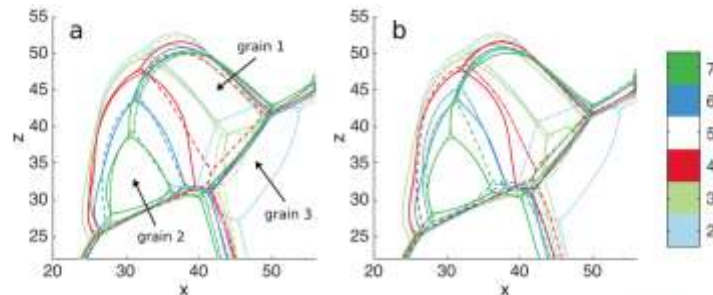
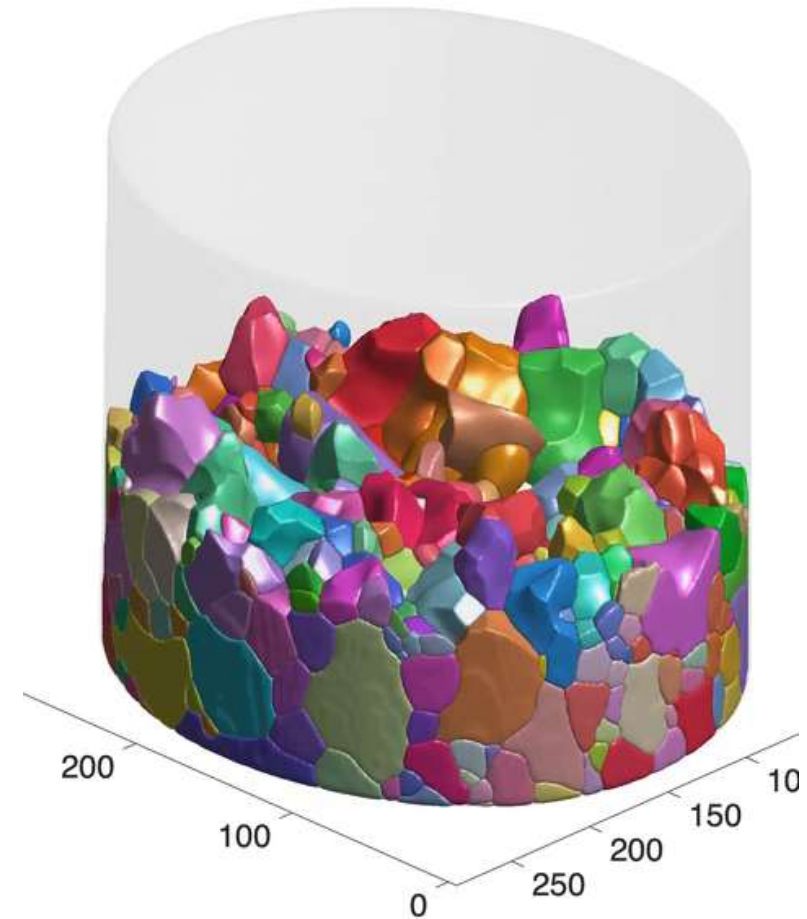
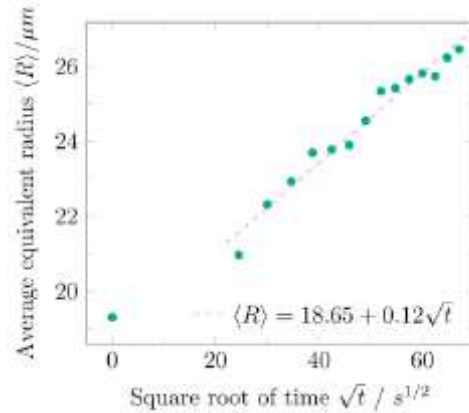
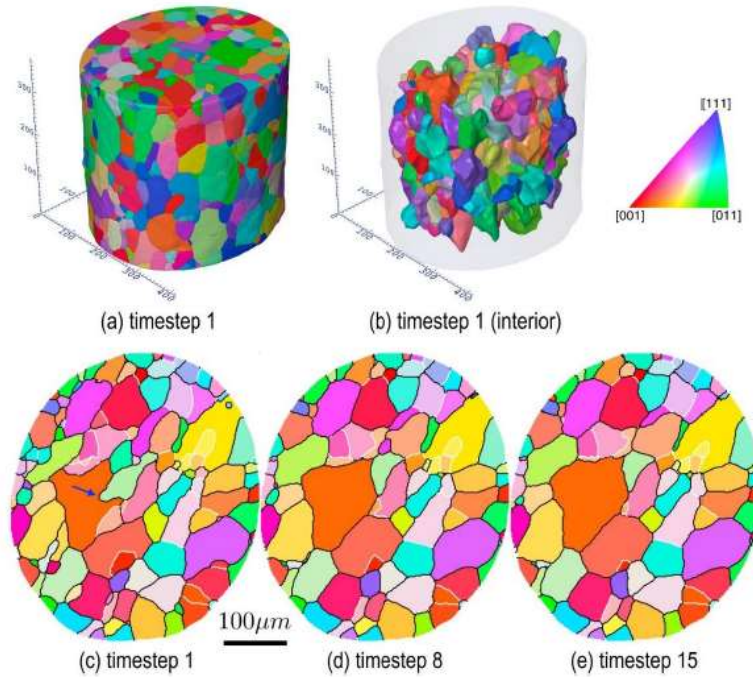


Fig. 9. Comparison between experiment (solid lines) and two types of simulations (dashed lines) for the topological charge depicted in Fig. 8. (a) Simulations at time-step t_n with the resulting optimized reduced mobilities from local fittings in Table 2. (b) A two-step model running from time-step 2 to time-step 7. In both cases, time-step 5 is omitted due to a relatively larger uncertainty on the experimental data. The unit of axes is voxel size. Contours in the 2D section for various time-steps are identified by the bar to the right.

Grain boundary mobilities in polycrystals

Jin Zhang^{a,b}, Wolfgang Ludwig^c, Yubin Zhang^d, Hans Henrik B. Sørensen^e, David J. Rowenhorst^f, Akinori Yamanaka^g, Peter W. Voorhees^d, Henning F. Poulsen^{a,b}

Acta Materialia 191 (2020) 211–220

Thank you



STREAMLINE

Haixing Fang

Wolfgang Ludwig

Pierre-Olivier Autran

James Ball

Pedro Damas-Resende

Zheheng Liu

Henri Gleyzolle

Eric Gagliardini

Emmanuel Papillon

Jose Maria Clement

