



FDMNES

User's Guide

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Introduction

The FDMNES program calculates the spectra of different spectroscopies related to the real or virtual absorption of x-ray in material. It gives the absorption cross sections of photons around the ionization edge, that is in the energy range of XANES in the EXAFS. The calculation is performed with all conditions of rectilinear or circular polarization. In the same way, it calculates the structure factors and intensities of anomalous or resonant diffraction spectra (DAFS or RXD). FDMNES also allows the comparison of the simulated spectra to experimental ones with the help of objective criteria.

The code uses two techniques of fully relativistic monoelectronic calculations (DFT-LSDA) with optionally the Hubbard correction (LSDA+U). The first one is based on the Finite Difference Method (FDM) to solve the Schrödinger equation. In that way the shape of the potential is free and in particular avoid the muffin-tin approximation. The second one uses the Green formalism (multiple scattering) on a muffin-tin potential. This approach can be less precise but is faster. The program includes also a multi-electronic extension using the "Time-Dependant DFT" with a local kernel. The program is symmetrized. Symmetry operations are calculated automatically.

The next section contains the information for practical purposes to run the program and in particular the description of the indata files. An introduction to x-ray spectroscopies is available in French.

The FDMNES program can be freely downloaded at the web address:

<http://www.neel.cnrs.fr/fdmnes>

In case of publication related to the use of the program thanks to cite:

Y. Joly

"X-ray absorption near edge structure calculations beyond the muffin-tin approximation"
Phys. Rev. B **63**, 125120 (2001).

The FDMNES program highly benefited from the scientific contribution of Calogero Natoli who has provided a constant and essential help. He is in particular at the origin of all the developments using the multiple scattering theory and the extensions to resonant diffraction and magnetism. Oana Bunau realized the extension towards TD-DFT and participated to the inclusion of self-consistency and Hubbard correction. The program also benefited from the expertise of Delphine Cabaret, Hubert Renevier, Sergio Di Matteo, Christian Brouder, Aline Ramos and Emilio Lorenzo without whom different advances would have not been realized. Finally, this work has been made greatly easier with the support of Denis Raoux.

Running the program

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A- General presentation

I- Computer configuration

FDMNES run on all the computers, having at least 256 Mo of RAM, under LINUX or Windows. The programming language is Fortran 90. It uses the LAPACK library. There is no graphical output. The user must have got a Fortran 90 compiler. Nevertheless an executable files for Windows XP and LINUX 64 bits are also provided. The code can be run on parallel processors if it is linked with the MPI library.

II- The package

Several groups of files can be downloaded: the *fdmnes* program itself and a set of examples of indata and output files.

- *fdmnes.exe* is the executable program for Windows XP,
- *fdmfile.txt* is an indata file,
- *xsect.dat* and *spacegroup.txt* are data necessary for the program,
- *prog* is a directory containing all the subroutines (*main.f*, *general.f...*); and the file "*mpif.h*".
- *entree* and *xanout* are directories containing a set of examples of *fdmnes* indata and output files.

III- Differences with previous versions

In the versions before 2005, the calculations were necessarily performed in two steps, first the main calculation, second, the convolution. Two different programs had to be run. Now these two steps can be executed together with the unique package *fdmnes*. Nevertheless, the user can separate the different phases of the calculation as previously, as is explained in the chapter describing the main indata file. For the convolution part, the broadening must be now specified is the width of the levels. Previously, it was the half-width.

Different improvements were also added. The program can furnish the different scattering tensors in their Cartesian or spherical basis. It is now possible to compare the calculated spectra to experimental ones and to perform calculations on different grid of parameters in order to fit them.

Some details have also changed: the old "*indata.gen*" file is now called "*fdmfile.txt*". In the main indata file, after the keyword "*range*", it is not anymore necessary to put the number of energies, the first line is not anymore reserved for a comment and the character "!" makes that what follows in the line is not red.

From October 2005 the non resonant magnetic scattering is taken into account for the RXS.

In October 2006 some bugs were corrected. The self absorption is now calculated for resonant diffraction. The fit of parameters by comparison with experimental spectra is now possible.

In May 2007, the program symmetrization was improved. It takes into account the atoms with non spherical configuration. The magnetic moments do not need anymore to be

parallel. The description of the local atomic base is modified (see keyword "*crystal*"). Calculations in the finite difference method mode are now faster when three fold axis are present.

In March 2008, the symmetrization for the multiple scattering mode without spin-orbit is improved. The program contains also a new module allows the building of new output files extracted from the "scan" files obtained after the convolution. These new files contain spectra or azimuthal scan at specific energy.

In February 2009, the self-consistent calculations are introduced. There is by default an automatic evaluation of the Fermi energy.

In January 2011, the Hubbard correction (LDA+U) and the "Time-Dependant-DFT" are introduced. This last works in the local mode (TD-LSDA, or ALDA).

IV- Compilation

All the Fortran routines must be compiled and linked with "*mpif.h*" in the same directory because these files are called during the compilation. "*sub_util.f*" is a file containing different routines from the Lapack and Blas library. It can be advantageous to replace it by the calling to these libraries.

When using a Cray, one has to:

- modify the calling to lapack routines have a different name (zgetrf, zsytrf, zgetri, zsytri in routine *invcomp*). Lines to modify are spotted by the comment *!cray*.
- modify in the procedure *timesec* is in the file *main.f*, the calling to time, changing the two lines :

```
tps4 = secnds(0._4)
```

```
tps = dble(tps4)
```

by the only line:

```
tps = second()
```

V- Parallelization

Thanks to Keisuke Hatada, Kuniko Hayakawa and Rainer Wilcke, the user shaving the access to a cluster of computer can using the MPI library, run the program in parallel mode. For this one has to delete the files "*mpif.h*" and "*not_mpi.f*" when compiling and make the call to the corresponding library.

VI- Running

After compilation, the program can be run following the usual procedure available on your system. The files "*xsect.dat*" and "*spacegroup.txt*" must be set in the same directory than the executable file because they are used in some cases.

As soon as the program is running, it calls the file "*fdmfile.txt*". This file must be in the same directory than the executable file. It only contains the number of independent calculation to perform, followed the name of the indata file of each of these calculation. For example:

```
! Indata file for fdmnes
```

```
1
```

→ number of indata files

example/cu/cu_inp.txt

→ name of the indata file

VII- Troubleshooting

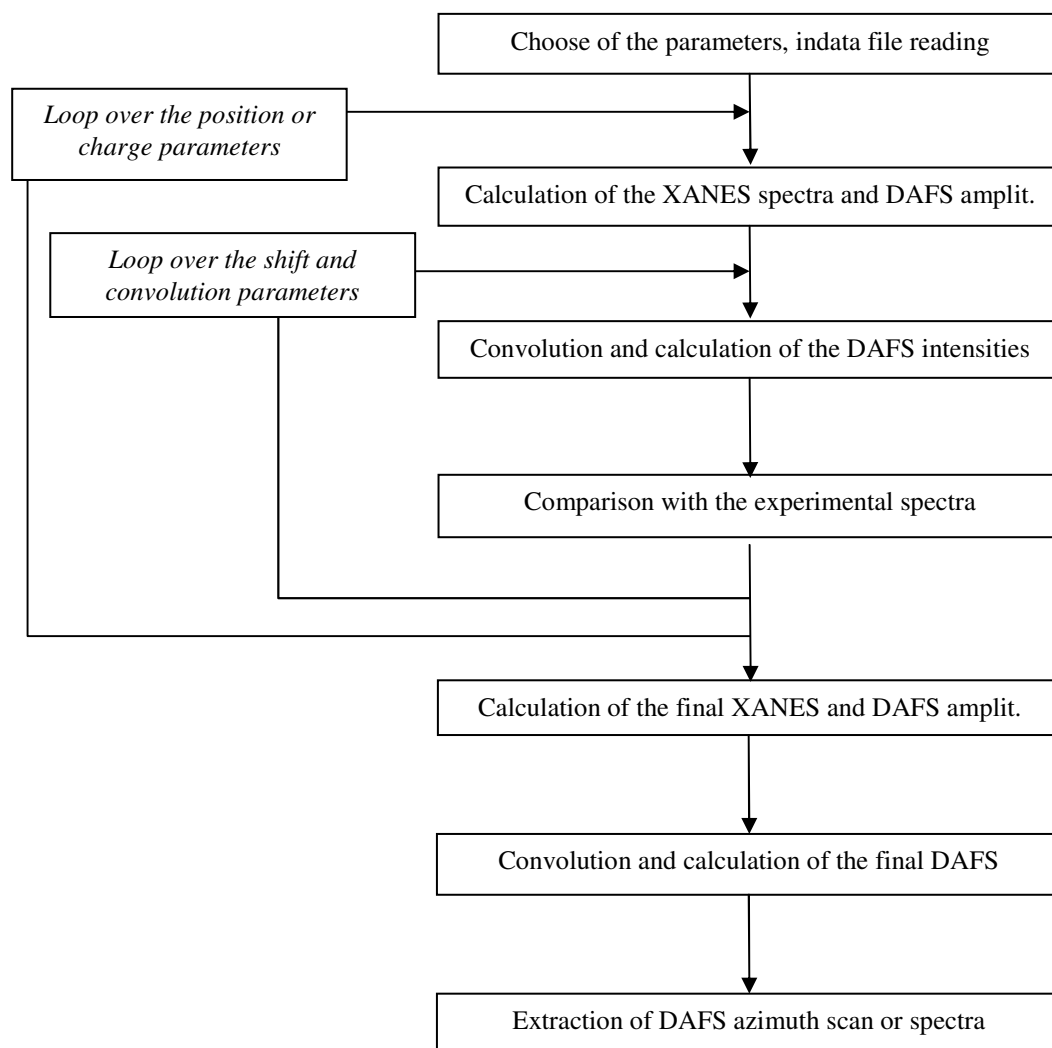
When the program stops without reason, check if an “*fdmnes_error.txt*” file has been created. When created, this file contains a message explaining the trouble. Most often it is due to an error in the indata file. It can also be due to the fact that the indata file is not found. One has to check several points:

- 1) The called files are in the good directory. The name must contain all the paths.
- 2) The extensions are correct.
- 3) Their names are OK (under linux, upper and lower cases must be respected).
- 4) When using downloaded indata files, some problems of compatibility between systems can occur. It can be better to write again completely these files.
- 5) Avoid the tabulation. Their presence can give strange comments in “*fdmnes_error.txt*”.

When the program stops without “*fdmnes_error.txt*” file, it can be due to a problem of space memory. (Sometimes one gets a message with “stacking fault”). In this case try again putting in the indata file the keyword “Memory_save”. This keyword can be useful when there are many non equivalent atoms. An approximation (in fact very good) is then done on the potential calculation. This option saves some memory space.

VIII- Structure of the calculation

The program allows the calculation of spectra from a grid of parameters (position, charge...), to convolute them with other parameters (convolution width, energy shift...), then to compare them to experimental spectra with the help of objective criteria. These different steps can be performed together or separately. For a complete calculation one gets the following scheme:



Nevertheless, it is possible to perform the different steps separately. The comparison with the experimental spectra is also not obligatory. When used, the parameter fit must be performed with care. In practice many calculation are limited to the step “XANES and DAFS calculation” and “Convolution and calculation of DAFS intensities”. These two steps can also be performed together or separately.

The next chapter treats about the principal indata file for the step “XANES and DAFS calculation”. Generally this file is sufficient to describe all the necessary data for the calculation because the program calculates its atomic bases and the potential. Nevertheless, the user can prefer use its own atomic bases or uses directly the potential calculated by the band structure program FLAPW WIEN-2k. In both cases, some other files must be furnished. They are described further in the manual. The indata necessary for the steps “Convolution”, “comparison with the experimental spectra” and “Extraction of azimuth scan or spectra” can be set in the same indata file, but they are explained separately in the chapter C, D and E.

B- Main indata file

1) General Structure

It contains most of the inputs necessary for the calculation. All the data in input and output are in Angstrom and electron-Volt. Many parameters are chosen by default. One can modify those using keywords. Text can be in upper or lower case. Blank lines or beginning by ! are not taken into account. Between number, one must put at least one blank. Tabulations are forbidden. When getting problem when opening these indata files, one has to check if their name is correct. Moreover some compilers do not like files written under other system (MAC, DOS, LINUX...). In case of difficulties when the program wants to open one of these downloaded files, it can be useful to completely write them again.

The indata file contains several blocks of data, each ones starting with a specific keyword. The end of the indata file is noted by the keyword "end". Whatever is after is not read. Here comes an example of indata file:

! Fdmnes indata file

! Calculation for the copper K-edge in copper cfc

Filout

example/cu/cu_out

→ Name of the output files (without extension)

Range

-10. 0.2 0. 0.5 10. 1. 40.

Radius

3.0

→ Cluster radius

Crystal

*3.610 3.610 3.610 90. 90. 90.
29 0.0 0.0 0.0
29 0.5 0.5 0.0
29 0.5 0.0 0.5
29 0.0 0.5 0.5*

→ crystal structure

→ Mesh parameters (Å and degrees) : a, b, c, α, β, γ

→ Atomic number, position

Convolution

→ to get convoluted spectra

Efermi

-6.

End

→ end of the indata file

Two blocks are necessary for any calculations. The first one starts with the keyword "rayon" or "radius" followed by the value of the radius inside the calculation is performed. The second one is necessary to describe the material structure. When that one comes from the output of the WIEN-2k package, it begins with the keyword "flapw". In the other cases, the molecule structure or the elementary mesh, in case of a 3D structure, is described in the file. This description starts respectively with the keyword "molecule" or "crystal" (or "cristal").

All the keywords related to the convolution or to the fit of the parameters are treated in chapter C and D.

Output file names

By default the output file name is *fdmnes_out*. This name can be modified by the use of the keyword "filout" followed by the name we want (without extension). There will have several output files by adding the extensions:

_bav.txt output file giving details
_.txt contains only the spectra by column

If a calculation is performed on several non equivalent crystallographic sites, one gets the extensions:

_i.txt, _j.txt ... where i and j are the index of the sites (see keyword *absorber*)

In option or depending on the type of calculation, one can also get the files:

_conv.txt convoluted spectra scan (keyword *Convolution*).
_scan.txt dafs versus angles for azimuthal scan (keyword *rxs*).
_sda.txt state density for the atom number a (keyword *density*).
_atoma.txt results for one atom at position number 'a' (keyword *allsite*).
_atoma_scan.txt rxs scan results pour for the atom a (keyword *allsite* and *rxs*).
_tddft.txt output with the TDDFT option (keyword *tddft*).
_tddft_scan.txt azimuthal scan in the TDDFT option (keyword *rxs* and *tddft*).
_tddft_conv.txt convoluted spectra in TDDFT (keyword *Convolution* and *tddft*).

For the analysis of the spherical tensors (keyword *spherical* and *sphere_all*)

_sph_atoma.txt spherical tensors of the atom a.
_sph_atoma_int.txt integral of the spherical tensors of the atom a.
_sph_signal_atoma_xan.txt contribution of each atomic spherical tensor on the average xanes signal.
_sph_signal_atoma_poli.txt contribution of each atomic spherical tensor on the xanes polarisation number i.
_sph_signal_atoma_rxsi.txt contribution of each atomic spherical tensor on the rxs reflection number i.
_sph_signal_xtal_xan.txt contribution of the crystal spherical tensor on the average xanes signal.
_sph_signal_xtal_rxsi.txt contribution of the crystal spherical tensor on the rxs reflexion number i.
_sph_xtal.txt spherical tensors of the crystal.
_sph_xtal_int.txt integral of the spherical tensors of the crystal.
_sph_xtal_rxsi.txt spherical tensor of the crystal for the rxs reflexion number i.

For the analysis of the cartesian tensors (keyword *cartesian*):

_car_atoma.txt cartesian tensors of the atom a.
_car_xtal.txt Cartesian tensors for the crystal
_car_xtal_rxsi.txt Cartesian tensors for the crystal for the rxs reflection number i.

II- Basic keywords

II-1) Output file names

The different output files have names with the same root. The extensions are automatically added depending on the chosen option. To define this root use :

Filout

example/cu/cu_out

→ Name of the output files (without extension)

The files can eventually be in a subdirectory.

II-2) Radius of the cluster

The final states are calculated inside a sphere, whose radius is defined with the keyword "*radius*" (or "*rayon*"). Only the atoms inside this sphere are considered.

Radius

→ Obligatory keyword preceding the radius of the cluster.

3.5

→ value in Angstrom of the cluster radius.

II-3) Cluster or crystal structure

Under "*crystal*" or "*molecule*" stand all the data describing respectively the unit mesh or the molecule. If the calculation is done using the flapw output, this block is useless because the structure is read in one of the flapw output files. Under the keyword, come the mesh parameters (Å) and the angles (degrees). Then come all the atoms (and not only the non equivalent ones, but when one specifies the space group using the keyword "*spgroup*"). By default and in the absence of the keyword "*absorber*", the absorbing atom chemical species corresponds to the first atom in the list.

Example 1. fcc copper crystal:

Crystal

```
3.610 3.610 3.610 90. 90. 90.
29 0.0 0.0 0.0
29 0.5 0.5 0.0
29 0.5 0.0 0.5
29 0.0 0.5 0.5
```

→ Crystal structure (or "*cristal*")

→ a, b, c, α , β , γ

→ Atomic number, position

Example 2. FeO₆ octahedron:

molecule

```
1.900 1.900 1.900 90. 90. 90.
26 0.0 0.0 0.0
8 1.0 0.0 0.0
8 -1.0 0.0 0.0
8 0.0 1.0 0.0
```

→ a, b, c, α , β , γ

→ Atomic number, position

```

8  0.0  -1.0  0.0
8  0.0   0.0  1.0
8  0.0   0.0 -1.0

```

The atomic structure can also be given in cylindrical or spherical coordinates. To use cylindrical coordinates, it is sufficient to give only two numbers under "*molecule*". The program will understand they are *a* and *c* and that the positions of the atoms are given by *r*, ϕ and *z*. Thus to describe the same octahedron as previously:

molecule

```

      1.900 1.900          → a, c
26  0.0    0.0  0.0      → Atomic number, position
8   1.0    0.0  0.0
8   1.0  180.0  0.0
8   1.0   90.0  0.0
8   1.0  -90.0  0.0
8   0.0    0.0  1.0
8   0.0    0.0 -1.0

```

To use spherical coordinates, only one number (*a*) must be set after "*molecule*". The position of the atoms is then given by *r*, θ , ϕ . For the same octahedron:

molecule

```

      1.900          → a
26  0.0    0.0  0.0      → Atomic number, position
8   1.0   90.0  0.0
8   1.0   90.0  90.0
8   1.0   90.0 180.0
8   1.0   90.0 270.0
8   1.0    0.0  0.0
8   1.0  180.0  0.0

```

It is possible, in the non magnetic case, to specify only the non-equivalent atoms. Then one has to give the space group under the keyword "*Spgroup*". The complete name as in the international table must be given. For example, for the magnetite, one gets:

Spgroup

```

Fd-3m:1          → it is also possible to write 227:1

```

Crystal

```

8.3940 8.3940 8.3940 90.0 90.0 90.0

26  .6250  .6250  .6250  ! Fe 16d
26  .0000  .0000  .0000  ! Fe 8a
8   .3800  .3800  .3800  ! O 32e

```

Note that for the calculation of the symmetry, it is important to define the atom position with a sufficient number of digits (say 10). For instance, for graphite one has to write:

Spgroup
P63mc

Crystal

2.456 2.456 6.696 90. 90. 120. = a, b, c, alpha, beta, gamma
 6 0.0 0.0 0.0
 6 0.3333333333 0.6666666667 0.0

Putting in the last line for example 0.3333, would create false atoms. Program would stop immediately with an error message.

It is also possible to have an occupancy rate not full for the atoms. For this “_t” must be added after keyword “crystal” or “molecule” and a fifth column after the atom position contain this weight :

Crystal_t → (or **Molecule_t**)
 2.456 2.456 6.696 90. 90. 120.
 6 0.0 0.0 0.0 0.8 → weight 80 %
 6 0.25 0.25 0.0 1. → weight 100%

This value is taken into account in absorption just for the absorbing atom and for RXD as a weight on the scattering amplitude.

II-4) Atomic electronic densities

An electronic configuration is used by default for all the atoms. It is possible to modify it by the use of the keyword “atom”. Moreover under “crystal” or “molecule” one must not anymore put the atomic number, by the atom type number. For example in case of a FeO₆ octahedron with the Fe 3d⁶4s² and O 2s²2p⁴ configuration:

Atom → keyword preceding the atomic electronic densities
 26 2 3 2 6. 4 0 2. → atomic number of the chemical specie of type 1, number
 8 2 2 0 2. 2 1 4. of valence orbital and (n,l,po) of each of these orbitals

molecule

1.900 1.900 1.900 90. 90. 90. → a, b, c, α, β, γ
 1 0.0 0.0 0.0 → Atom type, position
 2 1.0 0.0 0.0
 2 -1.0 0.0 0.0
 2 0.0 1.0 0.0
 2 0.0 -1.0 0.0
 2 0.0 0.0 1.0
 2 0.0 0.0 -1.0

Important remark: contrary to what one can think, the formal charges attributed to the atoms in the ionic compounds are far from the true charge. Thus one has to perform exchange of charge between atoms with care and in a moderate way. A good technique is to set the good number of “d” electron, following the formal charge, but keeping the neutral atom, putting electrons in the large radius 4s or 4p orbitals for a transition metal.

II-5) Absorbing atoms

All the atoms present in the structure participate to the scattering. By default, the calculated spectra correspond to the sum of the scattering produced by all the atoms of the same atomic number than the first one in the list under "*crystal*" or "*molecule*".

When using the keyword "*extract*" or one wants to select some of the absorbing atoms one has to use the keyword "*absorber*". For example, if one wants that the absorbing atom is the n^{th} of the list and only this one (under "*crystal*" or "*molecule*"), put:

Absorber

3 → absorbing atom number (here the 3rd in the list).

If this example goes with the FeO₆ structure defined just above, the result in the output file would correspond to a calculation of an oxygen atom summed over the 6 atoms because they are equivalent by symmetry. If one wants the result from a unique site, use the keyword "*symsite*" or "*allsite*" (see further on).

If there are non equivalent sites, one has to perform independent calculations. For this, one must specify each atom number under "*absorber*". The name of each of the corresponding output file get the suffix "*_n*" where n is the atom number.

Filout

copper_out

Absorber

1 5 → atom numbers from the results will be put in the output file *copper_out_1* and *copper_out_5*

The summation over these different sites and the eventual energy shifts is to be performed during a following step.

II-6) Energy range

The energy range E that one defines in the *indata* is the energy of the photoelectron relative to the Fermi level. The kinetic energy, E_c , of the photoelectrons when they leave the cluster must be positive. E_c is related to the input energy by:

$$E_c = E - W - V_m,$$

where W is the work function and V_m the average potential (equivalent to the muffin-tin ground) in the cluster.

In the case of a calculation using the *flapw* the coulombian part of the average potential V_m is zero. So V_m is greater than zero. Thus the work function is not anymore subtracted and the input energy is defined relatively to the vacuum.

By default the energy range is -5 to 60 eV by 0.5 eV step. One can change the range, the step or even have a variable step using:

Range

1. 0.5 60.

→ keyword for the energy range (or "*gamme*")

→ E_{\min} , step, E_{\max}

Other example with variable step:

Range

1. 0.1 10. 0.5 20. 1. 60.00 → E min, step, E intermediate, step ...

To get a continuously increasing step (k step constant) put:

Rangel

1. 0.1 200. → E min, step at the Fermi level, E max

By default, the output energy range is relatively to the Fermi level. If one wants that the output energy is the photon energy put the keyword:

energpho

II-7) Multiple scattering mode

If one wants to calculate in the multiple scattering mode use the keyword:

Green

Then the potential is automatically a muffin-tin one. The mode is faster than the finite difference method, so one has to use it first.

II-8) Threshold type

By default the threshold is the K one. For other threshold put the keyword:

Edge

→ keyword preceding the threshold type (or “*seuil*”)

L1

→ threshold (K, L1, L2, L3, M1 ...)

It is possible in a single run to calculate 2 edges with the same initial (n,l), that is the edges L2 and L3 or M2 and M3 or M4 and M5 ... For this write:

Edge

→ keyword preceding the threshold type

L23

→ threshold L2 and L3, or M23, M45, N23, N45.

In this case, one gets two output files with the suffixes `_L2` and `_L3`.

II-9) Multipolar expansion

By default only the transition electric dipolar component (E1E1) is calculated. This is modified by the keywords:

<i>Quadrupole</i>	→ quadrupolar calculation (E1E2 and E2E2)
<i>Octupole</i>	→ octupolar calculation (E1E3)
<i>Dipmag</i>	→ magnetic dipole calculation (E1M1) and (M1M1)
<i>E1E2</i>	→ calculation of E1E2
<i>E2E2</i>	→ calculation of E2E2
<i>E1M1</i>	→ calculation of E1M1
<i>M1M1</i>	→ calculation of M1M1
<i>No_E1E1</i>	→ No calculation of the dipolar-dipolar component (E1E1)
<i>No_E2E2</i>	→ No calculation of the quadrupolar-quadrupolar component
<i>No_E1E2</i>	→ No calculation of the dipolar-quadrupolar component
<i>No_E1E3</i>	→ No calculation of the dipolar-octupolar component

II-10) Polarization and dichroism

By default the calculation is performed along 1, 2 or 3 orthogonal polarizations in the dipole mode (and up to six in quadrupolar) depending of the symmetry. The polarizations are chosen along the axis of an internal basis in the program. By default the result is given in the output file only for their average, corresponding to a powder. If one wants to have the XANES along specific orientations of the polarization or the wave vector (in quadrupolar) orientations or if one wants to make circular dichroism, one has to introduce:

Polarize

Thus, one gets the calculations for the different independent polarizations, then, in the last column, the average. To get specific polarization orientation, the keyword must be followed by the polarizations:

Polarize

```
1.0 1.0 0.0
1.0 -1.0 0.0
0.0 0.0 1.0
```

For a quadrupolar calculation, one has to specify the wave vector after the polarization:

Polarize

```
1.0 1.0 0.0 0.0 0.0 1.0
1.0 -1.0 0.0 0.0 0.0 1.0
0.0 0.0 1.0 0.0 0.0 0.0
```

Each line contains the polarization vector, the wave vector. If the wave vector is zero this polarization is calculated in the dipolar approximation.

It is possible to perform an average on different polarizations adding a new number at the end of the line is the corresponding weight. If at least two weights are non zero, in the output, there will be a new column with the weighted average of the different polarization.

Polarize

```
1.0 1.0 0.0 0.0 0.0 1.0 1.
1.0 -1.0 0.0 0.0 0.0 1.0 1.
0.0 0.0 1.0 0.0 0.0 0.0 0.
```

If one wants to have a circular polarization, one just has to put this one to zero followed by the value of the wave vector:

Polarize

```
0.0 0.0 0.0 0.0 0.0 1.0
```

In the output one will have the calculation in right polarization, then the left polarization, then the difference. In this example, the calculation corresponds to the polarization $x+iy$ and $x-iy$.

II-11) Anomalous or resonant diffraction

In case of anomalous (or resonant) diffraction (DAFS, DANES, RXS, RXD) calculation, put the keyword "*dafs*" (or "*rxs*") followed by the index of the beams to calculate. The orientation of the polarization and wave vector can be described by different way. When working in σ - σ , σ - π , π - π or π - s , or in circular polarization, put the number 1, 2, 3, 4 or 5 for the polarization respectively σ , π , circular right, circular left or rectilinear along a general direction, in input then in output. Then one specify the azimuthal angle between the incidence plane with a (I, J, Q) base such that:

Dafs

```
0 0 2 1 2 45.    → reflection indexes,  $\sigma$ ,  $\pi$ , azimuth
0 0 2 1 1 45.    → reflection indexes,  $\sigma$ ,  $\sigma$ , azimuth
```

When the polarization is rectilinear but not σ or π , but with an angle α such that $\alpha = 0$ when it is σ and 90 when it is π , one must write:

Dafs

```
0 0 2 1 0. 5 10. 45.    → reflection indexes,  $\sigma$ , angle, rectilinear, angle, azimuth
0 0 2 1 0. 5 -10 45.
```

Note that in this case, one must specify both incoming and outgoing polarization angles, even when one of them is σ , π or circular. When it is circular, the angle is not taken into account.

(I, J, Q) is such that:

Q is the normalized diffraction vector

$$\vec{I} = \vec{Q} \wedge \frac{\vec{k} \wedge \vec{Q}}{|\vec{k} \wedge \vec{Q}|}$$

$$\vec{J} = \vec{Q} \wedge \vec{I}$$

but when Q is along Oz, in that case \vec{k} is substituted by \vec{i} . $\vec{k} = \frac{\vec{c}}{c}$ and $\vec{i} = \frac{\vec{a}}{a}$ are the bases vectors of the direct elementary mesh. One then gets, versus the Bragg angle β and the azimuthal angle φ the incoming and outgoing wave vectors:

$$\vec{v}_e = \cos\beta \cos\varphi \vec{I} - \cos\beta \sin\varphi \vec{J} - \sin\beta \vec{Q},$$

$$\vec{v}_s = \cos\beta \cos\varphi \vec{I} - \cos\beta \sin\varphi \vec{J} + \sin\beta \vec{Q}.$$

For the σ and π polarizations, one gets:

$$\vec{e}_\sigma = \sin\varphi \vec{I} + \cos\varphi \vec{J}$$

$$\vec{e}_{\pi e} = \vec{v}_e \wedge \vec{e}_\sigma = \sin\beta \cos\varphi \vec{I} - \sin\beta \sin\varphi \vec{J} + \cos\beta \vec{Q}$$

$$\vec{e}_{\pi s} = \vec{v}_s \wedge \vec{e}_\sigma = -\sin\beta \cos\varphi \vec{I} + \sin\beta \sin\varphi \vec{J} + \cos\beta \vec{Q}$$

For example: orthogonal mesh, beam (h,0,0), $(\vec{I}, \vec{J}, \vec{Q}) = (\vec{k}, -\vec{j}, \vec{i})$

1 1 0. \rightarrow corresponds to σ - σ with polarization along $-j$.

1 1 90. \rightarrow corresponds to σ - σ with polarization along k .

orthogonal mesh, beam (0,0,1), $(\vec{I}, \vec{J}, \vec{Q}) = (\vec{i}, \vec{j}, \vec{k})$

1 1 0. \rightarrow corresponds to σ - σ with polarization along j .

1 1 90. \rightarrow corresponds to σ - σ with polarization along i .

If one does not put the angle, this implies that one performs a phi scan and all the amplitudes are calculated for all the angles by 2° step. There is then a supplementary output files with the extension "_scan.txt".

It is also possible to perform a 360° scan by 2° step for the rectilinear incoming or outgoing polarizations. For this, one has to write 10 in place of the sigma-pi notification:

Dafs

0 0 2 10 1 45. \rightarrow incoming polarization is scanned

0 0 2 2 10 45. \rightarrow outgoing polarization is scanned

The first value (angle = 0°) corresponds to the σ polarization. 90° corresponds to π polarization.

It is possible to write the exact polarization directions. This can be useful for peculiar uses as in photoemission. In case of rectilinear polarization write:

Dafs

0 0 0 \rightarrow Reflection index

0. 0. 1. 0. 1. 0 $\rightarrow \epsilon_e, v_e$

0. 0. 1. 0.7071 0.7071 0. $\rightarrow \epsilon_s, v_s$

For circular polarization, it is complex:

Dafs

0 0 0 \rightarrow reflection index

0.7071 0.7071 0. 0. 0. 0. 0. 1. 0. $\rightarrow \epsilon_e, v_e (\epsilon_{xr} \epsilon_{xi} \epsilon_{yr} \epsilon_{yi} \epsilon_{zr} \epsilon_{zi} v_{ex} v_{ey} v_{ez})$

0.7071 -0.7071 0. 0. 0. 0. 0. 1. 0. $\rightarrow \epsilon_s, v_s$

When one does not want phase term between atoms, for example to simulate photoemission, just write 0 0 0 as reflection index.

Note that it is possible to choose any other origin for the azimuth, just using another vector than \vec{k} (or \vec{i}) to define the basis vector \vec{I} and \vec{J} . For this just write:

Zero_azim

0. 1. 1. \rightarrow vector in the direct crystal base

It is now possible to calculate the self absorption corresponding to the incoming and outgoing photon polarizations. For this put the keyword:

Self_absorption

Then one gets in the output files, after each reflection two new columns containing these data. The unit, like for XANES is the Megabarn. They contain after the convolution also the absorption coming from the other atoms and the other edges. That is that the absorption before the edge is not zero. The new data allows the correction due to the self absorption in order to compare with experimental spectra.

II-12) Spin polarized calculation

If one wants to make a spin polarized calculation put the keyword:

Magnetism

By default this calculation is done neglecting the spin-orbit coupling. If one wants to take into account this last put the keyword:

Spinorbite

The keyword "*magnetism*" is thus useless. If the polarized potential comes from LAPW, "*spinorbite*" or "*magnetisme*" must be before "*flapw*".

With the spin-orbit interaction, the calculation is automatically relativistic. To get a non relativistic calculation put:

Nonrelat

When the calculation does not use the flapw output, the use of the keyword "*atom*" is necessary to specify the electronic configuration, different for the spin up and spin down parts. The orbital occupancy must be given for each spin in doubling the corresponding columns.

Example of metal fcc nickel:

Magnetism

Atom

28 2 3 2 5. 4. 4 0 0.5 0.5

Crystal

3.52387 3.52387 3.52387 90. 90. 90.
 1 0.0 0.0 0.0
 1 0.5 0.5 0.0
 1 0.5 0.0 0.5
 1 0.0 0.5 0.5

The configuration is $3d^9 4s^1$ with 5 electrons 3d up and 4 electrons 3d down, 0.5 electron 4s up and 0.5 electron 4s down.

In case of an antiferromagnetic structure, one must put a minus sign in front of the atom type number to specify the atom with the reverse spin. For example in V_2O_3 monoclinic:

Atom

23 3 3 2 2. 0. 4 0 1. 1. 4 1 0.5 0.5
 8 2 2 0 1. 1. 2 1 2. 2.

Crystal

7.255 5.002 5.548 90.0 96.75 90.0 / a, b, c, alfa, beta, gamma
 1 0.34380 0.00080 0.29910 V8
 1 0.65620 0.99920 0.70090 V6
 1 0.84380 0.99920 0.29910 V3
 1 0.15620 0.00080 0.70090 V7
 -1 0.84380 0.50080 0.79910 V4
 -1 0.15620 0.49920 0.20090 V5
 -1 0.34380 0.49920 0.79910 V2
 -1 0.65620 0.50080 0.20090 V1

 2 0.40700 0.84500 0.65200 O 1 8f
 2 0.09300 0.84500 0.34800 O 2 8f
 2 0.59300 0.15500 0.34800 O 3 8f
 2 0.90700 0.15500 0.65200 O 4 8f
 2 0.90700 0.34500 0.15200 O 5 8f
 2 0.59300 0.34500 0.84800 O 6 8f
 2 0.09300 0.65500 0.84800 O 7 8f
 2 0.40700 0.65500 0.15200 O 8 8f
 2 0.25000 0.19100 0.00000 O 1 4e
 2 0.75000 0.80900 0.00000 O 2 4e
 2 0.75000 0.69100 0.50000 O 3 4e
 2 0.25000 0.30900 0.50000 O 4 4e

II-13) Relativistic calculation

To perform a relativistic calculation, put the keyword:

Relativism

II-14) Spin axis

By default the spin axis is along the c axis but in the trigonal unit mesh where it is along the c axis of the associated hexagonal unit mesh. This axis can be orientated along any (but uniform) direction:

Axe_spin

-0.08909 0. -0.15025 → in unit mesh

Another way to specify this axis is using the Euler angles. The orthogonal basis to do that is such that z is along c , but for trigonal system where z is along the hexagonal axis, x is along b x c and y is along z x x . One then has to write:

Ang_spin

45. 90. 0. → rotation around z , then around y , then around x . In this case spin axis is along (1,1,0) in the internal basis

Note that it is possible to define non collinear spin using atomic local basis. In this case one has to define the Euler local angles after the keyword `crystal` or `molecule`, in the line just before the corresponding atoms. See “non spherical atom”. Example:

Crystal

7.7400 7.7400 3.8400 90. 90. 90.

45. 90. ! spin axis along (1 1 0)

1 0.0 0.0 0.0

2 0.25 0.25 0.5

135. 90. ! spin axis along (-1 1 0)

1 0.5 0.0 0.0

2 0.75 0.25 0.5

315. 90. ! spin axis along (-1 -1 0)

1 0.0 0.5 0.0

2 0.25 0.75 0.5

225. 90. ! spin axis along (1 -1 0)

1 0.5 0.5 0.0

2 0.75 0.75 0.5

II-15) Self-consistent calculations

By default a calculation cycle is performed over the occupied states to determine the Fermi energy. The XANES calculation corresponds to the cycle 2, but this one is done with the initial potential. To avoid this Fermi (non SCF) energy determination, use the keyword :

No_Fermi

To perform a self-consistent calculation, use the keyword:

SCF

The calculation is then really self-consistent. The potential is calculated again at each cycle. The final XANES calculation uses this SCF potential. By default the convolution uses the calculated Fermi energy as cutting energy.

When using SCF calculation, the cluster radius for the SCF is the same than the one for the XANES. For a single evaluation in one cycle of the Fermi energy the radius is equal to the minimum between the XANES cluster radius and 3.5 Å. In both cases, this radius can be change using the keyword :

R_self

3. → Value of the radius for the SCF and/or Fermi energy evaluation.

With SCF calculation, the maximum number of cycle is fixed to 30. To modify this number use the keyword:

N_self

20

The electronic density of the cycle n+1 is interpolated between the electronic density of the two previous cycles. The initial weight of the cycle is 0.1. This weight decreases or increases according to the quality of the convergence between the initial value and one tenth of this one. To change this initial weight use:

P_self

0.05 → Initial weight.

The convergence is considered as realized when the variation of the total energy is less than $N_a \times \Delta E_{\text{conv}}$, where N_a is the number of atom and ΔE_{conv} a criteria fixed at 1 eV. To modify ΔE_{conv} put:

Delta_E_conv

2. → Value of ΔE_{conv} .

The SCF calculation is done with a non-excited cluster. The excited potential of the absorbing atom is taken into account only in the K and L_1 edges in the XANES final calculation by difference. If one wants to perform a SCF calculation with an excited cluster, use the keyword:

SCF_exc

II-16) Memory save

For low symmetrical material, it can be useful to save some memory space. This makes an approximation (but a good one) on the potential calculation. For this use the following keyword:

Memory_save

II-17) Hubbard correction

To include a Hubbard correction, use the keyword:

Hubbard

5. → value in eV of the Hubbard parameter (U-J).

This correction is applied to the “localized valence orbitals” of the chemical specie first in the list under “*crystal*” or “*molecule*” or under “*atom*” if used. Most often, it is convenient to perform also a self-consistent calculation. To apply different Hubbard corrections on different atoms, just put several numbers in the same line, following the order of the chemical species. Note that in principal this correction can be applied only on insulating compounds.

II-18) TDDFT calculations

TDDFT calculations can be performed using the keyword:

TDDFT

Then a first monoelectronic (LSDA) calculation is performed, followed by the TDDFT cycle. Then a correction is performed making a mixing between the edges. This can be useful for the L_{23} edges of the transition $3d$ elements. Calculations use a local Kernel with its Coulomb and exchange-correlation parts. To have only the Coulomb part, use the keyword:

RPA

III- Sophisticated keywords

III-1) Use of atomic electronic densities coming from external files

When using electronic densities coming from external files, these ones must be specified with the keyword “*atom*”. One eventually has to specify one atom more because the absorbing atom is a special chemical specie different of the non excited atoms. Its files must also contain the electron initial orbital before excitation (for instance the 1s one for a K edge). For the format of these files see chapter F. In the case there are two atoms:

<i>Atom</i>	→ keyword preceding the atomic electronic densities
<i>Fe_exc.txt</i>	→ name of the file containing the excited atom
<i>Fe.txt</i>	→ name of the file containing the type 1 atom
<i>Oxygen.txt</i>	→ name of the file containing the type 2 atom

One can also put one or several atoms calculated internally among the exterior files:

<i>Atom</i>	
<i>Fe_exc.txt</i>	→ name of the file containing the excited atom
<i>Fe.txt</i>	→ name of the file containing the type 1 atom
8 2 2 0 2. 2 1 4.	→ type 2 atom calculated internally

Once defined the atoms types, the molecule or mesh description uses the atom type number and not the atomic number. For example, for the cfc copper crystal:

<i>Atom</i>	
<i>c:\users\joly\dirac\cu\trdrslcu.e01</i>	→ excited copper file
<i>c:\users\joly\dirac\cu\trdrslcu.001</i>	→ standard copper file

<i>Crystal</i>	→ Crystal structure
3.610 3.610 3.610 90. 90. 90.	→ a, b, c, α , β , γ
1 0.0 0.0 0.0	→ Type number, position
1 0.5 0.5 0.0	
1 0.5 0.0 0.5	
1 0.0 0.5 0.5	

The first index is now the atom type number (but the excited one) in the list under “atom”.

III-2) Uses of atomic electronic densities of Clementi and Roetti

By default the program calculate the atomic basis using a Hartree-Fock-Dirac procedure. If you prefer the Clementi and Roetti basis use the keyword:

Clementi

Note that the Clementi basis exists only for $Z < 55$. When using this basis, the self-consistent calculations are not anymore possible as well as the automatic evaluation of the Fermi level (The program takes automatically the option « No_Fermi »).

III-3) Non spherical atomic electronic densities

When calculating using the finite difference method, it is possible to define non spherical atomic electronic densities. This is done under the keyword "crystal" or "molecule". For each non spherical atom one performs an expansion in spherical harmonics of these non spherical orbitals. In practice, on the line "atom type, position", one adds an integer giving the number of non spherical orbitals, then next lines the expansion in (l,m) of each of these orbitals with at the end the number of electron it contains. Optionnaly a rotation of the local atomic basis can be performed using the Euler angles (see keyword "*spin_axis*", for the Euler angle definition). In this case, the Euler angles must be given in the line in front of the atom. Example:

molecule

1.9	1.9	1.9	90.	90.	90.	→ a, b, c, α , β , γ
0.	45.	0.				→ 45° rotation around the Oy axis
1	.0	.0	.0	1		→ there is one non spherical orbital
	0.	0.	0.	0.	0.	→ (l,m) in the natural order, here (l=2,m=0) has 1
2	1.0	0.0	0.0			electron
2	-1.0	0.0	0.0			
2	0.0	1.0	0.0			
2	0.0	-1.0	0.0			

III-4) Orbital dilatation

It is possible to modify the valence orbitals defined above dilating or contracting them. This can be very useful for ionic material, for instance oxygen 2⁻, where the atomic bases are calculated for neutral atoms. For this purpose introduce the "*dilatorb*" then for each orbital one wants to dilate, the atomic type (number in the list "*atom*"), the valence orbital number and the expansion coefficient:

Dilatorb

```
3 1 0.3
3 2 0.3
```

III-5) Screening

By default, the screening is one electron on the first non full valence orbital of the absorber. If one wants to modify this value or the number of the orbital, put the keyword "*screening*" or "*ecrantage*" followed by the quantum number of the valence orbital and the value in number of electron (better inferior to 1 !) of the screening. In this case the cluster is not anymore necessary neutral.

Screening

```
3 2 0.2
```

→ n, l, and screening on the valence orbital

III-6) Mesh or molecule charge

The unit mesh must be neutral. A molecule is also often neutral. A test is performed in the program to verify this neutrality. If one wants to omit it, put:

chlibre

III-7) Exchange-correlation potential

By default the calculations are done using the real Hedin, Lundqvist and Von Barth potential. If one wants to use the Xalpha potential, one must introduce the keyword *Xalpha* followed by the value of the corresponding parameter:

Xalpha

0.3333

→ value of the Xalpha parameter

Remark : the value of the Xalpha parameter the closest to the other models is 2/3.

In the case of using the exchange-correlation potential coming from *flapw*, if one wants to keep this potential independent of the energy put the keyword "*xalpha*" with any value beneath. The potential will not be Xalpha but the one calculated by *flapw*.

If one prefers to use the Perdew and Wang's potential, put the keyword:

Perdew

This potential will be used with the parametrization of Moruzzi, Janak and Williams but keeping the energy dependency proposed by Hedin and Lundqvist.

III-8) Reference of the photoelectron wave vector

By default the reference of the electronic wave vector (and so of the kinetic energy outside the sphere of calculation and the muffin-tin ground in case of muffin-tin calculation) is taken as the average of the potential between the absorbing atom and the first crown when calculating in green and at the outer sphere frontier in FDM. It is possible to impose this ground potential at the Fermi energy. The dependence versus the kinetic photoelectron energy is then added automatically to this term. Put then the keyword:

V0imp

-11.5

→ value of the ground potential at the Fermi energy.

III-9) Maximum potential

For a calculation performed on a molecule, the potential increases when going away from the atoms. When this molecule is not in the gaz phase, but in solution or in any relatively dense surrounding, this incese is artificial. In case of calculation under the finite difference mode, this reach to faulse bounded states very thin in energy. It can be usefull to give a

maximum value to this potential to avoid this phenomena with the use of the keyword "*vmax*", followed by the value of this potential :

Vmax

-6.

III-10) Complex energy

It is possible to use a complex energy when working in the multiple scattering mode (green). For this, one has to specify the imaginary part (positive) of this energy in a table versus the photoelectron energy, under the keyword "*eimag*":

Eimag

0. 0.5

10. 0.7

30. 3.

50. 5.

100. 6.

When a uniform broadening is sufficient, it is not necessary to specify the photoelectron energy:

Eimag

0.1 → value of the uniform width (eV)

The use of a small width (0.1 eV) is sometimes useful for the calculations at low kinetic energy of the photoelectron because the localized level (3d or 4f) can be too thin in energy to be correctly evaluated. It is even truer for photoemission calculation (see keyword "*photoemission*"). If this broadening is small it has no effect on the forthcoming convolution to take into account the widths of the hole and final states.

III-11) Radius of the cluster for the superposition of the potential

The potential inside the sphere of calculation with a radius R_s set under the keyword "*radius*", is calculated by superposition. To avoid frontier problems, the atoms taken into account for the superposition are all the atoms inside the calculation sphere plus an outer shell. By default this outer shell is 2.5 Å thick, giving thus a new sphere with a $R_s + 2.5$ radius. If one wants a bigger radius (useful for the oxides) put the keyword:

Rpotmax

15. → radius of the cluster for superposition in Angström

III-12) l+1 approximation for the selection rule

If one wants to make the l+1 approximation where only the transition $\Delta l = +1$ is authorized (so one neglects $\Delta l = -1$, for example for the threshold L_{II} and L_{III} , the transitions are only toward the d states), put the keyword:

lplus1

In the same way if you want only the l-1 states put:

lminus1

III-13) Rydberg series

One can add an outer sphere having a $-1/r$ potential to analyze Rydberg series. In this case put the keyword "*Rydberg*":

Rydberg

III-14) Cluster origin

By default, the cluster origin is set on the absorbing atom. If one want that this origin is as in the cluster indata put the keyword:

Noncentre

If one wants to impose a specific center put:

Centre

0. 0.25 1. → coordinates in mesh parameter unit of the center

III-15) Cut of the potential

To get a cut of the potential in the output, along a line or a plan put:

Trace

0 1. 0. 0. 0. 0. 0. → l,a,b,c,p,q,r
 l = 1, cut along a line with vector (a,b,c) crossing the point (p,q,r) in unit mesh parameter
 l = 2, cut along the plane $ax + by + cz = p$
 l = 3, gives all the points

III-16) Density of state

To get the state density and its integral for any harmonics projected on the central atom put the keyword:

Density

There will get one output file more with the suffix *_sd0.txt*. Note that only the states members of the representations useful to get the asked spectra are calculated. Consequently, such produced density of state can be uncomplete. To get a complete projection for all the atoms put the keyword "*state_all*" in place of density. Calculation is then performed on all the representations (what can be longer). Then one gets n new output files with the extensions *_sdi.txt*, "*i*" being the atom number. In these files stand in column, first the integral of the total atomic electron density, then the density and its integral of each (ℓ, m) followed by the sum over m, that is the density and its integral for each ℓ . For magnetic calculation, the expansion is split in its "up" and "down" components.

Usually for a transition metal, at the end of the *d* band, the integral must be close to 10 (because there are 10 *d* electrons!). At the end of the rather unlocalized *s* and *p* bands, the integrals rarely reach 2 or 6, because the electrons are counted only inside the atomic sphere of radius R_{mstd} (see in the *bav* file for its value). These orbitals having a rather big radius they are not all inside such atomic sphere. Calculation being performed in the continuum one also has to recall that *n* is not anymore a "good" quantum number. Consequently the integral continue to increase indefinitely with energy. At the end of a *d* (or *f*) band, rather localized, one nevertheless reaches more or less at 10 (or 14). In this case one can find the Fermi level energy just looking the energy where the integral reach the supposed number of electron in the corresponding level of the atom. Sometimes the integral never reach 10 (or 14) or goes far higher. One of the reasons can be that the starting energy is too high, one thus loses the beginning of the band, another reason can be the energy grid is not sufficiently thin. One has, in this case to take a path smaller (down to 0.01 eV or even less for 4*f* elements). In order to avoid such a thin grid, it is also possible to broaden a bit these localized states using the keyword "*Eimag*" (this works only in "Green" mode).

III-17) Spherical tensors

To get the spherical tensors (in number of electron) put the keyword:

Spherical

One gets new files with the extension *_sph_atom1.txt* and *_sph_xtal.txt* for the atom and crystal. If one uses such file as indata for the convolution process one gets the f' and f'' of each tensor component of the atom. Another file *_sph_int_atom1.txt* contains the integral of the spherical tensors.

To get the contributions of the tensor components on each polarization and reflections, put in place of "*spherical*" the keyword:

Sphere_all

Then one gets a series of output files with the extensions *_sph_signal_xan.txt*, *_sph_signal_pol1.txt*, *_sph_signal_rxs1.txt* for the contributions on each polarization and reflections. The number after *pol* or *rxs* is the number of the polarization or reflection.

III-18) Calculation area boundary

By default in FDM, the meshing is performed in a sphere extending up to the last atom inside the sphere of radius given under "radius" plus the atomic radius (by default 0.65 Å) plus one interpoint distance (0.2 Å by default). In order to use a bigger sphere put

Overad

1.2 → distance over the last atom + its radius to take into account.

III-19) Displacement of the absorbing atom

To move the absorbing atom in reference to its position given under "molecule" or "crystal" put:

Dpos

0.2 0.0 0.0 → displacement vector in Angström

III-20) Getting back the tensor components from a previous calculation

It is possible to get back the tensor component from a previous calculation (is in the *fdmnes_out_bav.txt* type file) and to compute other polarizations putting the key word *extract* followed by the corresponding name:

Extract

c:\users\joly\xanout\calcul_prec_bav.txt

In case of calculations for different absorber position whose number is the number of output files, one has to extract the tensors from these n files if there are n non equivalent absorbing atoms. The use of the keyword "absorber" is then absolutely necessary. The list of the files under "extract" must correspond in number and in order to the list of atoms under "absorber":

Extract

c:\users\joly\xanout\calcul_prec_1_bav.txt

c:\users\joly\xanout\calcul_prec_2_bav.txt

Absorber

1 2

It is possible to make rotations of the local repair. It can be useful when using tensor calculated in a different base. It is possible de make 1, 2 or 3 rotations around the axes Oz, then Ox, then Oy. Each rotation defines a new local base, starting point of the next rotation. For this write:

Rotsup

30. 0. 45. → 30° rotation around Oz, 0° around Ox then 45° around Oy.

For the same purpose one can apply symmetry operation on the extracted tensors. They are codified as explained in the manual with the “*symsite*” keyword. They must be given in the same order than the extracted files. These codes are given after the keyword *extractsym*:

extract

xanout/fe3o4/w_g_ch015_3d55_rs80_1_bav.txt
 xanout/fe3o4/w_g_ch015_3d55_rs80_1_bav.txt

Extractsym

1 25 → code 1 for identity, 25 for inversion.

III-21) Energy shift of the spectra

If one has gotten a reference for the initial orbital, it is possible to give it under the keyword “*epsii*”. This will produce a shift of the output spectra equal to the difference between this energy and the energy calculated in the program. It is safer to perform this operation with the shift parameters during the convolution step.

Epsii

6253.1 → positive value in eV.

III-22) Expansion in path

In the multiple scattering mode it is possible to avoid the matrix inversion, but to make an expansion in path. One has to give the number of way:

Chemin

3 → number of path

III-23) Cartesian tensors

To get the atomic cartesian tensors put the keyword:

Cartesian

One gets a new output file with the extension _car.txt. Then one can use other keywords that make that from this output file will be calculated f' and f'' for any tensor component. For this purpose put the keyword “*fprim*” in the indata file.

III-24) Spectra by atom

If one wants to get the signal coming from each atom one has to add the keyword:

Allsite

One then gets, at most of the usual output files, new output files of the type atom1.txt, atom2.txt, etc... The suffixes _atom1, _atom2 correspond to the number of the atom.

III-25) Use of densities and potential coming from FLAPW

If the potential is imported from a FLAPW calculation, one has to introduce the keyword "*flapw*" in place of "*atom*". The keyword "*crystal*" or "*molecule*" becomes also unnecessary. Then must stand the names of the 5 FLAPW output files in case of spin unpolarized calculation and the 7 files in case of spin polarized calculation:

Flapw	→ names of the output FLAPW files
<i>tio2.struct</i>	structure and symmetry
<i>tio2.vcoul</i>	coulombian potential
<i>tio2.r2v</i>	exchange-correlation potential
<i>tio2.clmsum</i>	electronic density
<i>tio2.ti1s</i>	initial wave function

In case of polarized calculation the keyword "*magnetism*" must be before the keyword "*flapw*":

Magnetism

Flapw	
<i>tio2.struct</i>	structure and symmetry
<i>tio2.vcoul</i>	coulombian potential
<i>tio2.r2v</i>	exchange-correlation potential spin up and spin down
<i>tio2.clmsum</i>	total electronic density
<i>tio2.clmup</i>	valence electron electronic density, spin up
<i>tio2.clmdn</i>	valence electron electronic density, spin down
<i>tio2.ti1s</i>	initial wave function

The last file contains the wave function of the initial core orbital. If one is ok with the initial wave function calculated internally, one can avoid it. One must in this case substitute the keyword "*flapw*" by "*flapw_psi*".

If one wants to have an energy dependant exchange-correlation potential, one must add the keyword "*hedin*".

By default, the absorbing atom is the first one in the "*struct*" file list. If one wants that it is the n^{th} , put the keyword:

absorber
n

III-26 Jump in the indata file

It is possible to jump over a part of the information written in the indata file with the keywords "*Jump*" and "*Endjump*". All what is between them is not red. For example:

Jump*Quadrupole**Spinorbite****Endjump*** → the keywords "*Quadrupole*" and "*Spinorbite*" are not considered.**III-27 Comment**

It is possible to introduce a line of comment which will be copied in the output files.
For this purpose write:

Comment

Iron K-edge in magnetite

→ line of comment

IV-28 Atomic spectra

To get in last column the atomic absorption spectra (without the neighbour atoms), put the keyword:

Xan_atom**IV-29 Non resonant magnetic scattering**

The non resonant magnetic scattering is taken into account for the RXS. This term can be decreased, or put to zero, using a multiplicative factor:

No_res_mag

0.8

→ factor

An additional factor can be applied to the contribution from the orbital moment. This factor corresponds to L/2S (and not L/S):

No_res_mom

-0.3

→ factor for the orbital moment

By default, this factor is calculated using the Hund rules and multiplied by 0.2.

IV- Technical keyword

Here stand the technical keywords necessary for specific tests and optimization of the code, for expert users.

IV-1) Details on the calculations

To get details on the calculations done in every routine, put the keyword:

Check

3 3 3 3 1 1 1 1	→ Main, Lectur, Prepar, Agregat, Etafin, Distan, Reseau, Laplac,
1 1 1 1 1 1 1 1	→ Bord, Potato, Orbval, Poisso, Pot0, Ylmsrt, Ylmatm, Potex,
1 1 1 1 1	→ Phisort, Sphere, Newind, Mat+MSM, Coabs

Values can go from 0 to 4 giving more and more for each routine. By default, there is 1 for all the routines. One can also write:

<i>check_all</i>	→ equivalent to icheck = 3 for all the subroutines
<i>no_check</i>	→ equivalent to icheck = 0 for all the subroutines
<i>check_pot</i>	→ equivalent to icheck = 3 for all the subroutines concerning the potential
<i>check_mat</i>	→ equivalent to icheck = 3 for the matrices MSM or FDM
<i>check_sph</i>	→ equivalent to icheck = 3 for the subroutine "sphere"
<i>check_coabs</i>	→ equivalent to icheck = 3 for the subroutine "coabs"

IV-2) Symmetry

The point symmetry is calculated automatically. Anyway the expert users can impose it by the keyword:

Sym

2/m → Schoenflies coefficient or international table of crystallography coefficient. Be careful that these symmetries are possible only for the mesh axis defined in the indata.

IV-3) Tensor basis

By default the atomic tensors given in the output files are expressed in the orthonormal basis R_1 where z is the crystal c axis (but for the trigonal structure where z is the c axis of the associated hexagonal mesh), x is along $b \times c$ and y along $z \times x$. Nevertheless it is possible to express it in the basis where it is in fact calculated, the so-called basis R_2 . From R_1 , another rotation is performed for magnetic calculation when the spin axis is not along z . Then a another $\pm 45^\circ$ rotation is sometimes performed when a 2 fold axis or a symmetry plane is found along a diagonal. For this purpose put the keyword

Base_spin

IV-4) Calculation basis

By default the connection to the continuum is performed in real base (neuman, bessell). To use a complex base (bessel, hankel) put:

Basecomp

IV-5) Equivalent atoms

The program automatically calculates the total signal resulting from the equivalent atoms by symmetry. It is nevertheless possible, for the expert *fdmnes* user, to impose the symmetry relation between the atoms and thus to impose this summation. One must then use the keyword "*symsite*". Following this keyword, stands the number of non equivalent atoms (or number of group of atoms), then for each of them, the number of equivalent atoms followed the list of the relative symmetry to the first atom of the list and the atomic position in mesh unit. The symmetry are codified by number going from 1 to 64, see the list given bellow.

Symsite

```

3          ! Number of non equivalent atoms (or group of atoms)
4          ! Number of equivalent atoms, group 1
 1 0.2500 0.2500 0.2500
24 0.7500 0.7500 0.2500
23 0.7500 0.2500 0.7500
22 0.2500 0.7500 0.7500
4          ! Number of equivalent atoms, group 2
 1 0.0000 0.0000 0.0000
22 0.0000 0.5000 0.5000
23 0.5000 0.0000 0.5000
24 0.5000 0.5000 0.0000
4          ! Number of equivalent atoms, group 3
 1 0.5000 0.5000 0.5000
23 0.0000 0.5000 0.0000
22 0.5000 0.0000 0.0000
24 0.0000 0.0000 0.5000

```

The negative indexes correspond to the same symmetry plus time reversal (in case of magnetism).

Symmetry code:

1: identity	8: rot $2\pi/3$ around (1,1,-1)
	9: rot $4\pi/3$ around (1,1,-1)
2: rot $2\pi/3$ around (1,1,1)	
3: rot $4\pi/3$ around (1,1,1)	10: rot $2\pi/2$ around (1,1,0)
4: rot $2\pi/3$ around (1,-1,1)	11: rot $2\pi/2$ around (-1,1,0)
5: rot $4\pi/3$ around (1,-1,1)	12: rot $2\pi/2$ around (1,0,1)
6: rot $2\pi/3$ around (-1,1,1)	13: rot $2\pi/2$ around (-1,0,1)
7: rot $4\pi/3$ around (-1,1,1)	14: rot $2\pi/2$ around (0,1,1)

- 15: rot $2\pi/2$ around (0,-1,1)
 16: rot $2\pi/4$ around 0x
 17: rot $2\pi/4$ around 0y
 18: rot $2\pi/4$ around 0z
 19: rot $-2\pi/4$ around 0x
 20: rot $-2\pi/4$ around 0y
 21: rot $-2\pi/4$ around 0z

 22: rot $2\pi/2$ around 0x
 23: rot $2\pi/2$ around 0y
 24: rot $2\pi/2$ around 0z

 25: inversion

 26: rot $2\pi/4$ around 0x and inversion
 27: rot $2\pi/4$ around 0y and inversion
 28: rot $2\pi/4$ around 0z and inversion
 29: rot $-2\pi/4$ around 0x and inversion
 30: rot $-2\pi/4$ around 0y and inversion
 31: rot $-2\pi/4$ around 0z and inversion

 32: rot $2\pi/3$ around (1,1,1) and inversion
 33: rot $4\pi/3$ around (1,1,1) and inversion
 34: rot $2\pi/3$ around (1,-1,1) and inversion
 35: rot $4\pi/3$ around (1,-1,1) and inversion
 36: rot $2\pi/3$ around (-1,1,1) and inversion
 37: rot $4\pi/3$ around (-1,1,1) and inversion
 38: rot $2\pi/3$ around (1,1,-1) and inversion

 39: rot $4\pi/3$ around (1,1,-1) and inversion
 40: plane perpendicular with 0x
 41: plane perpendicular with 0y
 42: plane perpendicular with 0z

 43: diagonal plane $y = z$ containing 0x
 44: diagonal plane $x = z$ containing 0y
 45: diagonal plane $x = y$ containing 0z
 46: diagonal plane $y = -z$ containing 0x
 47: diagonal plane $x = -z$ containing 0y
 48: diagonal plane $x = -y$ containing 0z

 49: rot $2\pi/3$ around 0z
 50: rot $4\pi/3$ around 0z
 51: rot $2\pi/6$ around 0z
 52: rot $10\pi/6$ around 0z
 53: rot $2\pi/3$ around 0z, negative axe
 54: rot $4\pi/3$ around 0z, negative axe
 55: rot $2\pi/6$ around 0z, negative axe
 56: rot $10\pi/6$ around 0z, negative axe

 57: plane at 30° containing 0z
 58: rot $2\pi/2$ around axe at 30° perp. 0z
 59: plane at 60° containing 0z
 60: rot $2\pi/2$ around axe at 60° perp. 0z
 61: plane at 120° containing 0z
 62: rot $2\pi/2$ around axe at 120° perp. 0z
 63: plane at 150° containing 0z
 64: rot $2\pi/2$ around axe at 150° perp. 0z

IV-6) Orientations

It is possible to restrict the calculation to the tensor components which are really useful. The keywords "*ldipimp*" and "*lquaimp*" are used for this purpose respectively for the dipolar and quadrupolar components:

ldipimp

1 0 0 will calculate only the 0x component

lquaimp

0 0 1 will calculate only the xz component
 0 0 0
 0 0 0

IV-7) Bounded states

If one wants to modify the kinetic energy at are calculated the bounded states (beneath the average potential), put the keyword "*etatlie*" followed by the kinetic energy in eV:

etatlie
0.05

IV-8) Inversion of the matrix in the multiple scattering mode

By default one inverts the matrix $(\kappa^{-1} + i + G)$ where κ corresponds to the neuman-bessel normalization. To avoid eventual convergence problems (not seen yet), one can prefer to calculate $(1 + G\tau)^{-1}\tau$. For this use the keyword:

Normaltau

IV-9) Muffin-tin potential

If one wants to use the muffin-tin approximation in FDM put the keyword:

muffintin

IV-10) Non excited absorbing atom

To perform a calculation with a non excited absorbing atom, that is with an absorbing atom without a hole in its core level and without screening put:

Nonexc

IV-11) Modification of the grid of point parameters

By default the finite difference calculation is performed at order 4 with an inter point distance equal to 0.25 Å, the radius of the spherically symmetric area is around 0.65 Å (less for the light elements). To modify them write:

Rmt → to modify the muffin-tin radius
0.65

Adimp → to modify the interpoint distance
0.20

Iord → To modify the order of the Taylor expansion.
2

IV-12) Expansion in spherical harmonics

An expansion in spherical harmonics is performed in the atoms both in the multiple scattering mode and in the finite difference mode, but with a smaller radius. The maximum value of ℓ is obtained from the formula $kr = \sqrt{\ell_{\max}(\ell_{\max} + 1)}$, where k is the photo electron wave vector and r the muffin-tin radius (Rmtg in the bav file). By default, to this value one adds 1. To fix a value of ℓ_{\max} independent of energy just write:

lmax \rightarrow The ℓ_{\max} value is fixed at 3
3

If one wants to keep the energy dependant formula but adding 1, 2 or more to this formula, put a negative sign in from of 1:

lmax \rightarrow The ℓ_{\max} value is now given by the formula plus 2.
-2

When the energy is increasing, ℓ_{\max} increases. By default the maximum value of ℓ_{\max} is set at 2 for $Z = 1$, at 3 for $Z = 2$ then at 4 for $Z \leq 18$, then at 5 for $Z \leq 36$, then at 6 for $Z \leq 54$, then at 7 for $Z \leq 86$ and 8 for Z over. To avoid this limitation put the keyword:

lmaxfree

When working using the finite difference mode, the connection to the outer sphere also needs an expansion in spherical waves. This one uses the same formula that for the atom but r is now the radius of the outer sphere. To modify this ℓ_{\max} one proceeds exactly in the same way with positive value to fix an energy independent value, and negative value to keep the formula but adding to the formula. By default the additional value is 5. The key word is now *lmaxso*:

lmaxso
20

IV-13) Muffin-tin radius

By default the muffin-tin radius is calculated to have a jump of potential the smallest as possible and closed between the atoms. One can modify it by the use of 3 different keywords.

If one wants that this radius is calculated using the Norman procedure introduce the keyword:

Norman

If one wants than the muffin-tin radius is the half value of the interatomic distance, introduce the keyword:

Raydem

By default there is a 10% overlap between the spheres. If one wants modify this put the keyword "*overlap*" followed by the value of the overlap:

Overlap

0.15 → 15% overlap

If one wants that the muffin-tin radius be determined by the value it gets when the atomic potential is equal to the interstitial potential, put the keyword *rmtv0*, followed by the value of this potential (takes place the one define by the keyword *v0imp*):

Rmtv0

-12.

If one wants to impose the radius for each chemical species put *rmtimp* followed by the radius in Angström for each of them.

Rmtimp

0.65

1.1

The program gives in the output file the atomic charge integrated up to the muffin-tin radius. If one wants to get this value for a different radius put:

Rchimp

→ corresponding keyword

1.2

→ radius value for each type of atom

0.7

To limit potential jump between the interstitial area and the muffin-tin sphere, a linear interpolation on the potential is performed in the last 10% of the radius. For this put:

Supermuf**IV-14 Number format**

In the output files, the numbers are written with 15 characters with 7 after the point. To change this put:

Length_word

16

→ number of characters (between 11 and 17)

IV-15 Line length

Lines are red over a specific length. When reading files containing a large number of columns (for example calculation of more than 100 RXS spectra), the default line length (3020 characters) can be not sufficient. To modify this put:

Length_line

4000

→ maximum number of character in one line

IV-16 Résolution in initial state

The outputs are resolved in initial states only for magnetic calculation. For a non magnetic calculation, it is possible to get anyway the output resolved be initial state using the keyword:

Core_resolved

It is then possible in the step convolution to select one of these states.

C- Convolution

The *fdmnes* program allows performing:

1) the convolution by a lorentzian of absorption spectra eliminating the occupied states. For the anomalous diffraction, it makes the integration over energy of the unoccupied states then calculates the intensity of the diffracted peaks.

2) a weighted summation over different outputs of the *fdmnes* program. This summation can include a relative shift between the spectra.

This step of the calculation can be done together with the previous one, just adding the corresponding keywords. It can also be done independently, writing another input file contains only the keywords related to the convolution. Here comes an example:

Calculation	→ To give the file name resulting from the previous step.
<i>g_rs43.txt</i>	→ name of the file
Conv_out	→ To give the file name of the convoluted spectra
<i>g_rs43_conv.txt</i>	→ name of the file
Convolution	→ keyword to specify an “arctangeant” shape for the broadening”
Efermi	→ keyword to specify the Fermi energy (E_F) relative to the calculation
<i>-5.5</i>	

Keywords for the convolution

a) File names

The indata files for the convolution step are the output files of the previous step whose names are defined under "*filout*". When the convolution is performed together with the main calculation, it is not necessary to specify again this name. In the other cases one has to introduce it or them with the keyword "*Calculation*":

Calculation	→ To give the file names resulting from the previous step.
<i>g_rs43.txt</i>	→ name of the file

Another example with 2 indata files and with different shifts:

Calculation	→ To give the file names resulting from the previous step.
<i>Fe_rs64_01.txt</i>	→ name of the file 1
<i>1.0 0.2</i>	→ weight, shift
<i>Fe_rs64_02.txt</i>	→ name of the file 2
<i>1.0 -0.2</i>	→ weight, shift

The weight is applied on the XANES and the anomalous scattered amplitudes. For the RXS calculations, the output files, before convolution contain in second line the Thomson

factors plus the anomalous part of the non resonant atoms. The weighting applied on these terms is the weight divided by the sum over the files of the different weights. In this way is taken into account the average configuration for substitutions of atoms of different chemical species on the same sites, but when it is the resonant atom, one has not an occupancy rate equal to one. In this last case one also has to use the keyword "Thomson".

The name of the convolution output file is by default the input file name with the added suffix "_conv.txt". Anyway it is possible to impose another name with the keyword:

Conv_out

Fe_rs64_sum_conv.txt → name of the convoluted spectra file

If there is an azimuthal scan in anomalous diffraction, one has also to give the corresponding indata file names (type *_scan.txt*) and the new output file. This is set after the keyword "scan":

Scan

Fe_rs64_01_scan.txt → name of the indata dafs file 1

Fe_rs64_02_scan.txt → name of the indata dafs file 2

Scan_conv

Fe_rs64_scan_conv.txt → name of the new output file with the convoluted scan

To specify a working directory put the keyword directory followed by the directory name with at the end the separator (/ or \):

Directory

C:/Documents and Settings/joly/Mes documents/xanes/xanout/v2o3/

b) Fermi level

The states beneath the Fermi level are occupied. Thus the cross section is set to zero, before convolution. By default, when in the previous step the *No_Fermi* keyword was used, this level is set to -5 eV. Nevertheless most often, this value must be specified with the keyword:

Efermi → keyword to specify the Fermi energy (E_F) relative to the calculation
-5.5

The Fermi level is applied before the energy shift because in principle it is used to simulate the core level shift. Anyway it is possible to apply it after the shift adding the keyword:

Dec

It is also possible to have different values of the Fermi level for the different files. This is written in third column after each file name:

Calculation*Fe_rs64_01.txt*

1.0 0.2 -5.

Fe_rs64_02.txt

1.0 -0.2 -5.9

→ To give the file names resulting from the previous step.

→ name of the file 1

→ weight, shift, Fermi level

→ name of the file 2

→ weight, shift, Fermi level

The Fermi energy written under "*Efermi*" when specified, is then not considered.

c) Convolution width

The convolution to apply depends on the core level width (Γ_{Hole}) and the final state width. Γ_{Hole} is automatically determined. Nevertheless, it is possible to modify it with the keyword:

Gamma_hole

2.05

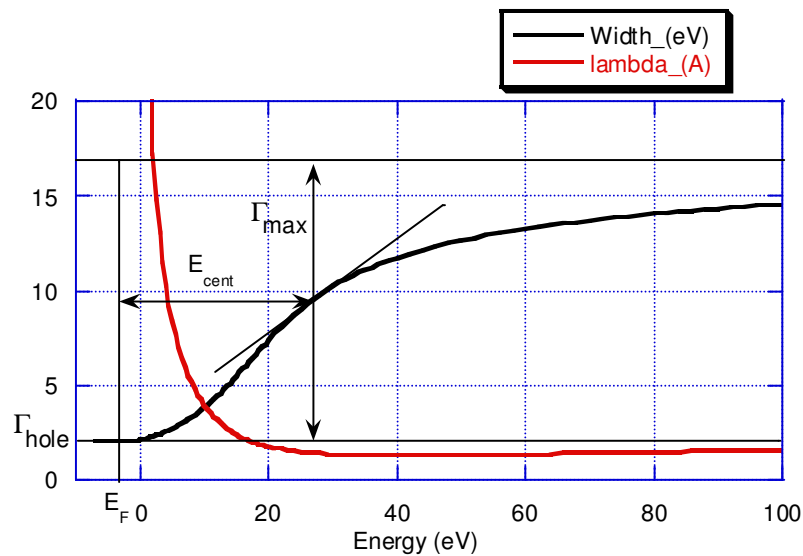
→ Core level width (Γ_{Hole}) in eV.

To have different values of Γ_{Hole} for a double edge like L_{23} , just put 2 numbers.

There are different ways to calculate the energy dependant broadening. The first one uses an arctangent formula:

$$\Gamma = \Gamma_{\text{Hole}} + \Gamma_m \left(\frac{1}{2} + \frac{1}{\pi} \arctan \left(\frac{\pi}{3} \frac{\Gamma_m}{E_{\text{Larg}}} \left(e - \frac{1}{e^2} \right) \right) \right)$$

with: $e = \frac{E - E_F}{E_{\text{cent}}}$ where Γ_m , E_{cent} and E_{larg} are respectively the maximum width (at high energy) of the final state, the center and the width of the arctangent function. The depth at the center of the arctangent is Γ_m/E_{larg} . A typical curve is given bellow:



Such a convolution is introduced with the keyword:

Convolution

Default values for E_{larg} , E_{cent} and Γ_m are respectively: 30, 30 and 15 eV. It is possible to modify them with the keywords:

Ecent

30. $\rightarrow E_{\text{cent}}$

Elarg

30. $\rightarrow E_{\text{larg}}$

Gamma_max

20. $\rightarrow \Gamma_m$

In the convolution, along the integration it is the width of the running energy which is taken. It is possible to use the width of the final state energy corresponding to the energy of the elastic photon. One then makes the integration with a constant width. This procedure seems improve the agreement with experiment especially in the pre-edge range where the other procedure increase the background. The author does not understand why. To do this use the keyword :

Gamma_fix

It is also possible to use the Seah-Dench formula for the calculation of the broadening. In this case one gets:

$$\Gamma = \frac{A\Gamma_m E_p}{\Gamma_m + AE_p} + \Gamma_{\text{Hole}}, \quad \lambda = \frac{1}{A\sqrt{E_p}} + \frac{\sqrt{E_p}}{\Gamma_m}, \quad \text{with: } E_p = E - E_F$$

This is performed with the keyword:

Seah

1. 20. $\rightarrow A, \Gamma_m$

One can also use a simple table with the keyword "*table*":

Table

0.0 1. \rightarrow Energy ($E-E_F$) and broadening $\Gamma(E-E_F)$
 10.0 1.5
 20.0 2.
 30.0 3.5
 100.0 5.

When the keyword "*Efermi*" is used without the keywords "*Convolution*" or "*Seah*", the width of the convolution is constant and equal to the width of the core hole. In all cases, the curve shape can be obtained in a specific output file named *gamma_conv.txt* when using the keyword:

Check_conv

To simulate the experimental resolution, it is also possible to convolute by a Gaussian. This convolution is performed after the lorentzian. For this write:

Gaussian → keyword
 1. 0.05 → width of the gaussian

This same convolution can be used as a first approximation to simulate the thermal vibration effects. This is valid only in the extended part of the spectra. In this case one applies the rule “ $kr = \text{constant}$ ” and the convolution width becomes proportional to the energy. The parameter to write is the relative vibration amplitude in relation to the interatomic distance. This is done, writing a second number under “*Gaussian*”:

Gaussian → keyword
 1. → width of the Gaussian, relative amplitude of the vibration

d) Other keywords

To not take into account the Thomson factor f_0 (as for a forbidden beam) put:

Forbidden

To get output spectra starting at a lower energy put:

Estart
 -8. → Value of the starting energy (eV)

When there are several indata files, the program uses only the intersection of the energy ranges. To use the total range put the keyword:

Nocut

To get in the output file, the f' and f'' values, put the keyword:

Fprime

In the output file for each reflection, there are three columns, f' , f'' then intensity. The intensity is for the mesh, f' and f'' are for one atom of the mesh. f' contains also f_0 . To get only f' , put also the keyword “*forbidden*”.

To get the isolated atomic f' and f'' :

Fprime_atom

The values of the atomic f' and f'' are then in a new output file with the extension `_fprime_ato.txt`

It can be useful to impose specific value for each RXS reflections for the non resonant (Thomson) structure factors and the resonant one of the other atoms. That is:

$$\sum_a p_a (f_{0a} + f'_a + if''_a) e^{i\vec{Q} \cdot \vec{R}_a}, \text{ with } f'_a \text{ and } f''_a \text{ considered only for the other chemical species}$$

and where p_a is the site a occupancy rate. This is the case for example for occupancy rate non equal to one. These complex terms taken as a constant in all the energy range can be introduced through the keyword:

Thomson

1.1670478E+02 1.0583769E+02 1.1477827E+02 -8.2670689E+01

In the second line stand these terms, here for 2 reflections. They replace the values calculated by the program and placed in the second line of the output file.

Photoemission

With the keyword "*photoemission*", it is the photoemission spectra which is calculated. In such calculation the cutting is simply above the Fermi level and not beneath it as in xanes. The convolution broadening is the independent of energy and is fixed by default to the hole width. This with can nevertheless been modified by the use of the keyword "*Gamma_hole*". Note that would have interest having previously performed a calculation in "green" mode with a minimum energy width (keyword "*eimag*") at least 0.1 eV, all this within an energy range convenient for photoemission.

Damping due to multielectronic effects

In a simple approach, the multielectronic phenomena can be seen as a simple damping in the absorption cross section and in the anomalous scattering amplitude in diffraction. This damping not taken into account, by default, can be introduced using the keyword "*S0_2*". It is typically around 0.8.

S0_2

0.8 → value of the damping

When setting $S0_2 = 0$, in diffraction, one keeps only the non resonant term.

Sélection of the transition from a peculiar initial state

The fdmnes output files, before convolution, contain the spectra by transition from all initial states when the calculations are magnetic and by threshold in the other case. It is possible to perform the convolution for only one of this component (avoiding thus their summation) with the keyword "*Selec_core*". One has then to write below the number of the selected initial state:

Selec_core

I → Number of the selected initial state. For a K edge resolved in spin, this correspond to the $1s_{1/2}$, that is to a pure spin down state, the number 2 corresponds to spin up.

D- Parameter optimization

It is possible to compare the calculated spectra to the experimental ones with the help of metric distances and R factor. The metric distances to compare the individual theoretical ($f_{th}^{(i)}$) and experimental ($f_{exp}^{(i)}$) spectra are given by:

$$D_1^{(i)} = \frac{1}{2} \int \left| \frac{1}{c_{th}^i} f_{th}^{(i)}(e) - \frac{1}{c_{exp}^i} f_{exp}^{(i)}(e) \right| de,$$

$$D_2^{(i)} = \frac{1}{2} \int \left| \frac{1}{C_{th}^i} F_{th}^{(i)}(e) - \frac{1}{C_{exp}^i} F_{exp}^{(i)}(e) \right| de,$$

with each time the normalization factors: $c = \int_{E_{min}}^{E_{max}} f(e) de$

The F are the integrals of f . The R factor is a conventional one (see E. Zanazzi and F. Jona, Surf. Sci. 62, 61 (1977)) given by:

$$R_X^{(i)} = \frac{13}{6} \frac{e \sum \left| c_i f_{th}^{(i)}(e) - f_{exp}^{(i)}(e) \right|^2}{\sum_e \left| f_{exp}^{(i)}(e) \right|^2}, \quad \text{with } c_i \text{ such that: } \frac{\partial R_X^{(i)}}{\partial c_i} = 0$$

Then for n different spectra, one gets the total metric distance or R factor by:

$$D_{1,2} = \sum_{i=1,n} p_i D_{1,2}^{(i)} \quad \text{and} \quad R_X = \frac{6}{13} \left(\frac{3}{2n} + \frac{2}{3} \right) \sum_{i=1,n} p_i R_X^{(i)}$$

where p_i is the relative weight for the spectra "i" given by $p_i = \frac{E_{max}^{(i)} - E_{min}^{(i)}}{\sum_{j=1,n} E_{max}^{(j)} - E_{min}^{(j)}}$

It is possible to vary some parameters in order to optimize the agreement between calculation and experiment. Then results are given for a multi-dimensionnal grid of parameters. The values of the metric distances are given in a special output file. By default for a complete calculation, the file name is the fdmnes conventional output file name with the suffix *_fit.txt*. For a calculation starting calculating the convoluted spectra, the default output file name is *fdmfit_out.txt*. This name can be modified using a specific keyword.

To do this task different keyword must be added in the main indata file:

Experiment <i>Nom_exp</i>	→ Keyword preceding the file names containing the experimental spectra.
Gen_shift <i>7108 7114 21</i>	→ Minimum and maximum energy shift between calculation and experiment and number of value to test
Metric_out <i>File_name_fit.txt</i>	→ Keyword preceding the output file name containing the metric distances
Parameter <i>Par_Gamma_max</i>	→ Keyword preceding any group of correlated parameters → Keyword to specify the convolution broadening Γ_m as a parameter

15. 25. 5 → First and last values and number of values

Parameter

Par_Posx

-0.05 0.05 3

Par_Posy

-0.05 0.05 3

In the previous example the parameters *Posx* and *Posy* of the atom 1 are completely correlated. Thus the atom is displaced along the diagonal.

The parameters can be fitted are:

For the convolution:

Par_ecent → Central energy for the arctangente
Par_elarg → Energy width for the arctangente
Par_efermi → Fermi energy
Par_gamma_hole → Hole width
Par_gamma_max → Maximum width for the final states
Par_gauss → Gaussian width (or resolution)
Par_shift → Energy shift
Par_aseah → First parameter of the Seah-Dench formula

For the spectra calculation:

Par_a → Contraction or expansion of the mesh parameter a in %
Par_b → Contraction or expansion of the mesh parameter b in %
Par_c → Contraction or expansion of the mesh parameter c in %
Par_abc → General contraction or expansion in %
Par_anga → Value of the unit mesh angle α
Par_angb → Value of the unit mesh angle β
Par_angc → Value of the unit mesh angle γ
Par_poporb → Orbital occupancy
Par_posx → Atom position along x
Par_posy → Atom position along y
Par_posz → Atom position along z
Par_dposx → shift of the atom position along x from the original position
Par_dposy → shift of the atom position along y from the original position
Par_dposz → shift of the atom position along z from the original position
Par_theta → position along θ for an atom in spherical coordinate
Par_phi → position along ϕ for an atom in spherical or cylindrical coordinate

Under each parameter must be written the first and last values of the parameter followed by the number of value. For the parameters *Par_posx*, *Par_posy*, *Par_posz*, *Par_theta*, *Par_phi* The number of the atom must also be specified. Under the parameter *Par_poporb* must be also set the number of the orbital found in the order given under the keyword *atom*. For example, to specify an exchange of charge between two atoms (titanium and oxygen), one has to write:

Atom

22 3 2 2. → 3d titanium orbital with 2 electrons for the initial occupancy
8 2 1 4. → 2p oxygen orbital with 4 electrons for the initial occupancy

Parameter**Par_poporb**

2. 0. 3 1 → The first orbital under *atom* is the 3d titanium.

Par_poporb

4. 6. 3 2 → The second orbital under *atom* is the 4p oxygen.

By default the metric distances are calculated in all the energy range is the intersection between the experimental and calculated spectra. It is possible to cut the lower or and the higher energy part of the spectra by the use of the keyword:

Emin

-10. → Minimum energy for all the spectra

Emax

100. → Maximum energy for all the spectra

It is possible to have different values for the different spectra:

Emin

-10. -5. -20. -20. → Minimum energy for each spectra

Emax

45. 100. 100. 100. → Maximum energy for each spectra

If the energy of the experimental spectra is in keV and not in eV, put the keyword:

Key

When there are several spectra to compare simultaneously, for the calculation they must be in the same output file. One then must give the number of the column contains the calculated spectra and associate it at the corresponding experimental file. This is done adding a line after each experimental file contains the number of the column. If there are 2 numbers, the first one is the number of the column in the experimental file, the second one being the number of the column in the calculated file. When there is no number, this means than in both experimental and calculated file, the spectra are in the second column, the first one being the energy. Example:

Experiment → keyword

Nom_exp_1.txt → Name of the file containing the first experimental spectra

2 → Number of the column in the calculated file containing the corresponding spectra

Nom_exp_2.txt → Name of the file containing the second experimental spectra

3 → Number of the column in the calculated file containing the corresponding spectra

Nom_exp_2.txt → Name of the file containing the third experimental spectra

4 3 → Numbers of the column in the experimental file *Nom_exp_2.txt*, followed by the number of the column in the calculated file

By default, three confidence factors are used. A fourth one “R_{xg}” can also be calculated. It is equivalent to R_x but with a unique normalization factor between experiment and calculation. When there is a single spectra, R_{xg} is equivalent to R_x. To get this confidence factor put the keyword:

R_{xg}

E- Extraction of DAFS scans and spectra

When realizing a DAFS simulation with azimuth dependence (scan), the output scan file after convolution contains the intensity of all the reflections at all energies and all azimuth angles. Often, it is useful to extract from this big file some spectra at specific azimuth angle or some scan at specific energy. It is what is done in this part with the following indata file:

!Indata file for FDMNES, Selection part

Selec_inp → keyword for the input file name (output of the convolution part)
xanout/fe3o4/2008_bland/cc_1221_1221_orig_1m10_scan_conv.txt

Selec_out → keyword for the output file name
xanout/test/fe3o4_scan_selec_conv.txt

Energy → keyword for the selected energy for the scan
 4. → value of the energy for the scan in eV.

Reflection → keyword for the selected reflections
 2 5 6 9 → number of the selected reflections

End

The azimuth scan is extracted at the energy (in eV) given after keyword “*Energy*”.

It is also possible to extract spectra at a given azimuth angle. For this, no keyword “*Energy*” but the keyword “*Azimuth*” and next line the value of the selected azimuth:

Azimuth
 30. → Value of the azimuth

It is possible to have several reflections in the output file at one energy (or one azimuth). It is also possible to have several energies but only one reflection for the scan or several azimuth angles but one reflection for the spectra.

Energy
 4 7.5 → selected energy for the scan

Reflection
 2 → reflection number 2 selected

or:

Reflection
 2 → reflection number 2 selected

Azimuth
 30. 60. 90. → Azimuth selected

F- File Atomic electronic density

When using atomic electronic density furnished by the user, these ones must be in files having determined formats. One must have one file by type of different atom, the absorber being considered as a different type of atom having the same atomic number than non excited atom. Example of file:

fer neutre 001 (3d6, 4s2)

Orbitale	n	l	j	Popul.	Energie
1S	1.0	0.0	0.0	2.000	-6917.713
2S	2.0	0.0	0.0	2.000	-804.490
2P	2.0	1.0	0.0	6.000	-695.277
3S	3.0	0.0	0.0	2.000	-91.497
3P	3.0	1.0	0.0	6.000	-59.577
3D	3.0	2.0	0.0	6.000	-8.092
4S	4.0	0.0	0.0	2.000	-5.482

NUCLEAR CHARGE= 26.000000

INTEGRAL OF CHARGE DENSITY= 26.000000

----- → all what is before is not red

26 18.00 2 3 2 6.000 4 0 2.000
600

r (A)	rho (ua ⁻³)	psi(3D)	psi(4S)
0.00016112638	11678.65000	0.0000000043	0.0023114080
0.00016413798	11675.19000	0.0000000046	0.0023542610
0.00016720588	11671.67000	0.0000000048	0.0023979030
0.00017033115	11668.09000	0.0000000051	0.0024423470

.....

The file must contains a line beginning by "---". All what is before is not red.

Following line

26 18.00 2 3 2 6.000 4 0 2.000 atomic number, number of heart electrons, number of valence orbitals then quantum numbers (n, l) and population in number of electron in each of these orbitals

Following line:

600 number of radius where are calculated the electronic densities and wave function

Following line:

r (A) rho (ua⁻³) psi(3D) psi(4S) → comment

Following line:

By column: column 1, radius in angstrom,
column 2 total electronic density (ua⁻³), $\sqrt{4\pi r}\psi(r)$
column 3, 4 .. wave function of the valence orbitals multiplied by $\sqrt{4\pi r}$

The absorbing atom (type 1) must include one column more containing the value of the initial wave function (for example 1s for a K threshold) before the absorption of the photon:

fer excite neutre E01 (3d7, 4s2)

<i>Orbitale</i>	<i>n</i>	<i>l</i>	<i>j</i>	<i>Popul.</i>	<i>Energie</i>
1S	1.0	0.0	0.0	1.000	-7204.473
2S	2.0	0.0	0.0	2.000	-859.982
2P	2.0	1.0	0.0	6.000	-763.986
3S	3.0	0.0	0.0	2.000	-96.485
3P	3.0	1.0	0.0	6.000	-64.835
3D	3.0	2.0	0.0	7.000	-9.151
4S	4.0	0.0	0.0	2.000	-5.544

NUCLEAR CHARGE= 26.000000

INTEGRAL OF CHARGE DENSITY= 26.000000

 26 17.00 2 3 2 7.000 4 0 2.000
 600

<i>r (A)</i>	<i>rho (ua-3)</i>	<i>psi(3D)</i>	<i>psi(4S)</i>	<i>pssi</i>
0.00016112638	6572.94900	0.0000000052	0.0023377680	0.0784658600
0.00016413798	6571.00800	0.0000000055	0.0023811100	0.0799206300
0.00016720588	6569.02300	0.0000000058	0.0024252490	0.0814021800
0.00017033115	6567.00900	0.0000000061	0.0024701990	0.0829109400

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